

**nanoGe Fall Meeting**  
Torremolinos - Spain  
October 22 - 26, 2018



**nanoGe**

# « Effect of electronically inert organic spacers on the optoelectronic properties of 2D hybrid perovskites »

**Nadège Marchal**

University of Mons  
Laboratory for Chemistry of Novel Materials

Supervisor: Prof. David Beljonne,  
Co-supervisor: Dr. Claudio Quarti

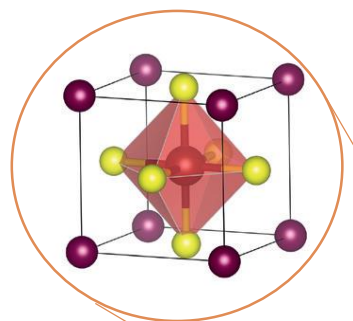
**UMONS**  
Université de Mons

 Faculté  
des Sciences



**materials**  
UMONS RESEARCH INSTITUTE  
FOR MATERIALS SCIENCE  
AND ENGINEERING

# Table of contents

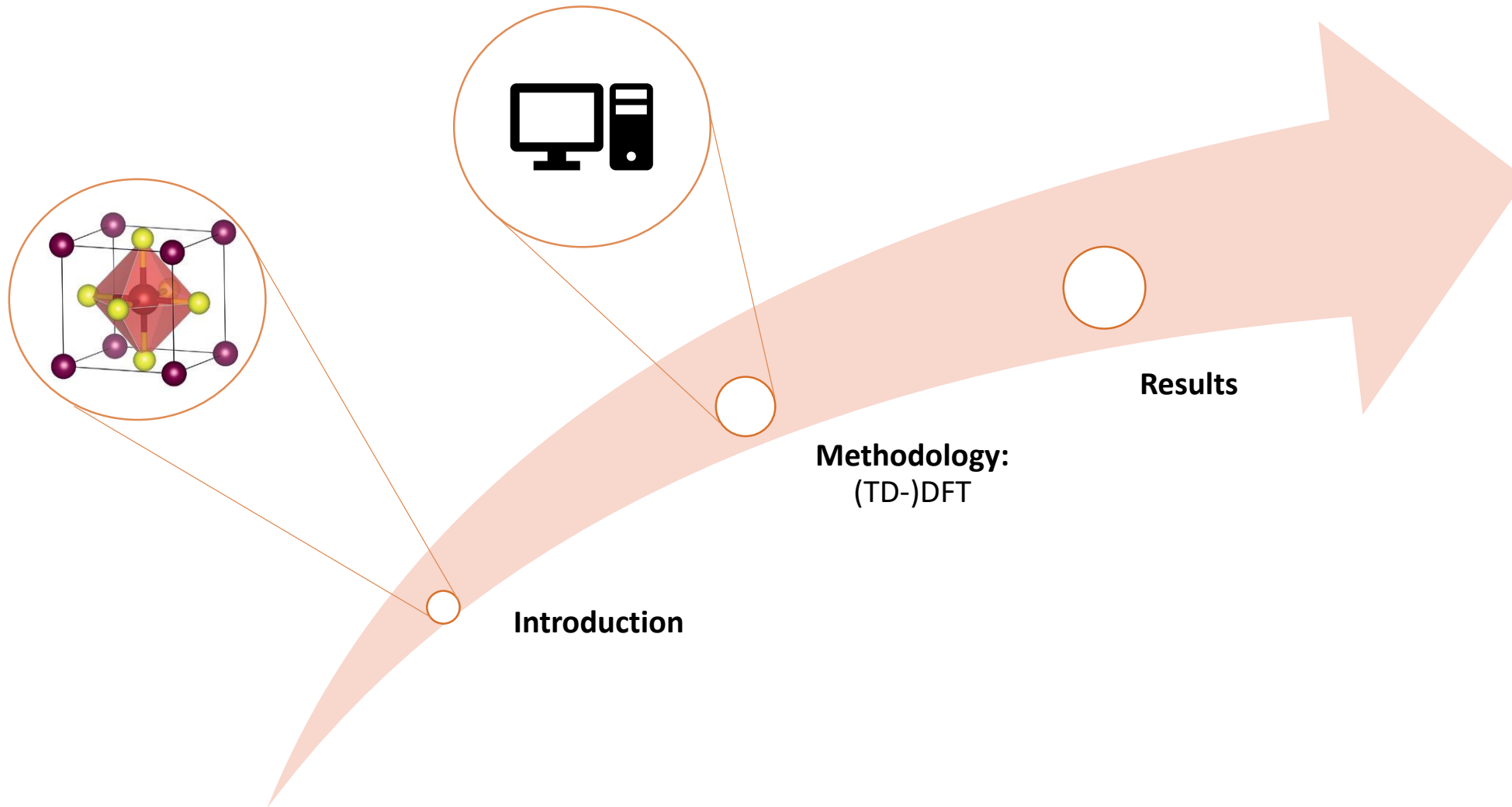


**Introduction**

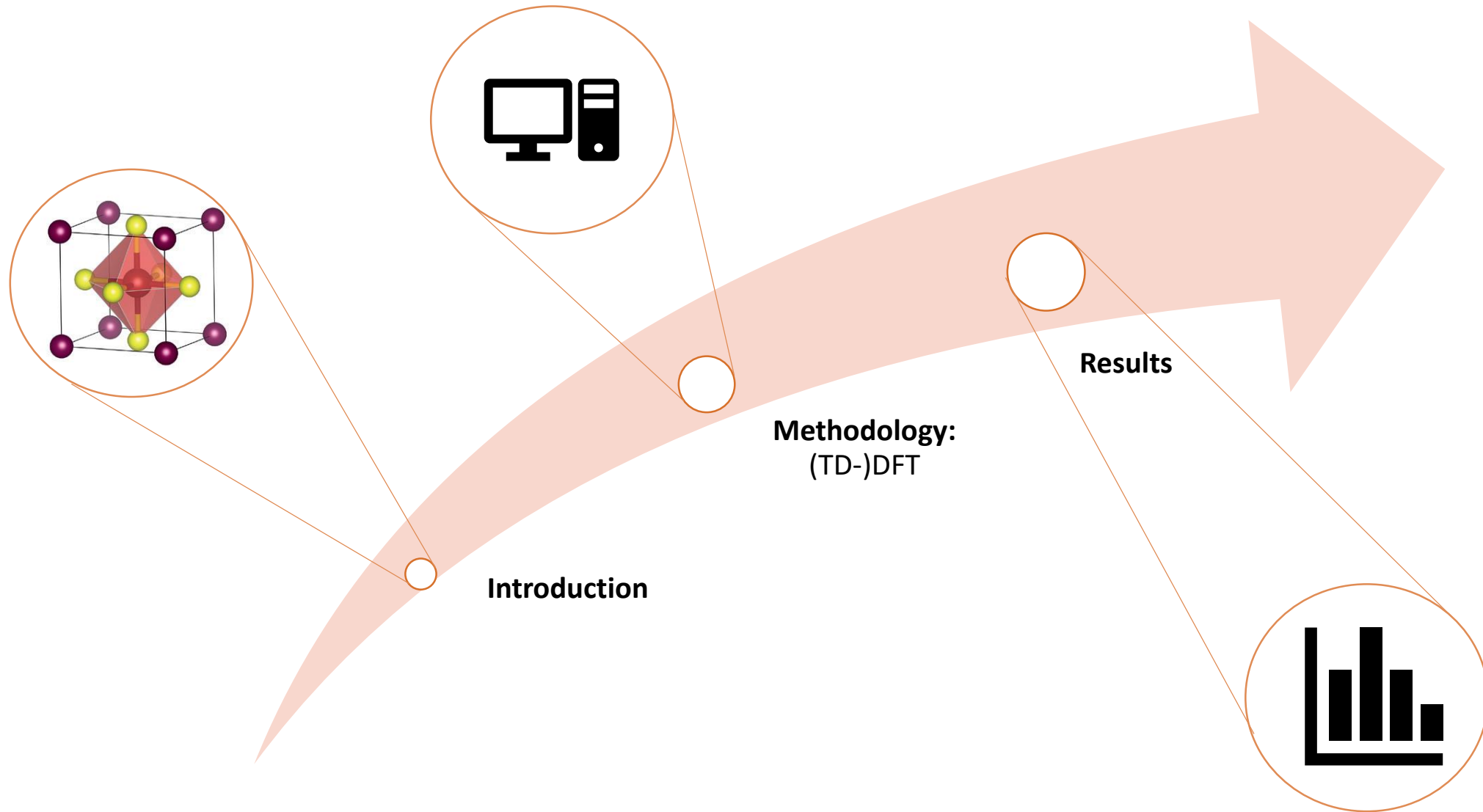
**Methodology:  
(TD-)DFT**

**Results**

# Table of contents



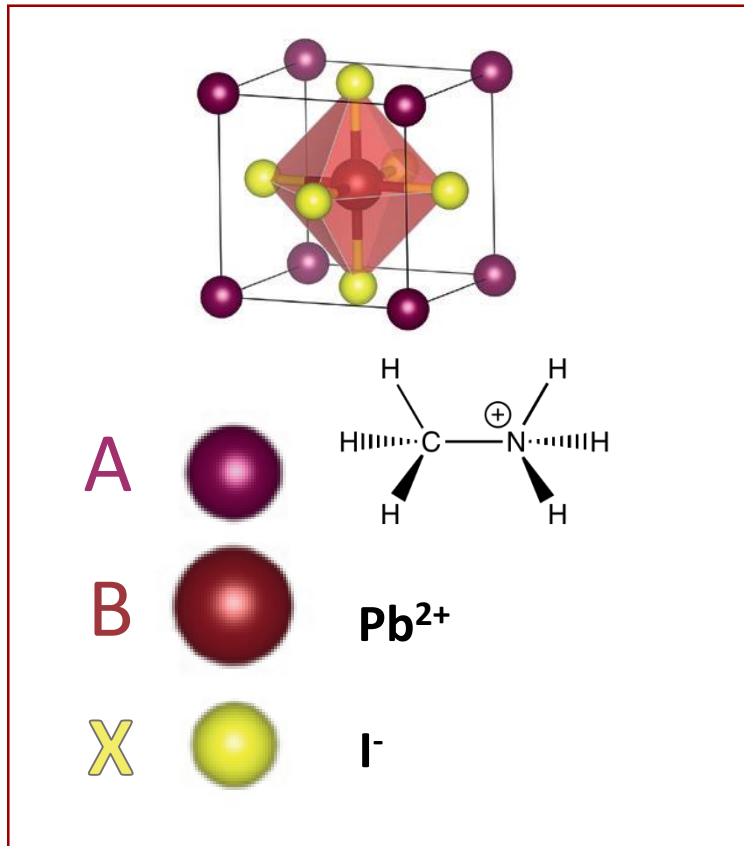
# Table of contents



# Hybrid perovskites ?

Methylammonium Lead iodide  
(MAPbI<sub>3</sub>)

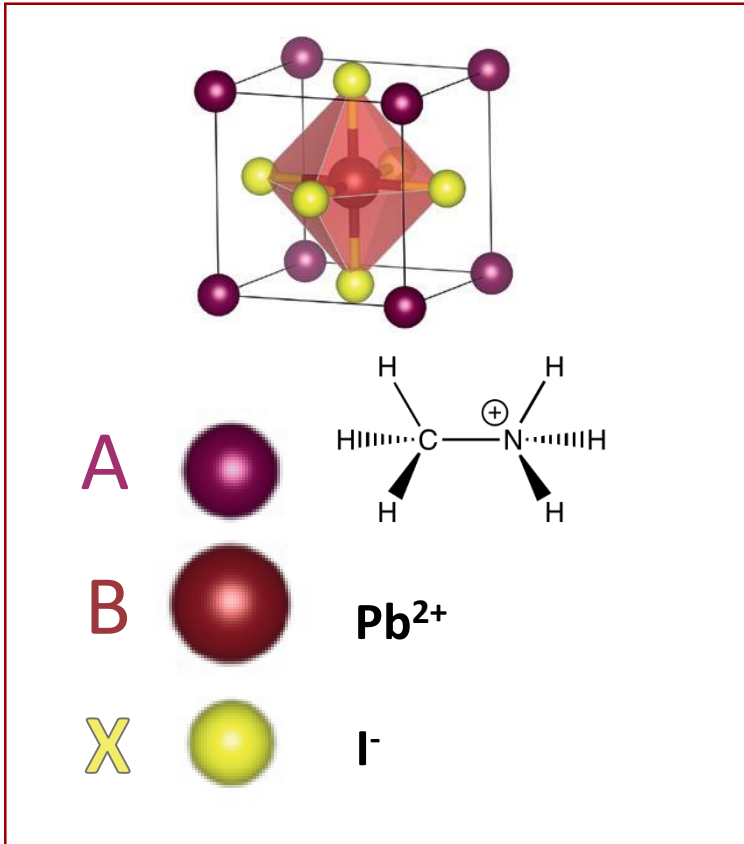
Perovskite structure → ABX<sub>3</sub>



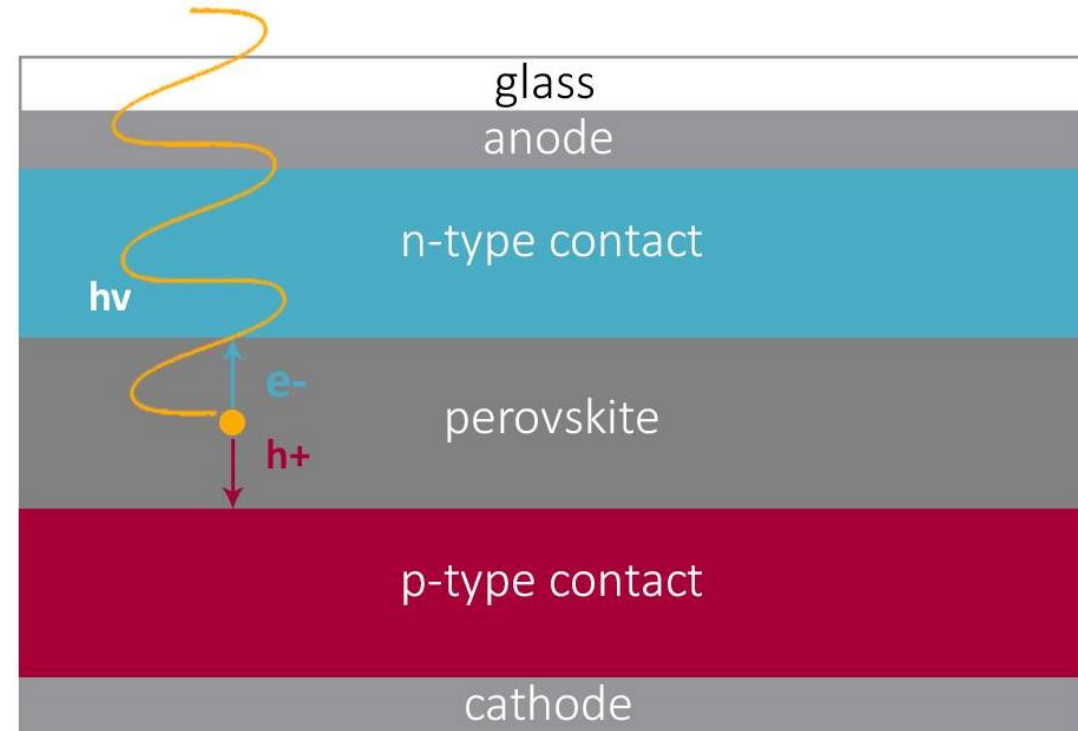
# Hybrid perovskites ?

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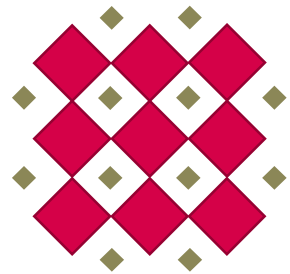
Applications : **Photovoltaics**



# Dimensionality

## 3D Perovskite

- Low temperature solution synthesis
- PV quantum efficiency (23,3%)
- Instability due to ionic diffusion in the inorganic network



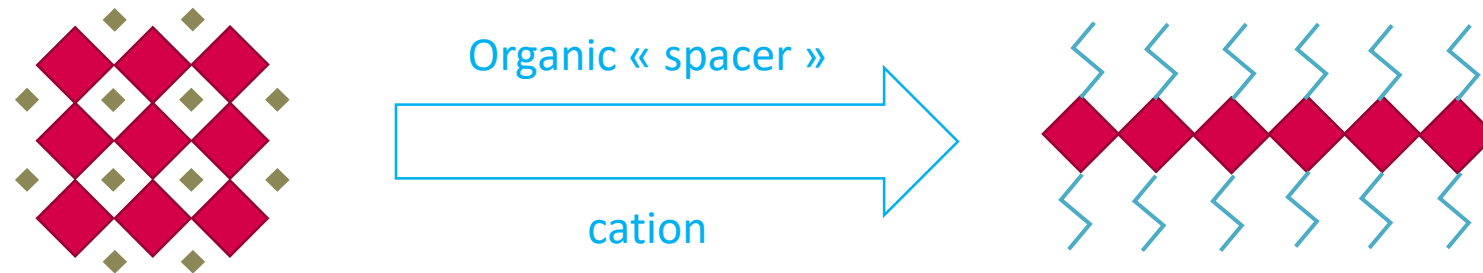
# Dimensionality

## 3D Perovskite

- Low temperature solution synthesis
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## 2D Perovskite

- Low temperature solution synthesis
- PV quantum efficiency (12% 2D - Mixed 3D/2D)
- Higher stability



G. Grancini, et al., *Nat. Comm.* 8 **2017**

M. Yuan, L. N. Quan et al., *Nat. Nanotechnol.* **2016**

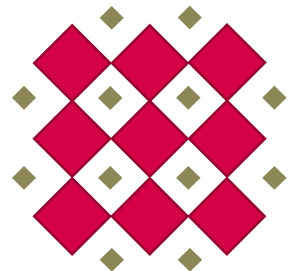


# Dimensionality

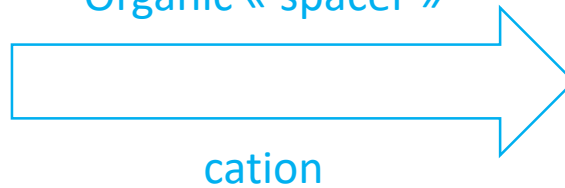
## 3D Perovskite

- Low temperature solution synthesis
- PV quantum efficiency (23,3%)
- Instability due to ionic diffusion in the inorganic network
- Goldsmicht tolerance factor

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \quad 0.7 > t > 1$$



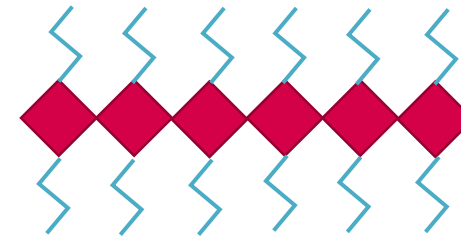
Organic « spacer »



cation

## 2D Perovskite

- Low temperature solution synthesis
- PV quantum efficiency (12% 2D - Mixed 3D/2D)
- Higher stability
- Chemical flexibility



G. Grancini, et al., *Nat. Comm.* 8 **2017**

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# Dimensionality : 3D → 2D

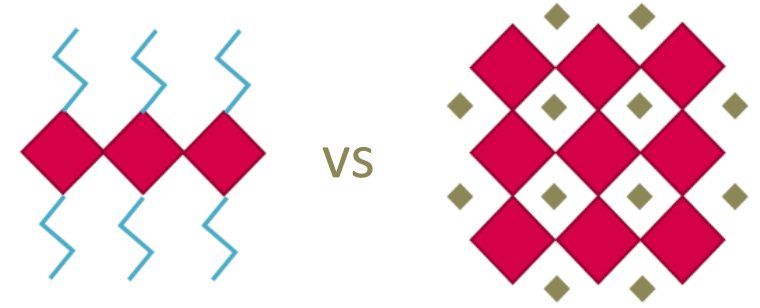
## 1. Anisotropy

→ Limited ionic diffusion, but also limited electron transport

## 2. Chemical flexibility

→ Reduced chemical space for 3D components (some cations, Pb, halide)

↳ Larger number of possibilities for 2D, including the use of functional organic cations



# Dimensionality : 3D → 2D

## 1. Anisotropy

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## 2. Chemical flexibility

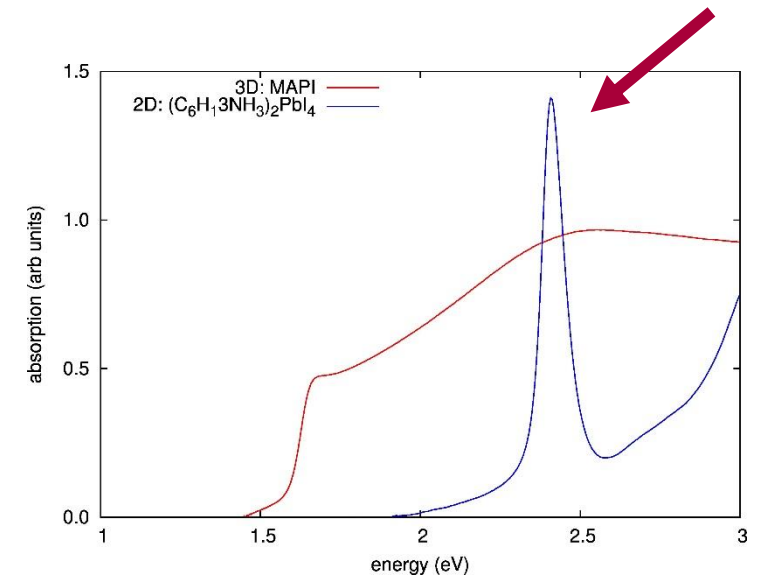
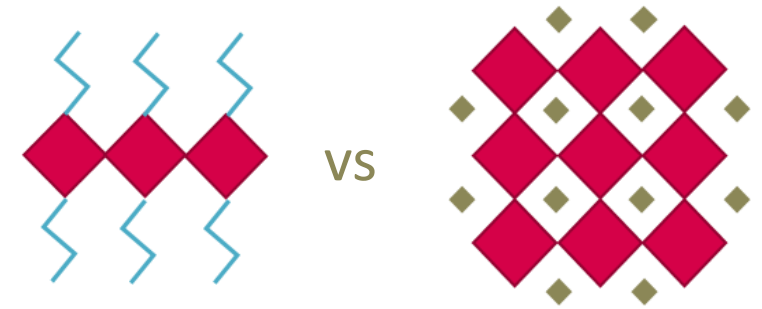
➡ Reduced chemical space for 3D components (some cations, Pb, halide)

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## 3. Electronic/dielectric confinement

➡ Stronger excitonic effect

➡ Larger bandgap (BG)



# Dimensionality : 3D $\rightarrow$ 2D

## 1. Anisotropy

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## 2. Chemical flexibility

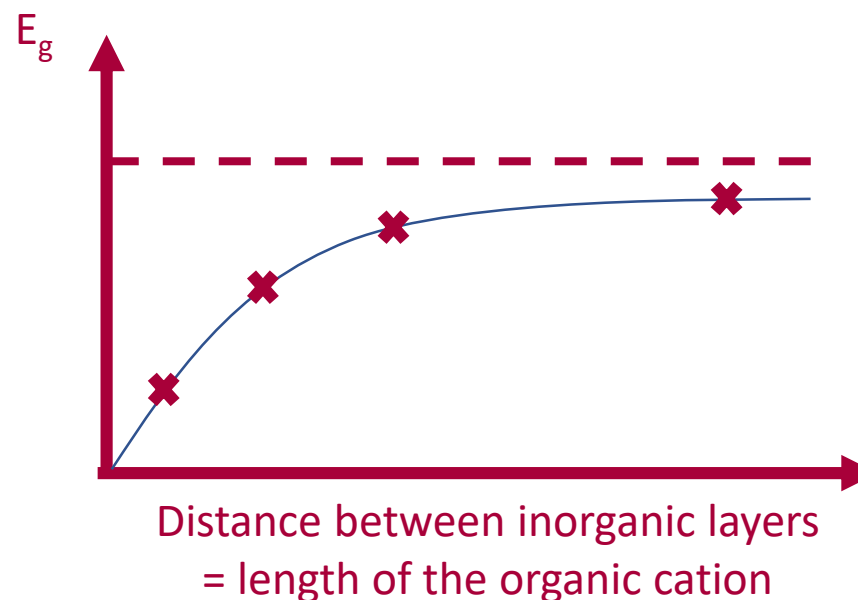
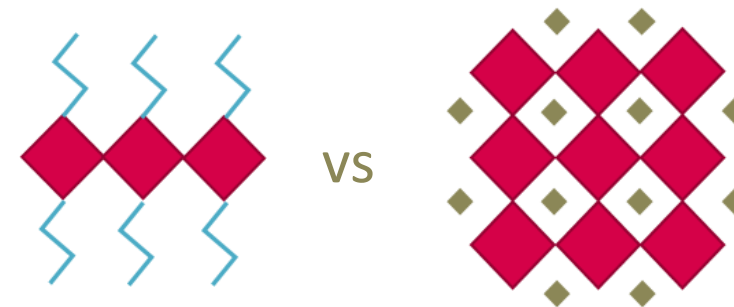
$\Rightarrow$  Reduced chemical space for 3D components (some cations, Pb, halide)

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## 3. Electronic/dielectric confinement

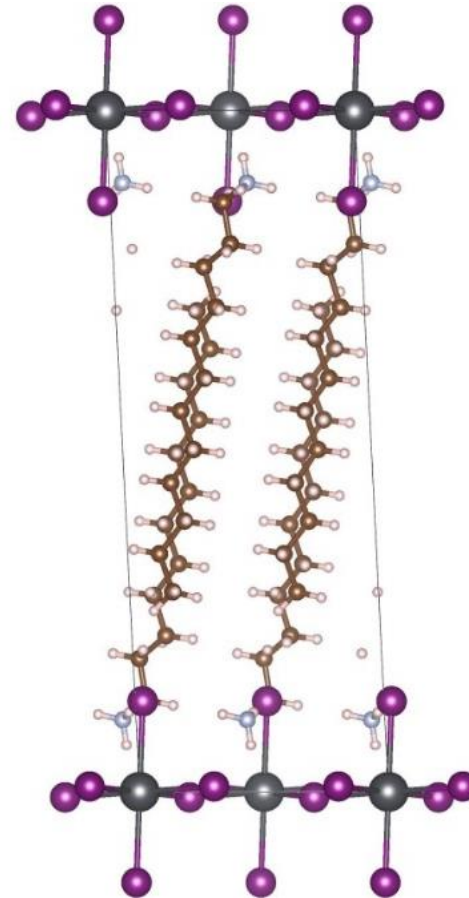
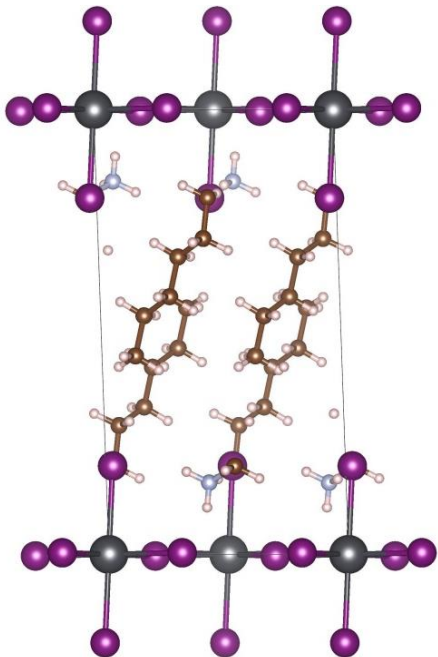
$\Rightarrow$  Stronger excitonic effect

$\Rightarrow$  Larger bandgap (BG)



Is it the only effect of the nature (length) of the organic cation?

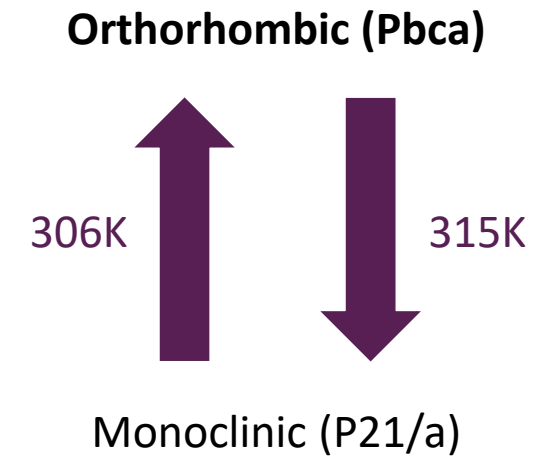
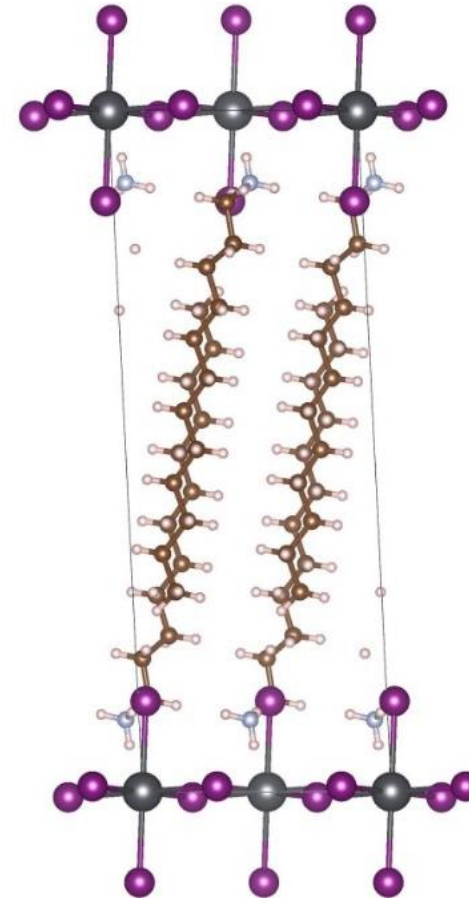
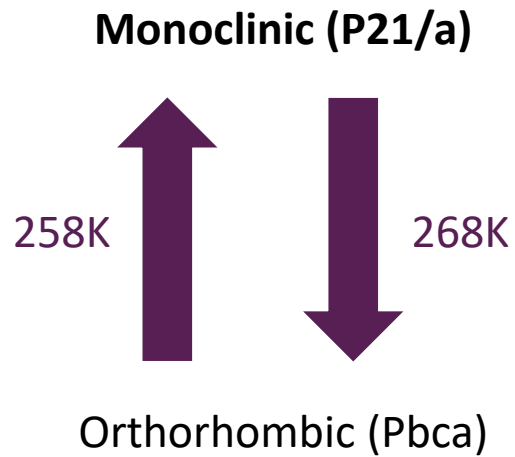
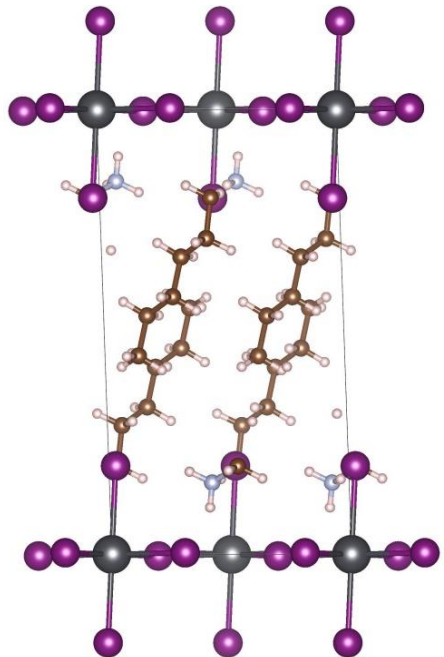
# Studied systems : $(C_nH_{2n+1}NH_3)_2PbI_4$



D. G. Billing, A. Lemmerer, *Acta Crystallogr. Sect. B: Struct. Sci.* **2007**

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# Methodology



- **Materials modeling & electronic properties**
- **Density Functional Theory (DFT) : Plane Wave basis set (PW)**
- **PBE (GGA functional)**

**DFT**

(static, ground state)

# Methodology



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- Density Functional Theory (DFT) : Plane Wave basis set (PW)
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## DFT

(static, ground state)

- Grimme correction:  $E_{DFT-D} = E[\rho] + E_{disp}$  et  $E_{disp} = -s_6 \sum_{i=j}^{N_{at}-1} \sum_{j=i+1}^{N_{at}} \frac{C_6^{ij}}{R_{ij}^6} \frac{1}{1 + e^{-d(\frac{R_{ij}}{R_r}-1)}}$



# Methodology



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- Spin-orbit coupling (SOC)
- Hybrid functional (PBE0)

# Methodology



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- Density Functional Theory (DFT) : Plane Wave basis set (PW)
- PBE (GGA functional)



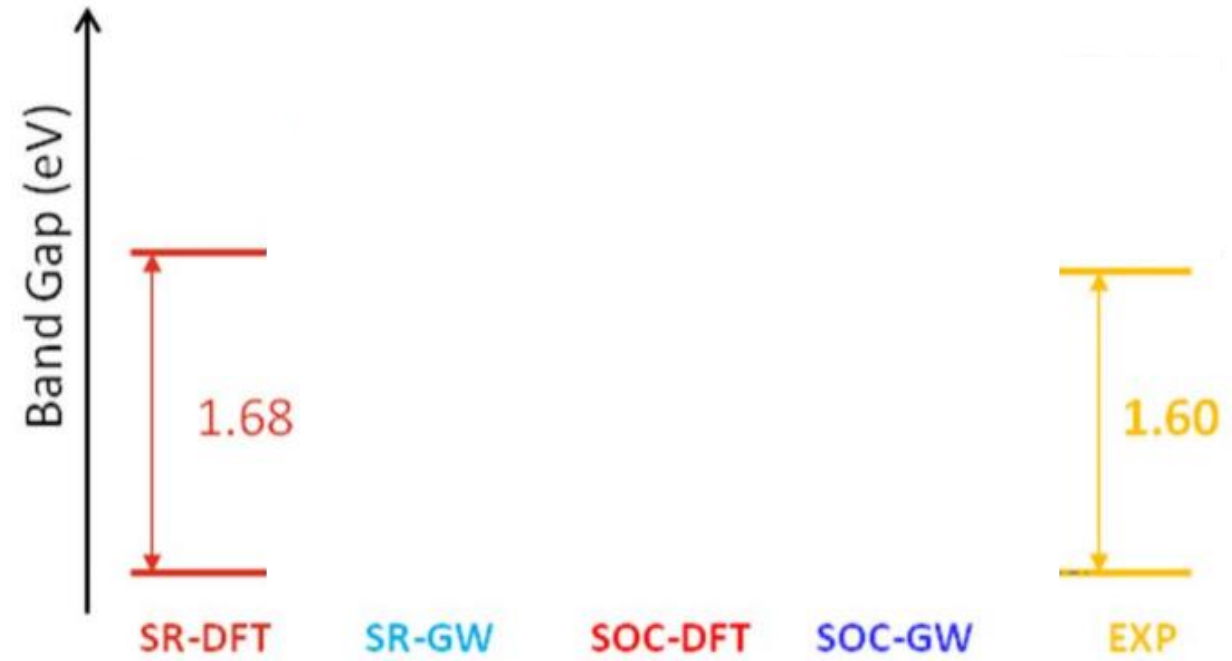
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# Methodological study

3D CASE



- PBE (GGA) agrees with experimental BG



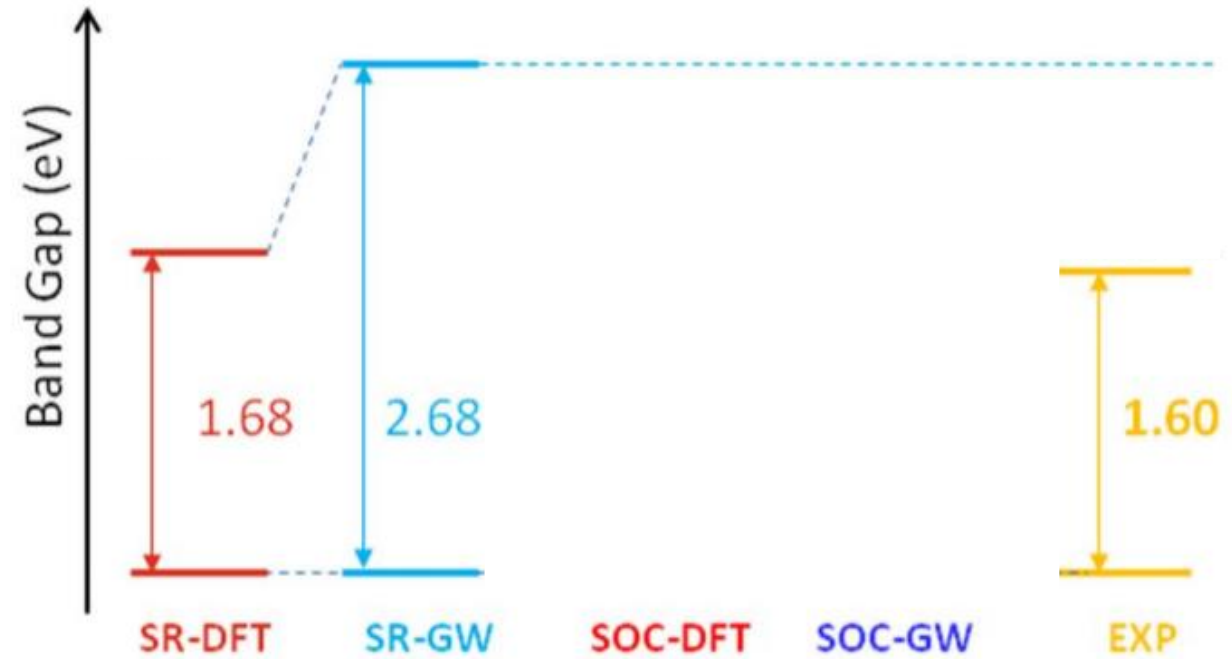
(P. Umari et al., Sci. Rep. 4, 4467, 2014)

# Methodological study

3D CASE

MAPbI<sub>3</sub>

- PBE (GGA) agrees with experimental BG
- GW approximation → BG opening of 1 eV



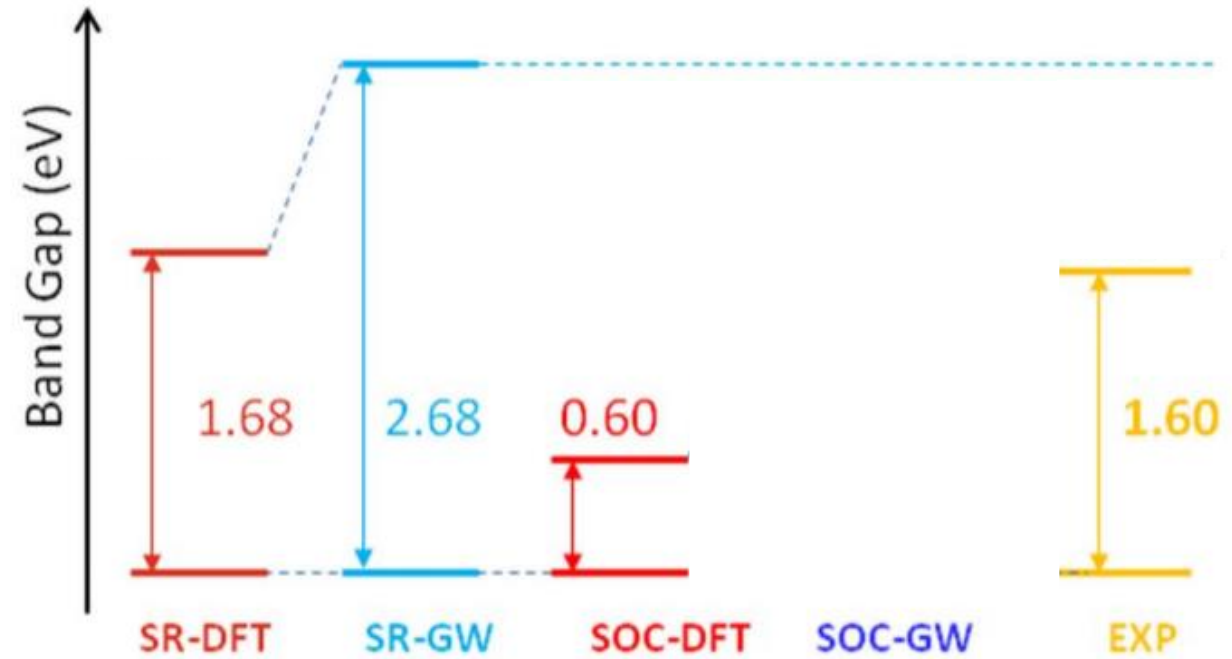
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# Methodological study

3D CASE

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- PBE (GGA) agrees with experimental BG
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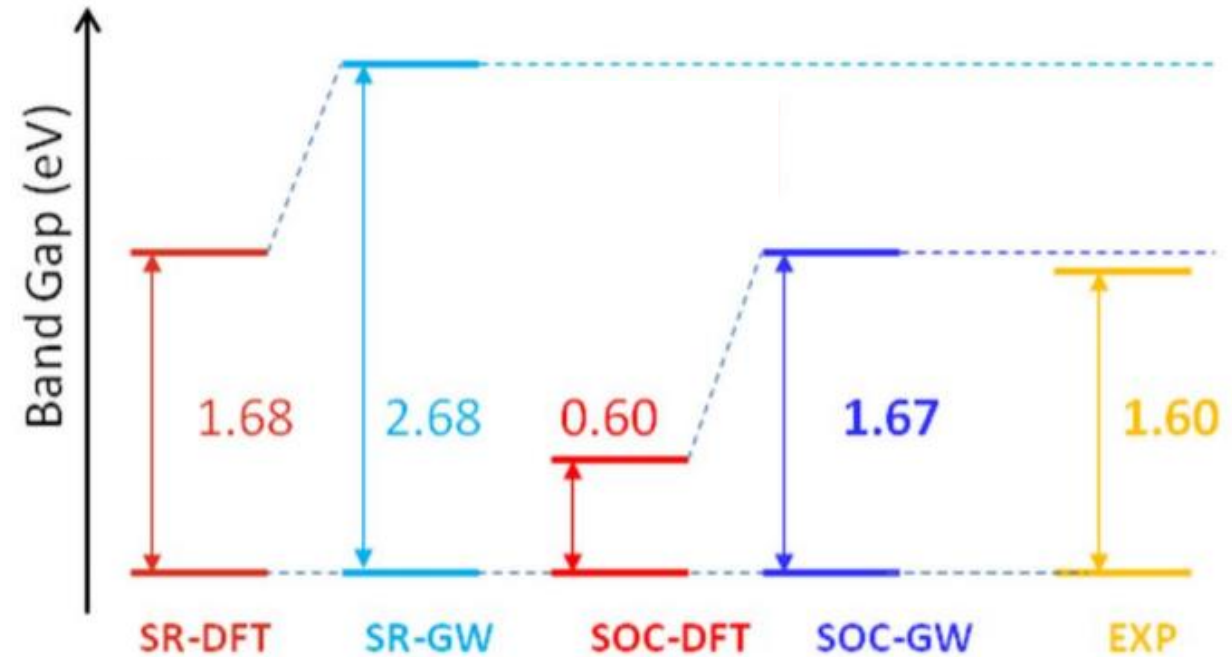
# Methodological study

3D CASE

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↳ Compensation of errors

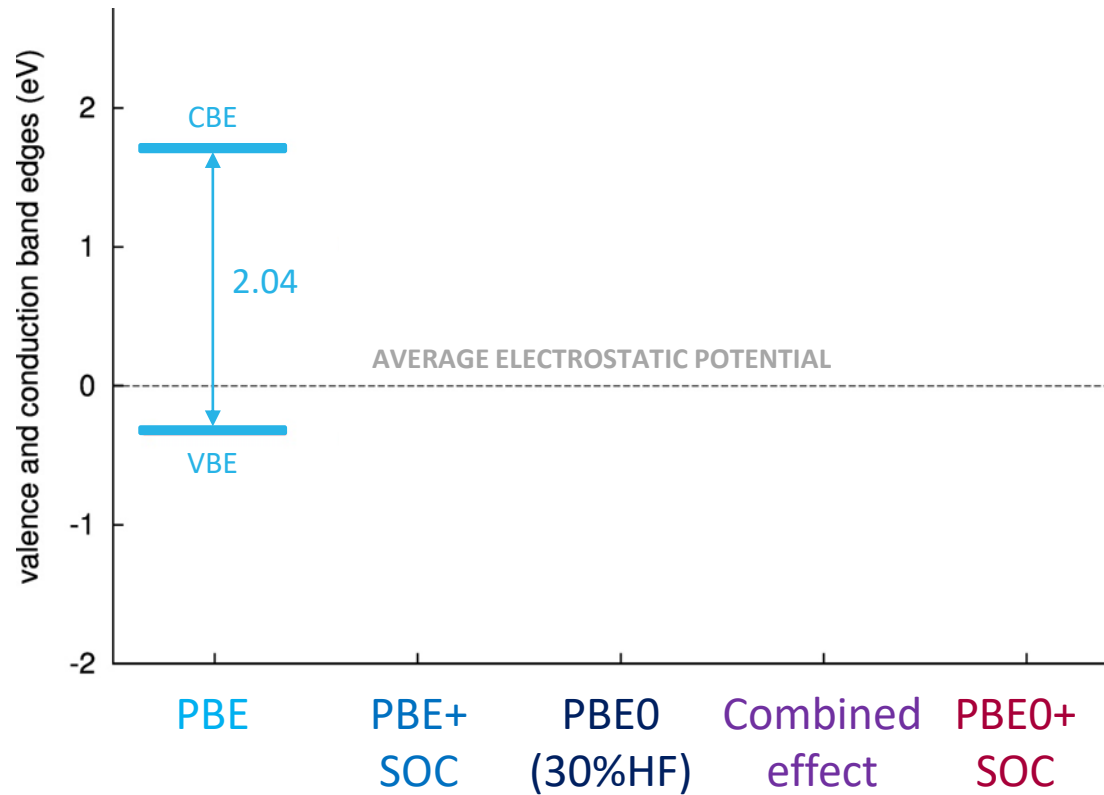


(P. Umari et al., Sci. Rep. 4, 4467, 2014)

→ PBE agrees with SOC-GW and is in good agreement with experimental data

# Methodological study

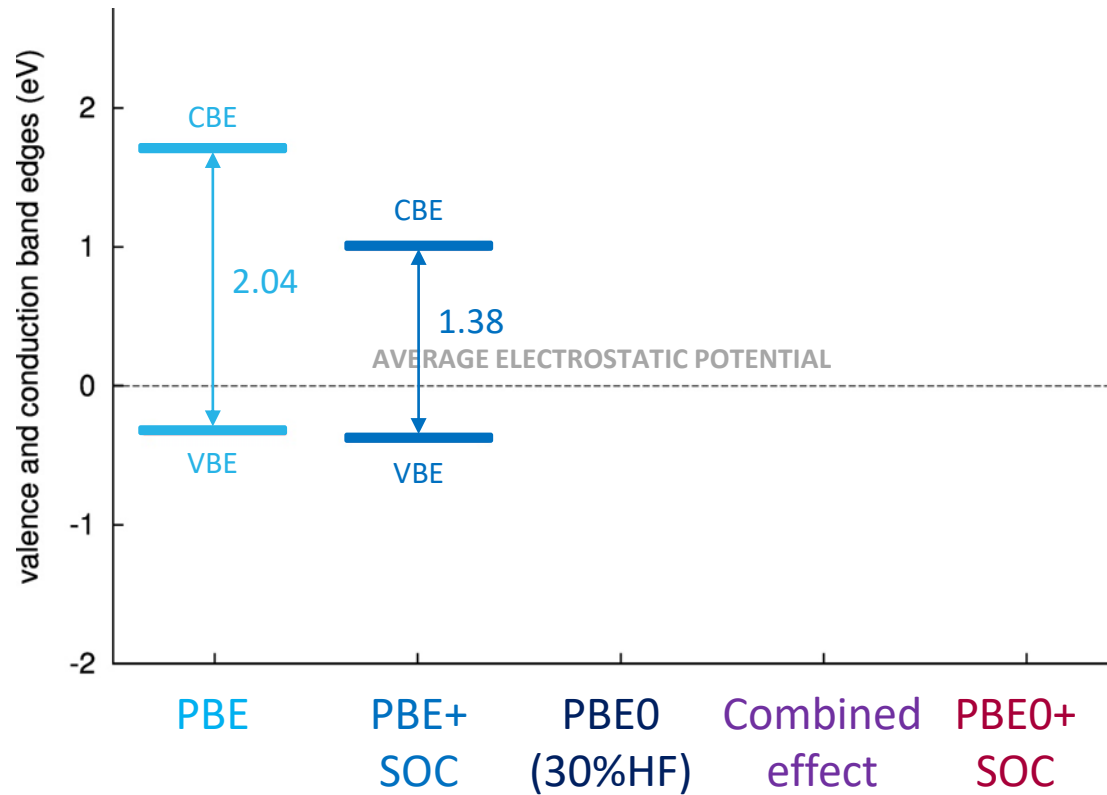
2D CASE



- PBE (GGA) BG does not agree with experimental data (2.7 eV)

# Methodological study

2D CASE

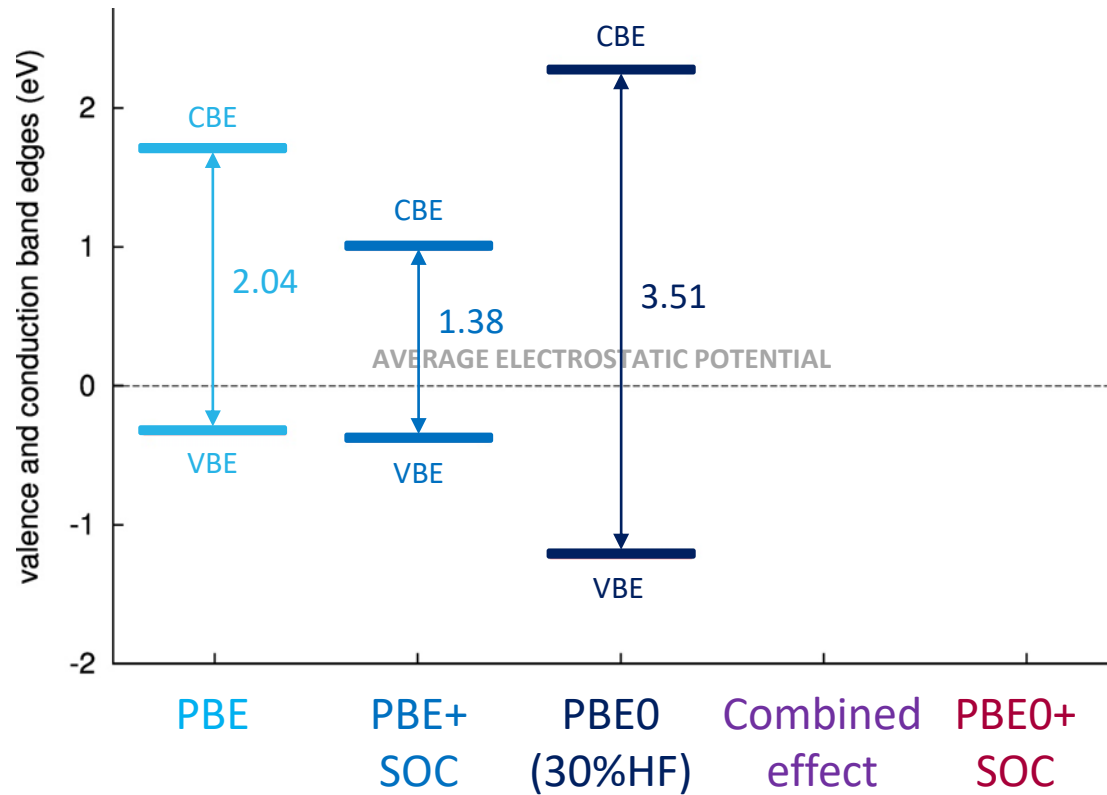


- PBE (GGA) BG does not agree with experimental data (2.7 eV)
- SOC → BG closing of 0.66 eV

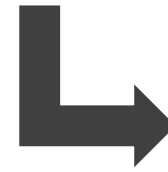


# Methodological study

2D CASE



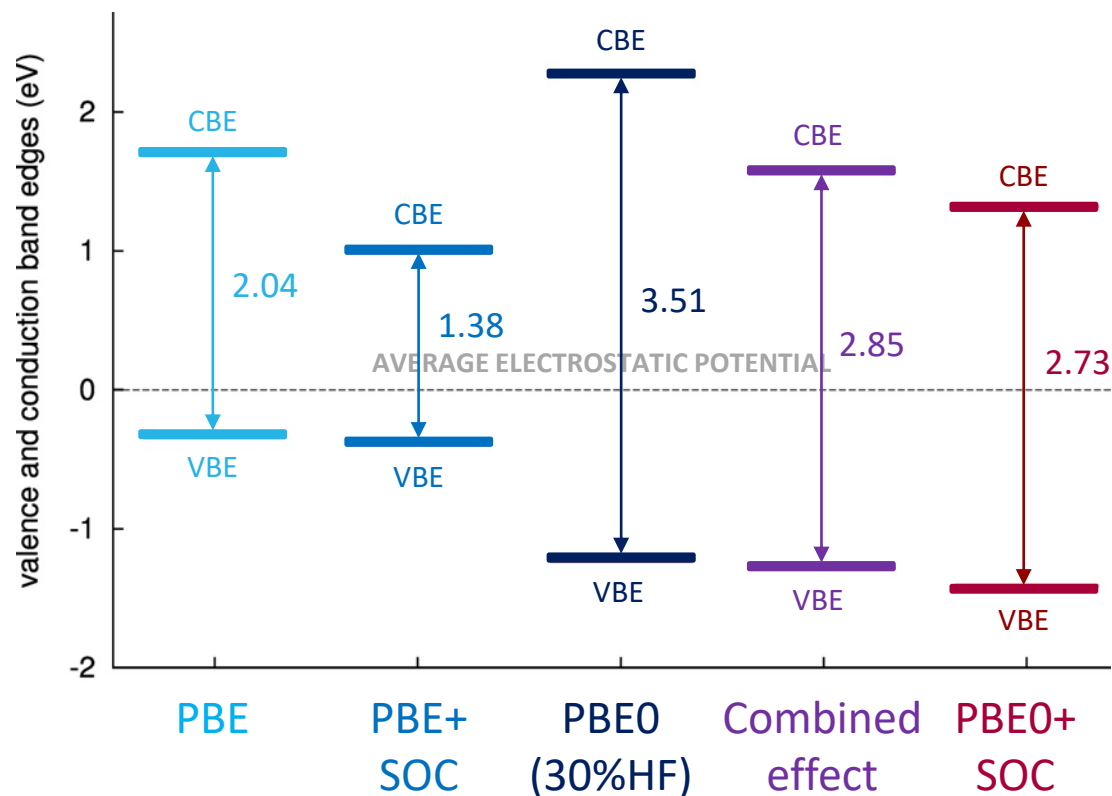
- PBE (GGA) BG does not agree with experimental data (2.7 eV)
- SOC → BG closing of 0.66 eV
- PBE0 → BG opening of 1.47 eV



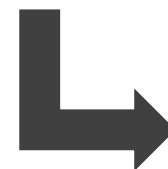
**No compensation of errors**  
Better Exchange interaction description  
overcomes SOC correction

# Methodological study

2D CASE



- PBE (GGA) BG does not agree with experimental data (2.7 eV)
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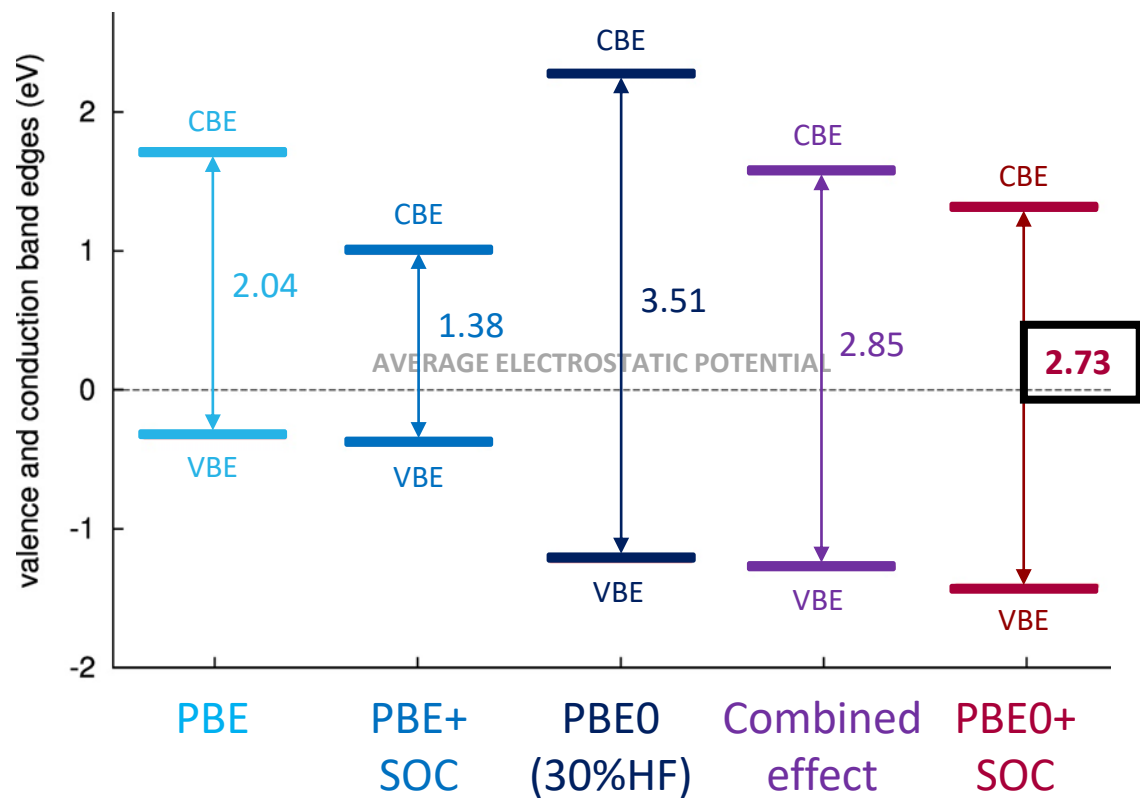


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- Calculation with both corrections lead to BG close to the average of the contributions from the separated calculations

# Methodological study

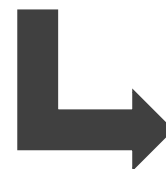
2D CASE



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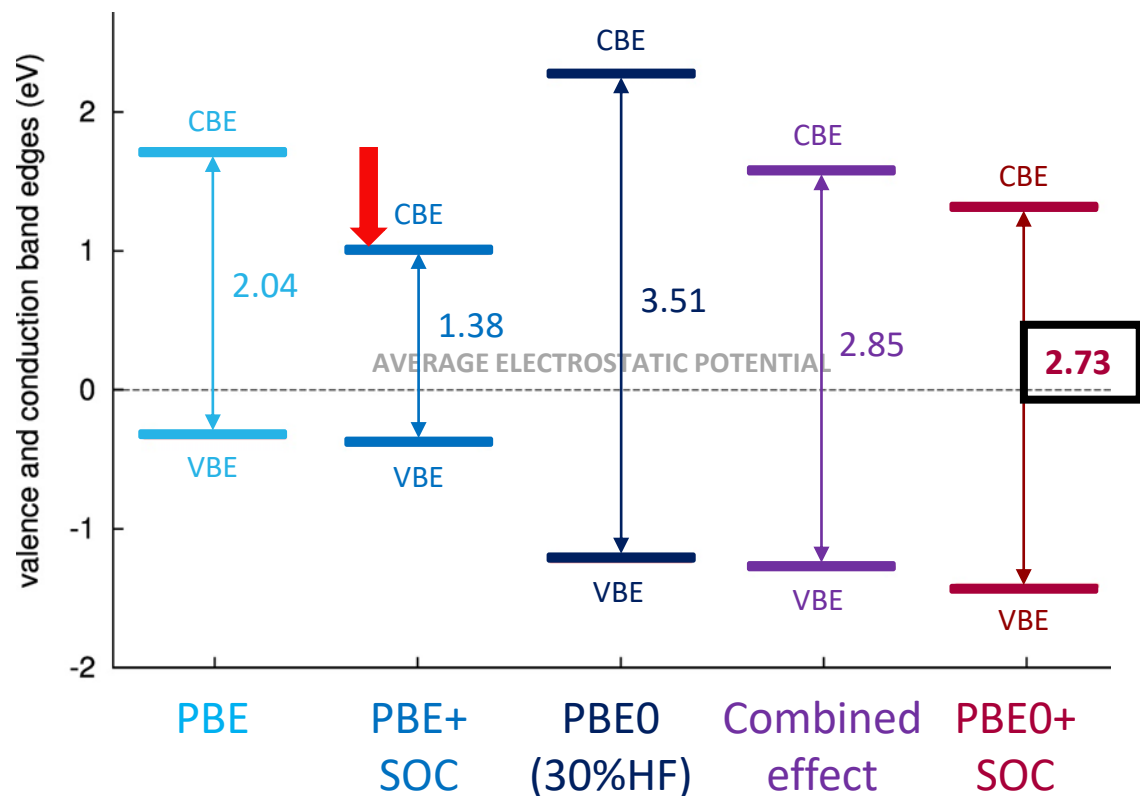
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# Methodological study



Effect of SOC mostly on CBE (splitting of  $j=1/2$  and  $3/2$  on Pb)

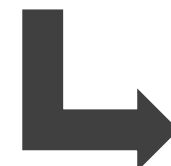
2D CASE



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# Methodological study

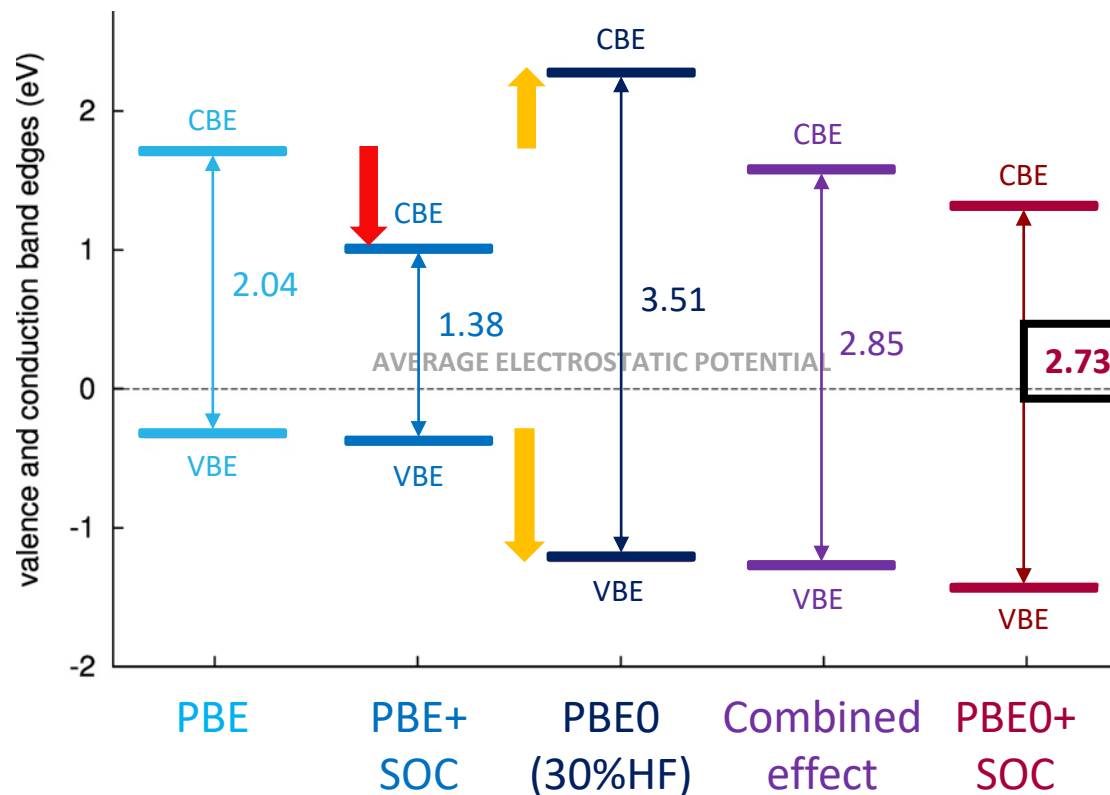
2D CASE



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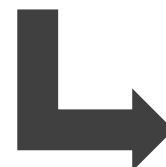
Effect on PBE0 on both bands, but more on VBE



- PBE (GGA) BG does not agree with experimental data (2.7 eV)

- SOC → BG closing of 0.66 eV

- PBE0 → BG opening of 1.47 eV

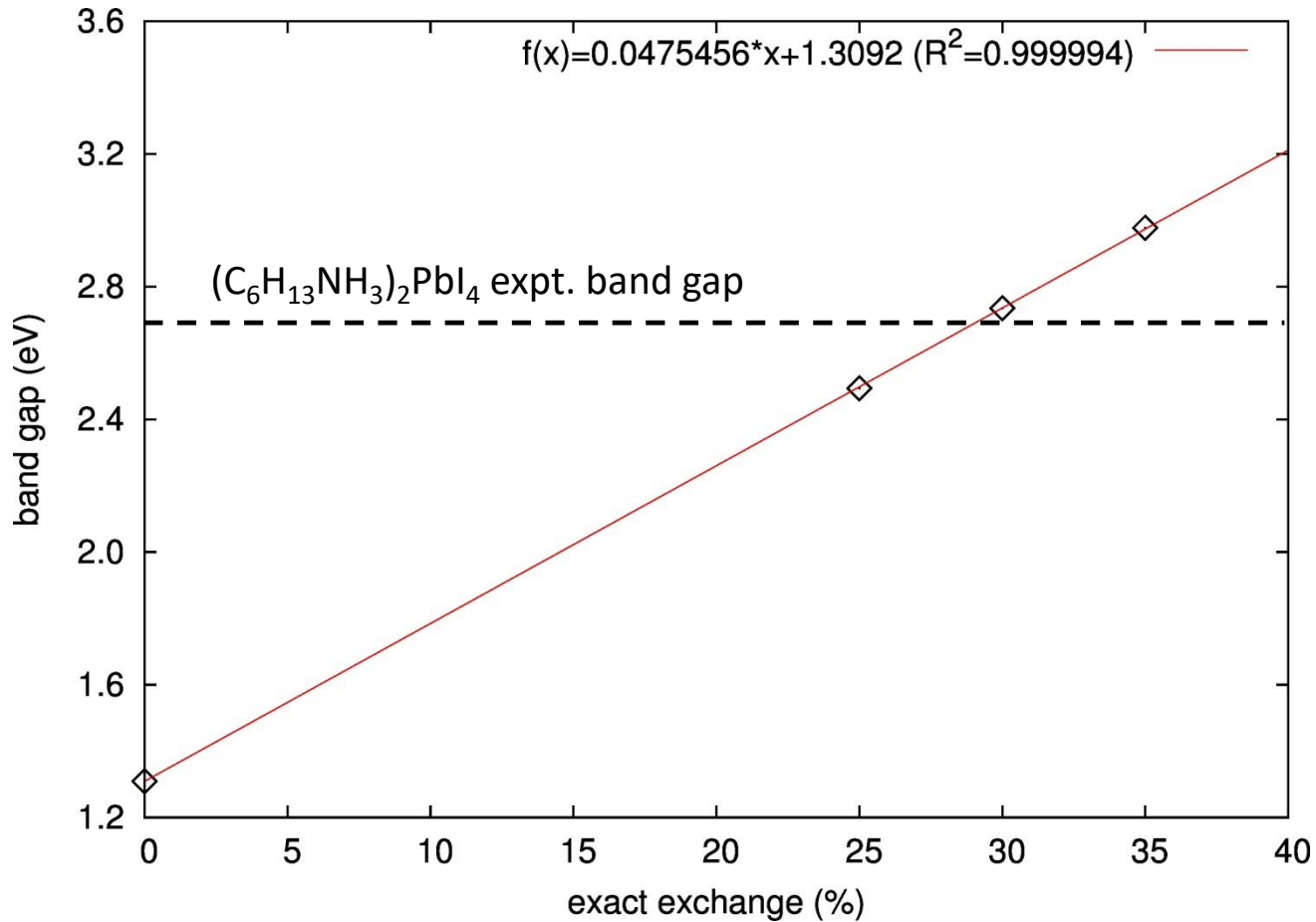


**No compensation of errors**  
Better Exchange interaction description overcomes SOC correction

- Calculation with both corrections lead to BG close to the average of the contributions from the separated calculations

# Methodological study

2D CASE



PBE0 calculation

→ The best agreement with the experimental BG (2.7 eV) is at **30% of Hartree-Fock exchange**

# Electronic properties

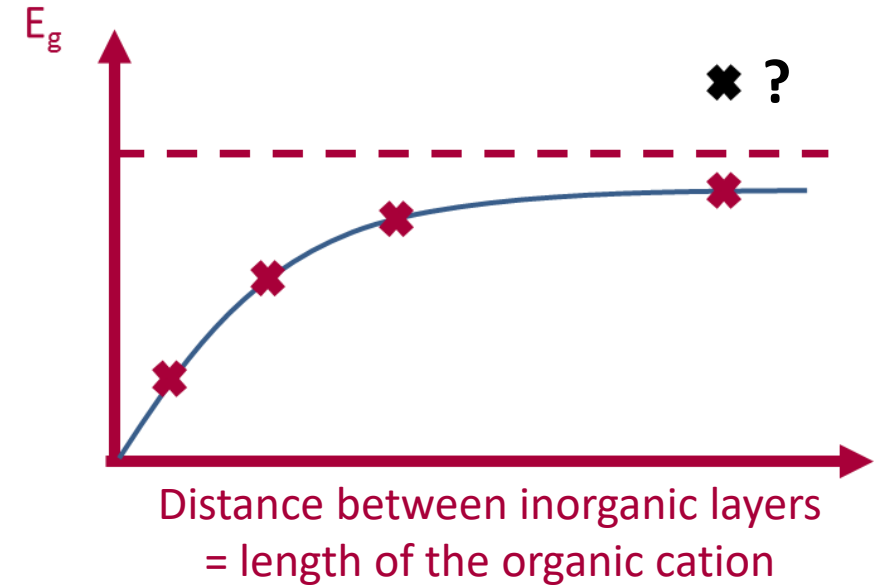
In (100) direction  
←→

PBE	$E_g$ (eV)	$m^*_h$	$m^*_e$
C6 (monoclinic)	2.06	0.24	0.17
C6 (orthorhombic)	2.04	0.24	0.17
C12 (monoclinic)	2.05	0.22	0.17
C12 (orthorhombic)	<b>2.38</b>	<b>0.40</b>	<b>0.24</b>
MAPI	1.6	0.15	0.13

# Electronic properties

In (100) direction  


PBE	$E_g$ (eV)	$m_h^*$	$m_e^*$
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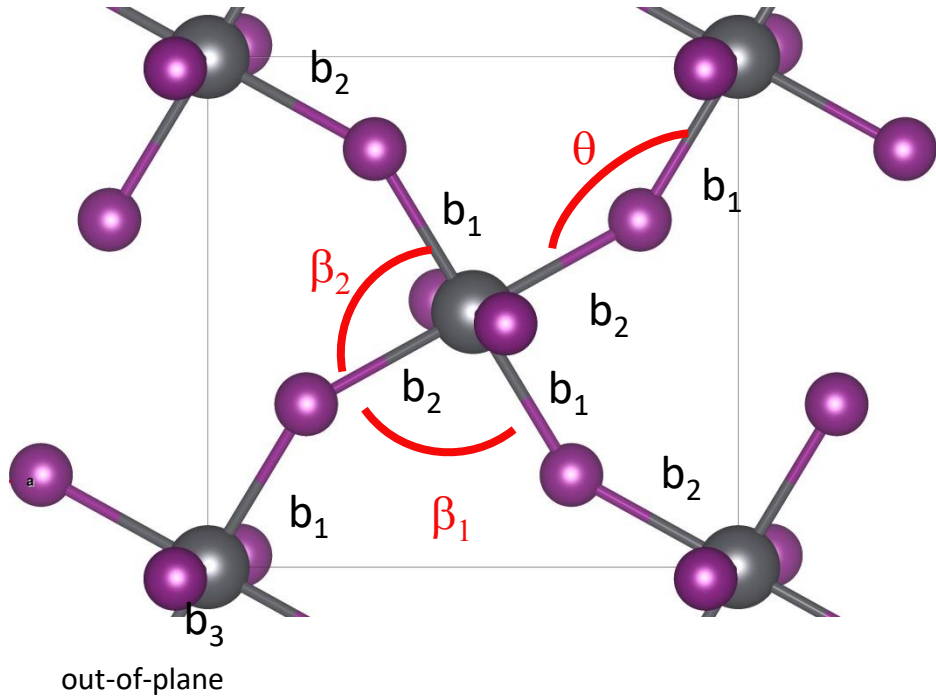


→ Different behaviour of C12 (orthorhombic) → WHY ?

→ Same observations at other levels of theory (SOC – PBEO)



# Structural effect

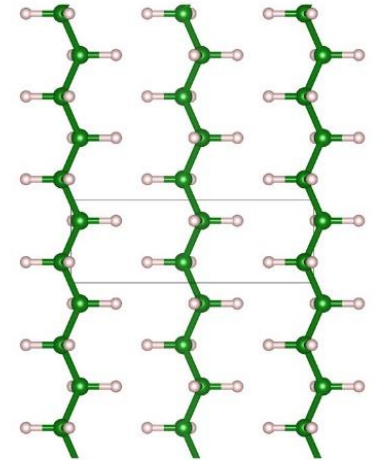


		BG	$b_1$	$b_2$	$b_3$	$\theta$	$\beta_1$	$\beta_2$
C6	mono	2.06	3.19	3.18	3.26	152	92	88
	ortho	2.04	3.20	3.20	3.25	154	89	91
C12	mono	2.05	3.22	3.21	3.22	153	88	92
	ortho	<b>2.38</b>	3.23	3.24	3.24	<b>143</b>	<b>94</b>	<b>86</b>

# Influence from the organic component

Side view

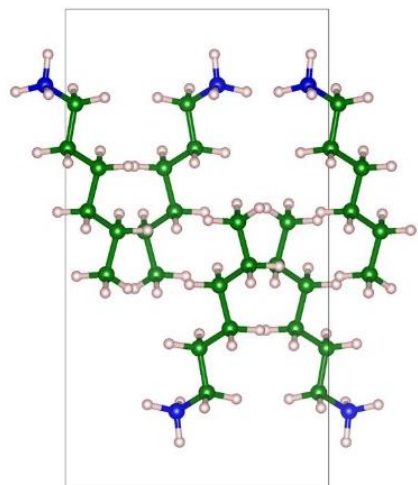
polyethylene



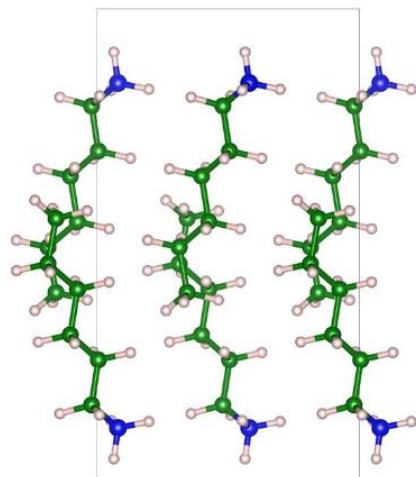
# Influence from the organic component

Side view

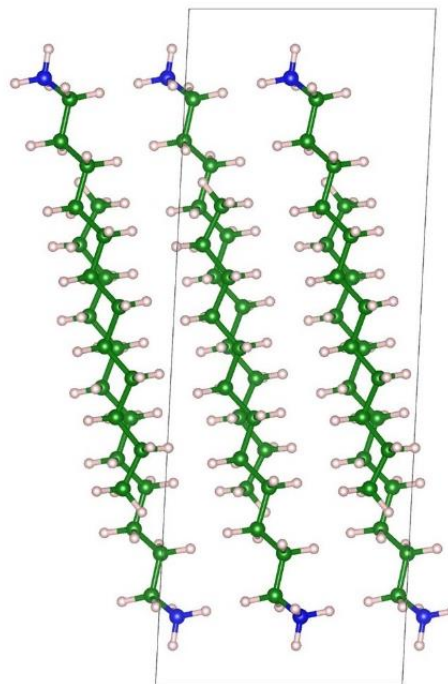
C6 mono



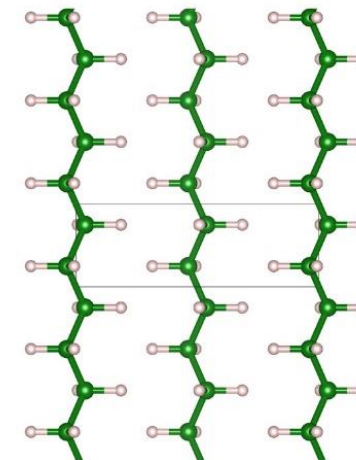
C6 ortho



C12 mono

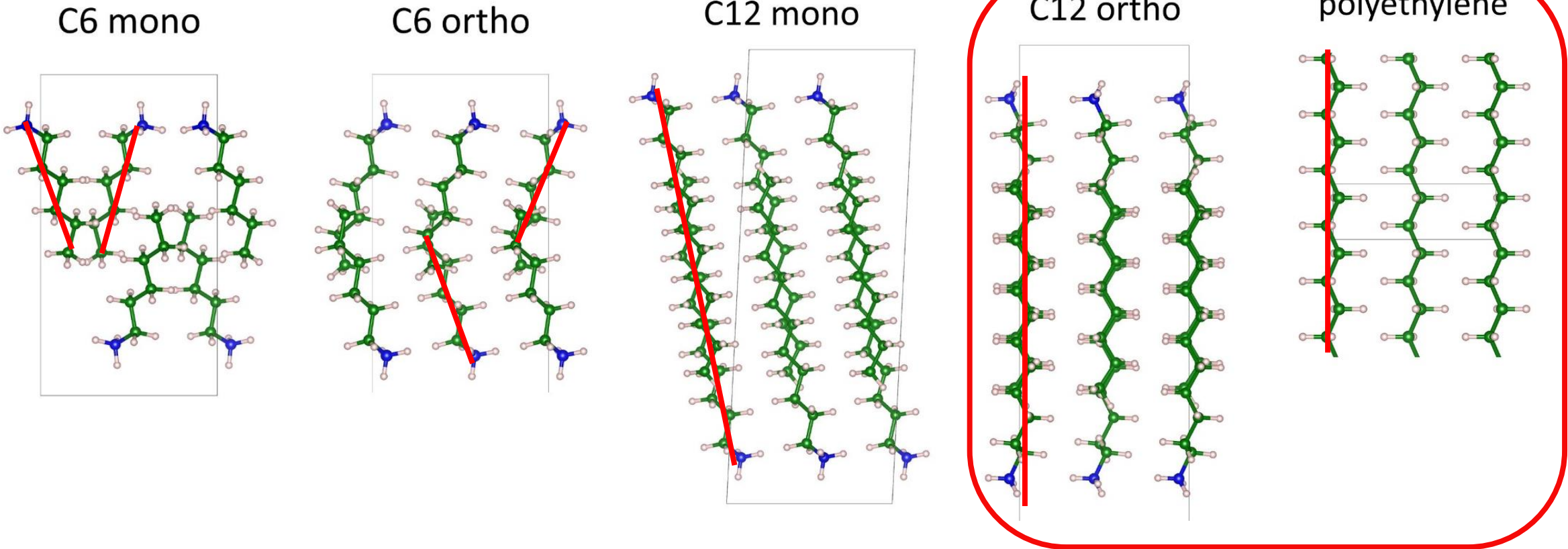


polyethylene



# Influence from the organic component

Side view

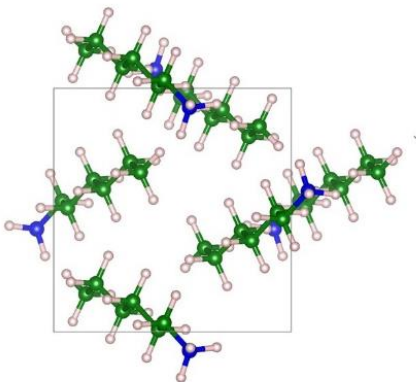


→ Alkyl chains stacked like PE in C12 orthorhombic → tilt of Pbl octohedras and increase the BG!

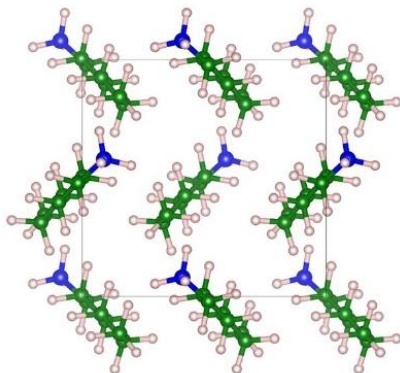
# Influence from the organic component

Top view

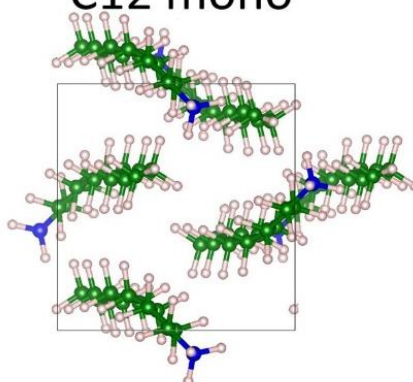
C6 mono



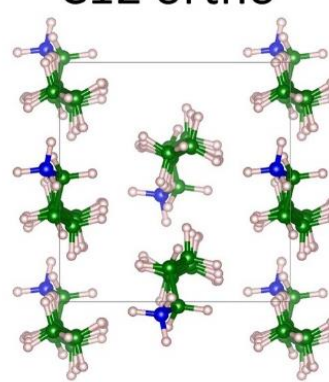
C6 ortho



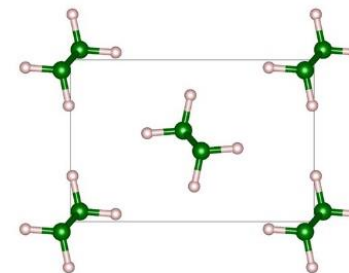
C12 mono



C12 ortho

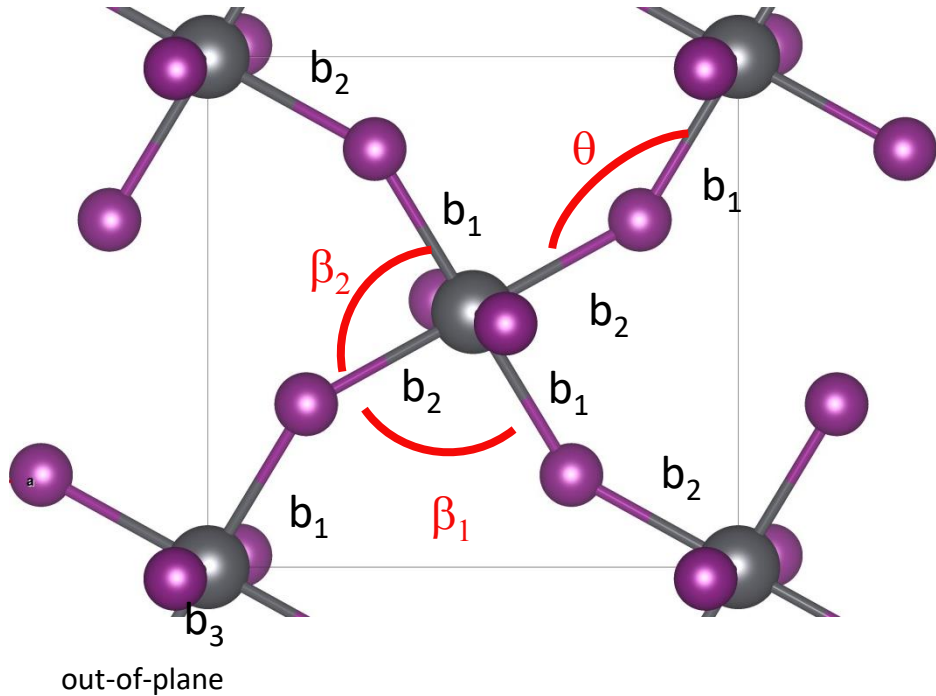


polyethylene



→ Alkyl chains stacked like PE in C12 orthorhombic → tilt of Pbl octohedras and increase the BG!

# Structural effect



→ Influence of organic component

↳ Alkyl chains stacked like PE

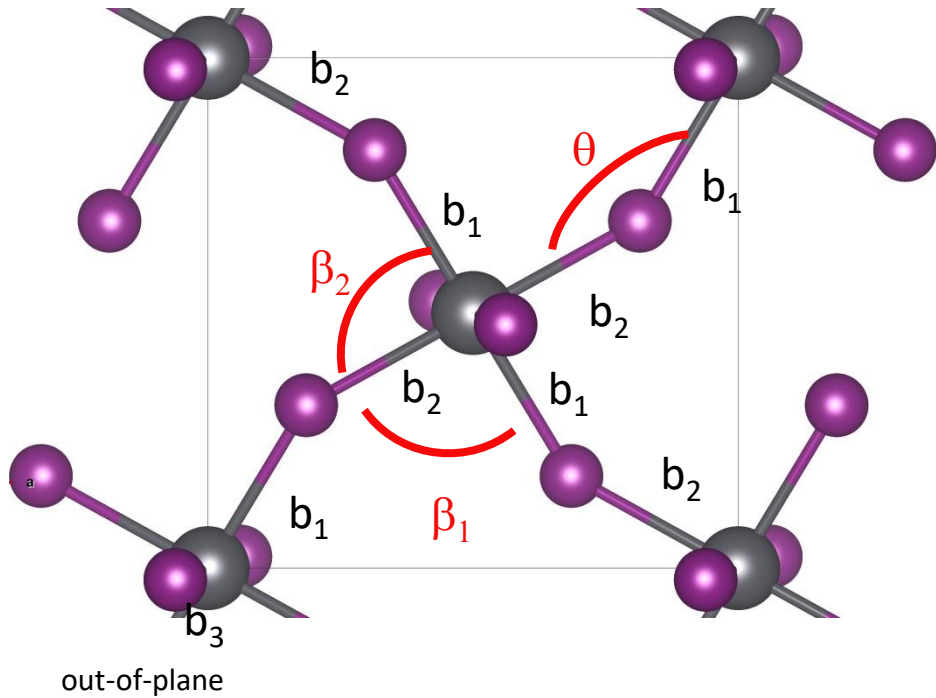
↳ Tilt of Pb-I octahedra: BG ↗

→ Consistent with the literature

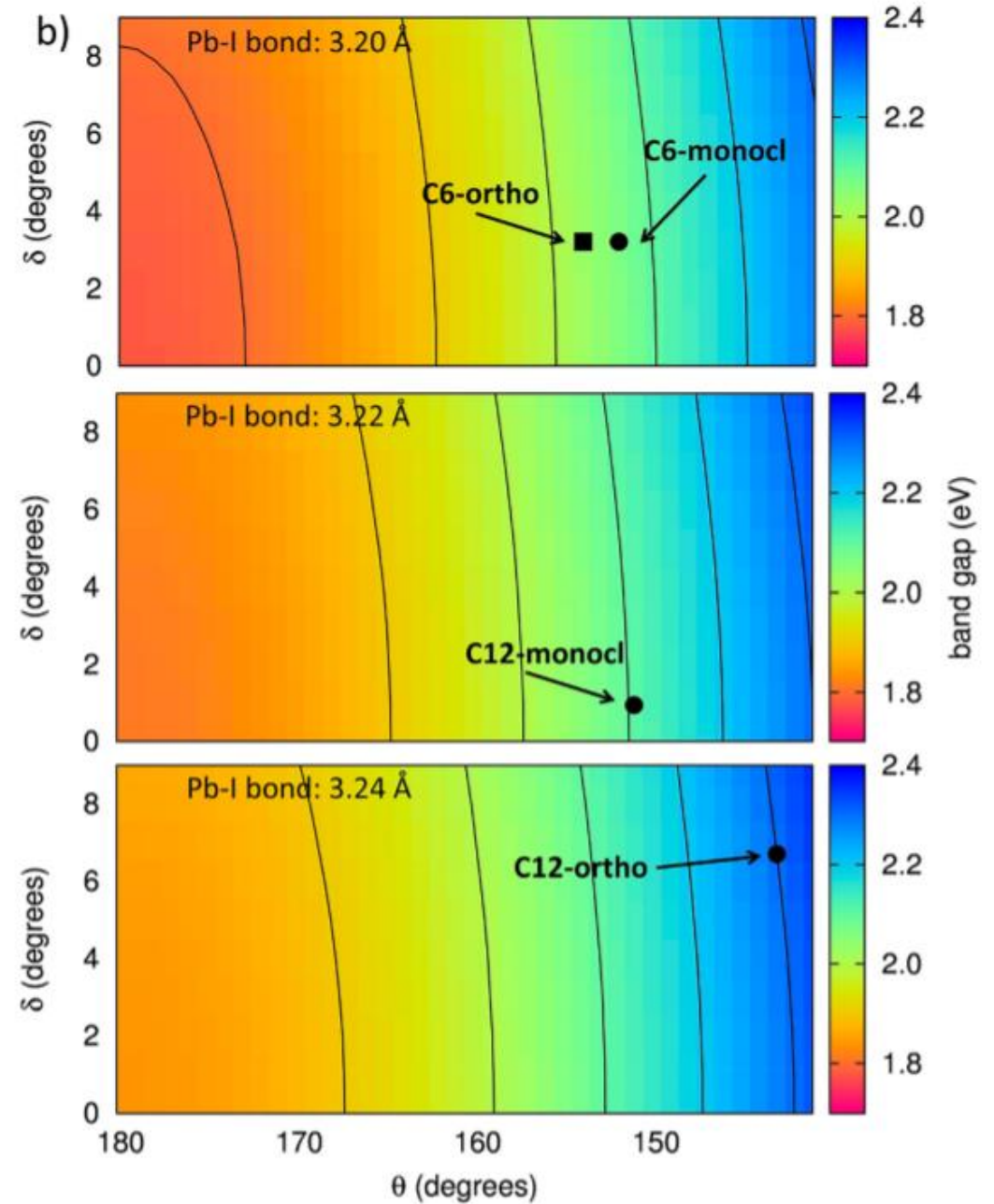
*Amat, Nano Lett.*, 14, p. 3608–3616 (2014)

		BG	$b_1$	$b_2$	$b_3$	$\theta$	$\beta_1$	$\beta_2$
C6	mono	2.06	3.19	3.18	3.26	152	92	88
	ortho	2.04	3.20	3.20	3.25	154	89	91
C12	mono	2.05	3.22	3.21	3.22	153	88	92
	ortho	<b>2.38</b>	3.23	3.24	3.24	<b>143</b>	<b>94</b>	<b>86</b>

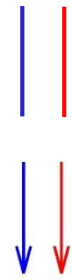
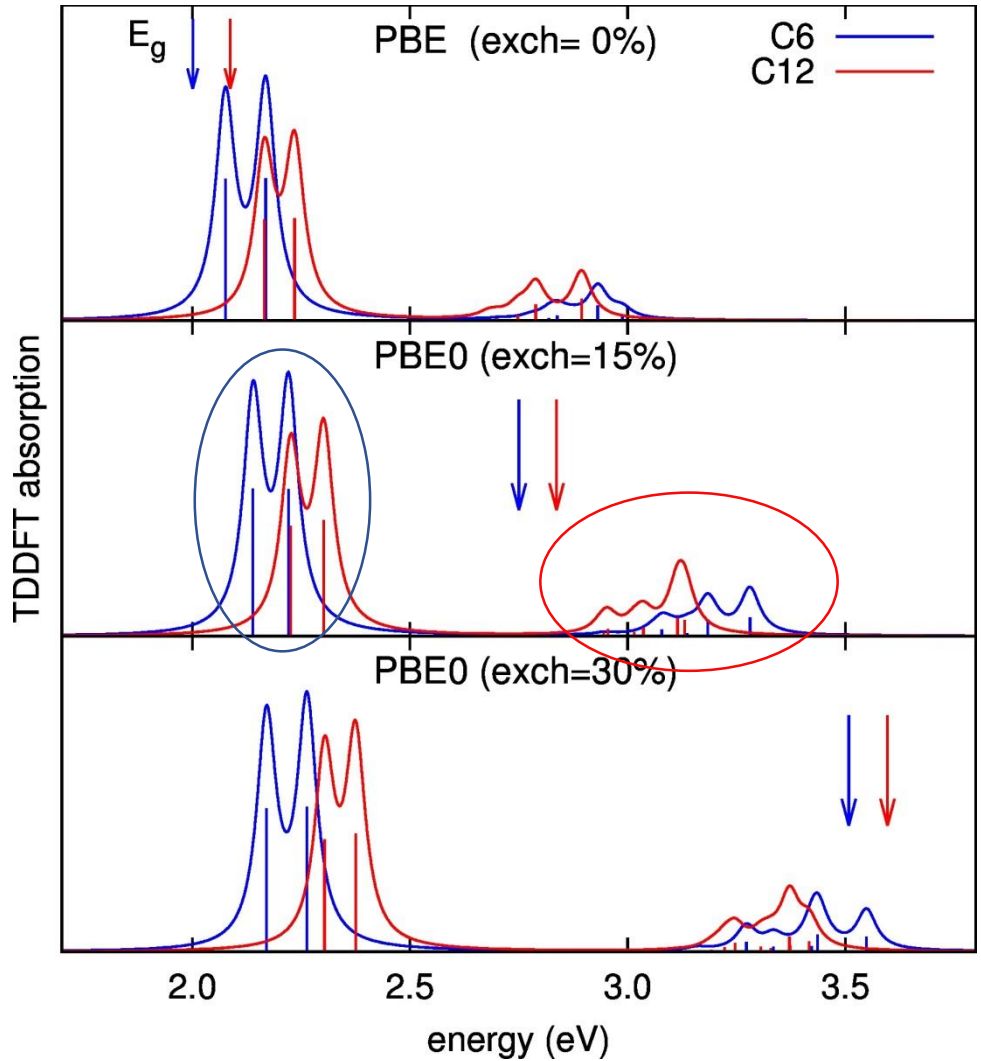
# Structural effect



		BG	$b_1$	$b_2$	$b_3$	$\theta$	$\beta_1$	$\beta_2$
C6	mono	2.06	3.19	3.18	3.26	152	92	88
	ortho	2.04	3.20	3.20	3.25	154	89	91
C12	mono	2.05	3.22	3.21	3.22	153	88	92
	ortho	<b>2.38</b>	3.23	3.24	3.24	<b>143</b>	<b>94</b>	<b>86</b>



# Optical properties

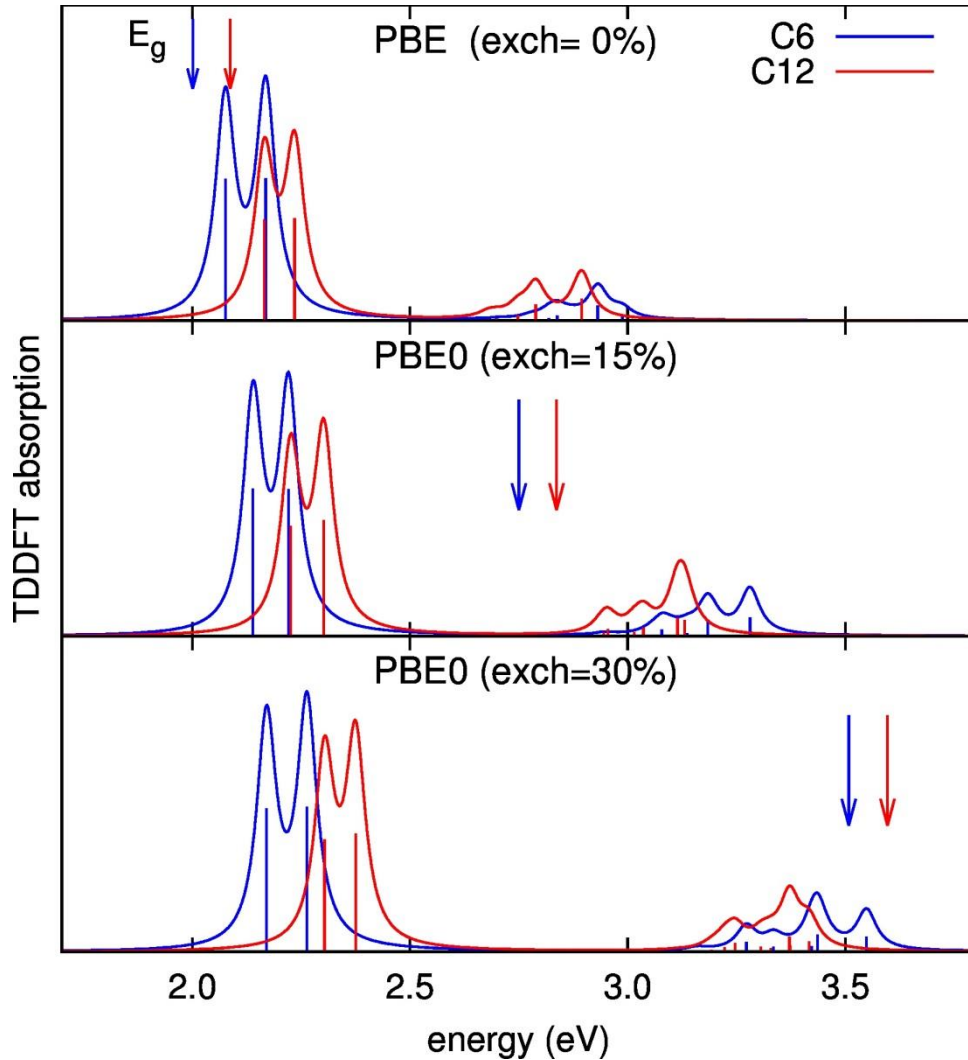


Energy of the excited states

$E_g$  of the 2 systems at different level of theory



# Optical properties

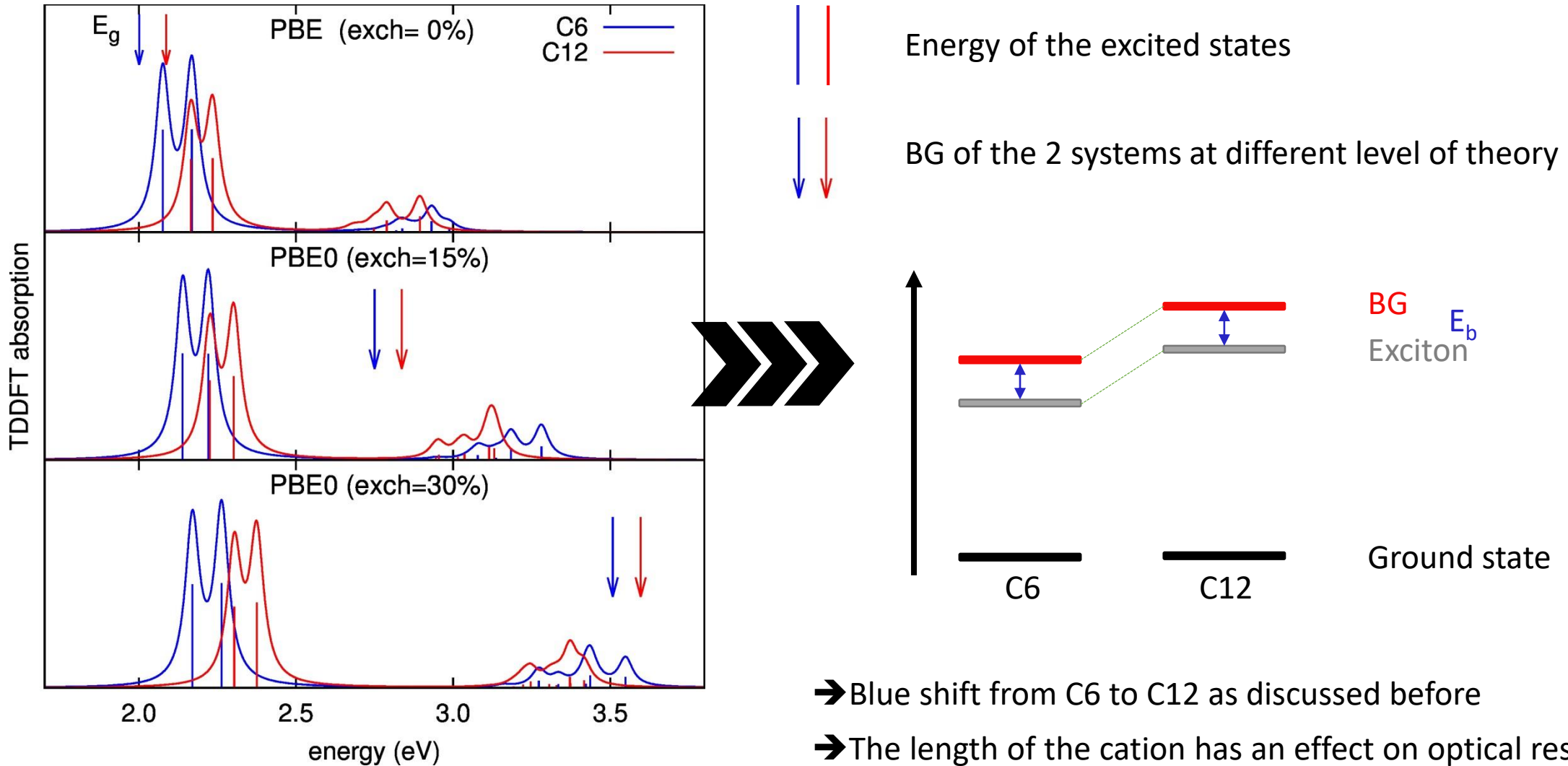


| | Energy of the excited states  
↓ ↓ BG of the 2 systems at different level of theory

	Experimental data		Calculations (HF 15%)	
	Excitonic transition (eV)	Exciton binding energy (eV)	Excitonic transition (eV)	Exciton binding energy (eV)
C6	2.37	0.36	2.14-2.22	0.6
C12	2.57	/	2.18-2.31	/

→ Consistant with experimental data

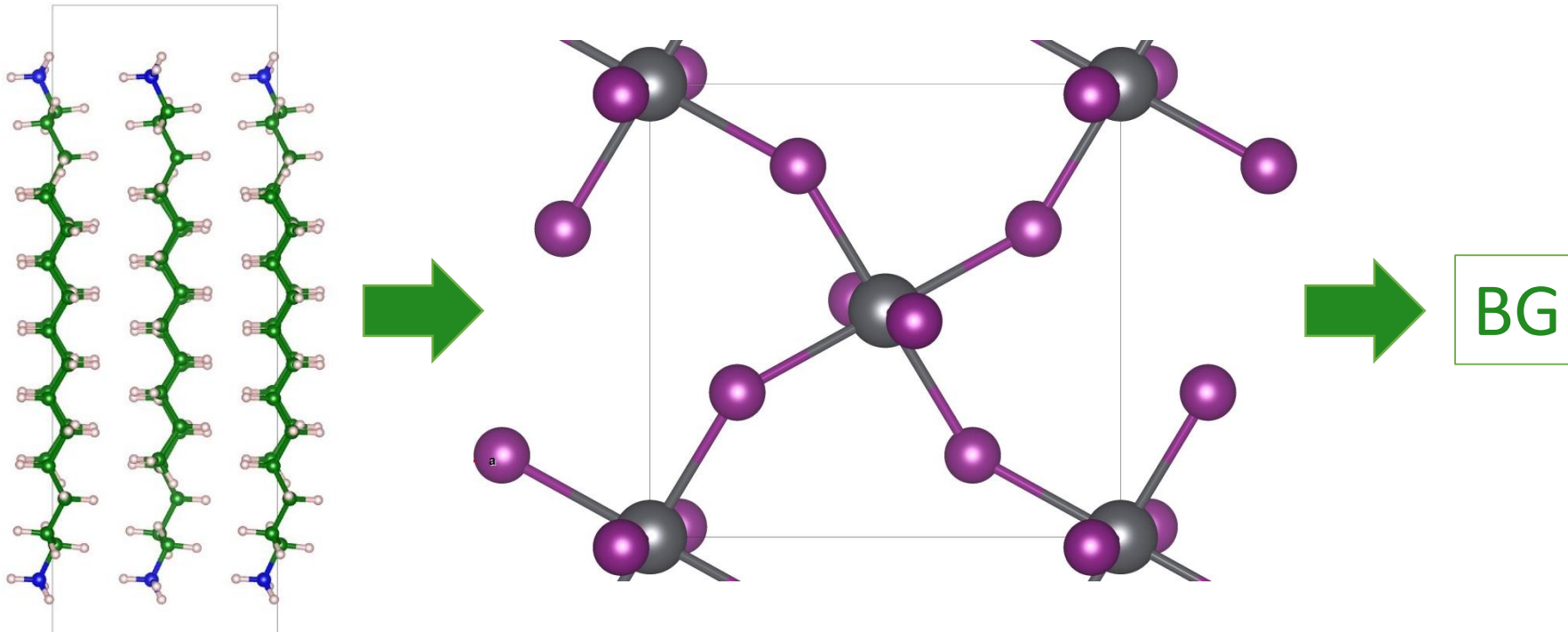
# Optical properties



- Blue shift from C6 to C12 as discussed before
- The length of the cation has an effect on optical response via the BG, no effect on exciton binding energy  $E_b$

# Summary

- Methodology → PBE0 30% of HF exchange + SOC lead to a good description of the electronic properties  
→ PBE0 15% HF seems to be better for optical properties (without SOC)
- Even if **the organic part is not directly involved in the description of the frontier crystalline orbitals**, it has an **indirect effect** on the electronic structure, namely the value of the bandgap! Same effect on the optical gap, no significant change on the exciton binding energy



# Acknowledgement

- Laboratory for Chemistry of Novel Materials
- UMONS Research Institute for Materials Science and Engineering
- nanoGe Fall Meeting



THANK YOU FOR YOUR KIND ATTENTION

# Phase transition

Orthorhombic  $\rightarrow$  Monoclinic

In C12PbI perovskites with should be accompanied by a sudden change in the band gap!!!

**THIS IS ACTUALLY THE CASE FOR C10PbI**

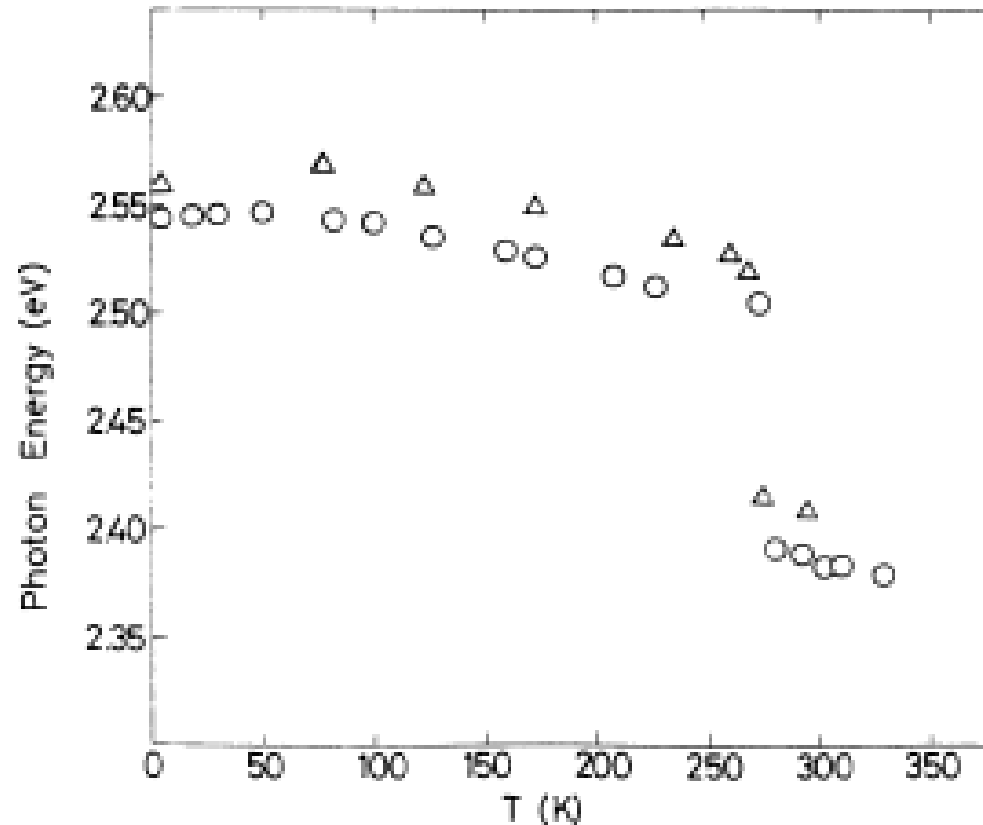


FIG. 7. Energies of absorption ( $\Delta$ ) and luminescence peaks ( $\circ$ ) in a single crystal of  $(C_{10}H_{21}NH_3)_2PbI_4$  as a function of temperature.

# Electronic properties

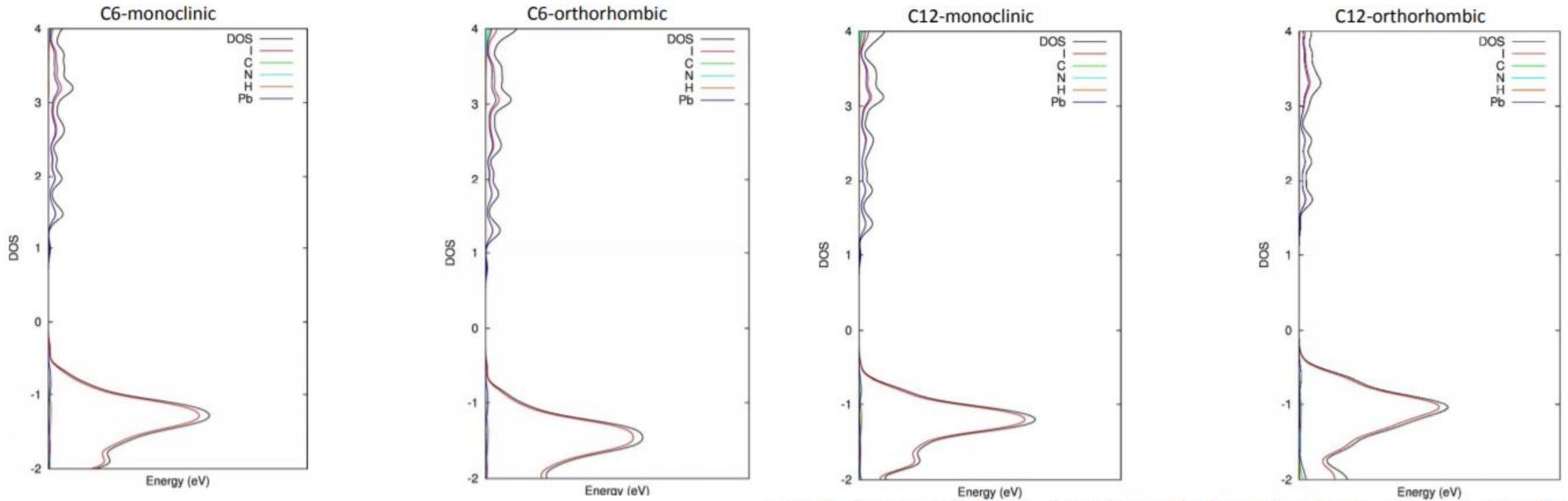
## Energy of the valence and conduction band edge for the C6 and C12 polymorphs

**Table S1.** Energy of the valence band edge (VBE) and conduction band edge (CBE) and band gap (Eg), computed for the monoclinic and orthorhombic polymorphs of C6 and C12, using the various computational approaches. All energies are referred to the averaged electrostatic potential of the crystal cell, computed using the various methods. Data in eV.

method	VBE	CBE	Eg	VBE	CBE	Eg
	C6-monoclinic			C6-orthorhomib		
PBE	-0.34	1.70	2.04	-0.47	1.55	2.02
PBE+SOC	-0.39	0.99	1.38	-0.55	0.80	1.35
PBE0	-1.23	2.28	3.51	-1.60	1.82	3.42
PBE CORRECT	-1.28	1.57	2.85	-1.68	1.07	2.75
PBE0+SOC	-1.41	1.32	2.73	-1.41	1.32	2.73
	C12-monoclinic			C12-orthorhombic		
PBE	-0.45	1.72	2.17	-0.23	2.12	2.44
PBE+SOC	-0.50	1.00	1.50	-0.37	1.34	1.71
PBE0	-1.57	2.03	3.60	-1.46	2.43	3.89
PBE CORRECT	-1.68	1.07	2.93	-1.46	1.65	3.11
PBE0+SOC	-1.45	1.42	2.87			

# Electronic properties

## Density of States (DOS) of the C6 and C12 polymorphs



**Figure S2.** Atomic Density of State of the C6PbI and C12PbI polymorphs. The contribution from the different chemical elements is listed. Electronic structure obtained with the PBE functional for the description of the exchange-correlation interaction, including spin-orbit coupling.

# Variable cell

## Comparison of the results for fixed cell and variable cell calculation

**Table S3.** Comparison of the band gap  $a$  ( $E_g$ , PBE), and structural parameters (bond lengths ,  $b_1$ ,  $b_2$ ,  $b_3$ , and Pb-I-PB angle  $\theta$ ) for fixed cell and variable cell relaxations.

system		$E_g$ (eV)	$b_1$ (Å)	$b_2$ (Å)	$b_3$ (Å)	$\theta$ (°)
<b>fixed cell</b>						
C6	mono	2.04	3.18	3.19	3.26	152.3
	ortho	2.02	3.20	3.19	3.25	153.8
C12	mono	2.17	3.22	3.21	3.23	153.4
	ortho	2.44	3.24	3.23	3.23	143.4
<b>variable cell</b>						
C6	mono	1.98	3.09	3.09	3.23	148.8
	ortho	2.04	3.10	3.10	3.23	146.5
C12	mono	2.00	3.07	3.08	3.23	146.3
	ortho	2.33	3.07	3.09	3.22	138.9



# Tuning the Optoelectronic Properties of Two-Dimensional Hybrid Perovskite Semiconductors with Alkyl Chain Spacers

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**S** Supporting Information

**ABSTRACT:** Layered two-dimensional organo-metal halide perovskites are currently in the limelight, largely because their versatile chemical composition offers the promise of tunable photophysical properties. We report here on (time-dependent) density functional theory [(TD)DFT] calculations of alkyl-ammonium lead iodide perovskites, where significant changes in the electronic structure and optical properties are predicted when using long versus short alkyl chain spacers. The mismatch between the structural organization in the inorganic and organic layers is epitomized for dodecyl chains that adopt a supramolecular packing similar to that of polyethylene, at the cost of distorting the inorganic frame and, in turn, opening the electronic band gap. These results rationalize recent experimental data and demonstrate that the optoelectronic properties of layered halide perovskite semiconductors can be modified through the use of electronically inert organic saturated chains.

