

# Doping of Monolayer Transition-Metal Dichalcogenides via Physisorption of Aromatic Solvent Molecules

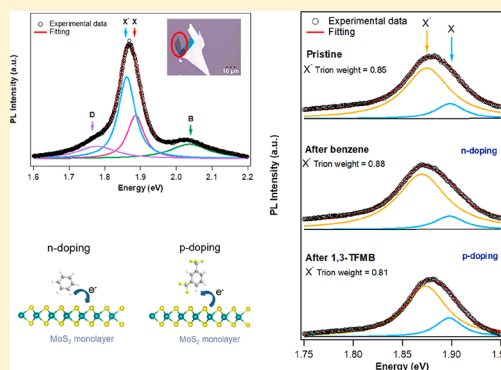
Ye Wang,<sup>†</sup> Amine Slassi,<sup>‡</sup> Marc-Antoine Stoeckel,<sup>†</sup> Simone Bertolazzi,<sup>†</sup> Jérôme Cornil,<sup>†,‡</sup> David Beljonne,<sup>\*,‡</sup> and Paolo Samorì<sup>\*,†,‡</sup>

<sup>†</sup>University of Strasbourg, CNRS, ISIS UMR 7006, 8 allée Gaspard Monge, F-67000 Strasbourg, France

<sup>‡</sup>Laboratory for Chemistry of Novel Materials, Université de Mons, Place du Parc 20, 7000 Mons, Belgium

## Supporting Information

**ABSTRACT:** Two-dimensional (2D) transition-metal dichalcogenides (TMDs) recently emerged as novel materials displaying a wide variety of physicochemical properties that render them unique scaffolds for high-performance (opto)electronics. The controlled physisorption of molecules on the TMD surface is a viable approach for tuning their optical and electronic properties. Solvents, made of small aromatic molecules, are frequently employed for the cleaning of the 2D materials or as a “dispersant” for their chemical functionalization with larger (macro)molecules, without considering their potential key effect in locally modifying the characteristics of 2D materials. In this work, we demonstrate how the electronic and optical properties of a mechanically exfoliated monolayer of MoS<sub>2</sub> and WSe<sub>2</sub> are modified when physically interacting with small aromatic molecules of common solvents. Low-temperature photoluminescence (PL) spectra recorded at 78 K revealed that physisorbed benzene derivatives could modulate the charge carrier density in monolayer TMDs, hence affecting the switching between a neutral exciton and trion (charged exciton). By combining experimental evidence with density functional theory calculations, we confirm that charge-transfer doping on TMDs depends not only on the difference in chemical potential between molecules and 2D materials but also on the thermodynamic stability of physisorption. Our results provide unambiguous evidences of the great potential of optical and electrical tuning of monolayer MoS<sub>2</sub> and WSe<sub>2</sub> by physisorption of small aromatic solvent molecules, which is highly relevant for both fundamental studies and device application purposes.



During the past decade, two-dimensional (2D) materials have attracted remarkable attention because of their unique physical and chemical properties. Among them, transition-metal dichalcogenides (TMDs) are semiconducting systems that exhibit an indirect-to-direct bandgap transition from bulk to monolayer, large exciton binding energies, and inversion symmetry breaking, which are attractive for a large range of applications from (opto)electronics to valleytronics.<sup>1–6</sup> The large surface-to-volume ratio in the crystal structure of monolayer TMDs makes them extremely sensitive to changes in the environment.<sup>7</sup> In this regard, the simple physisorption of atoms and molecules represents a powerful method to modulate their optical and electrical properties.<sup>7,8</sup>

Highly polar physisorbed molecules, including TCNQ and NADH,<sup>9</sup> hydrazine,<sup>10</sup> and benzene viologen,<sup>11</sup> have been utilized for chemical doping of monolayer TMDs. Interactions between these molecules and TMDs induce charge transfer, thus modifying the Fermi level and tuning the electronic and optical properties of the material. In monolayer TMDs, strong electron–hole Coulomb interactions enable the formation of stable optically generated excitons at room temperature;<sup>12,13</sup> these excitons are building blocks for the generation of many-body bound states such as electron-bound exciton (negative

trion) or hole-bound exciton (positive trion).<sup>14,15</sup> Accordingly, electron and hole concentrations modified by physisorbed molecules effectively influence the formation of excitons and trions in TMDs.

Although a great effort has been devoted to the noncovalent functionalization of monolayer TMDs via molecular physisorption to improve fundamental properties<sup>16</sup> such as photoluminescence tuning,<sup>9,10,17,18</sup> electron/hole doping,<sup>19,20</sup> and device contact improvement,<sup>21</sup> the simple interaction of a functionalized benzene ring (being frequently a fragment of solvent molecules) and monolayer TMDs has not yet been fully unraveled.<sup>22</sup> Despite the fact that typical solvent molecules are seemingly considered as inert media, aromatic molecules have been proved to cause n- or p-doping with distinct functional groups through  $\pi$ – $\pi$  interactions with graphene.<sup>23</sup> Unfortunately, similar studies on TMDs have not yet been reported. To gain a comprehensive understanding of the effect of physisorbed molecules on monolayer TMDs, we focused our attention here on benzene and its derivatives

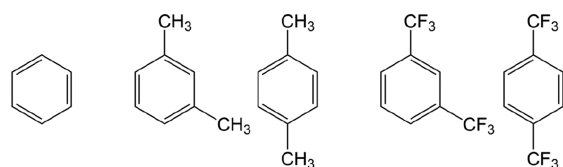
Received: December 11, 2018

Accepted: January 16, 2019

Published: January 16, 2019

depicted in Scheme 1 [i.e., benzene, *m*-xylene, *p*-xylene, 1,3-bis(trifluoro)methylbenzene (1,3-TFMB), and 1,4-bis-

**Scheme 1. Chemical Formula of Benzene, *m*-Xylene, *p*-Xylene, 1,3-Bis(trifluoro)methylbenzene (1,3-TFMB), and 1,4-Bis(trifluoromethyl)benzene (1,4-TFMB)**

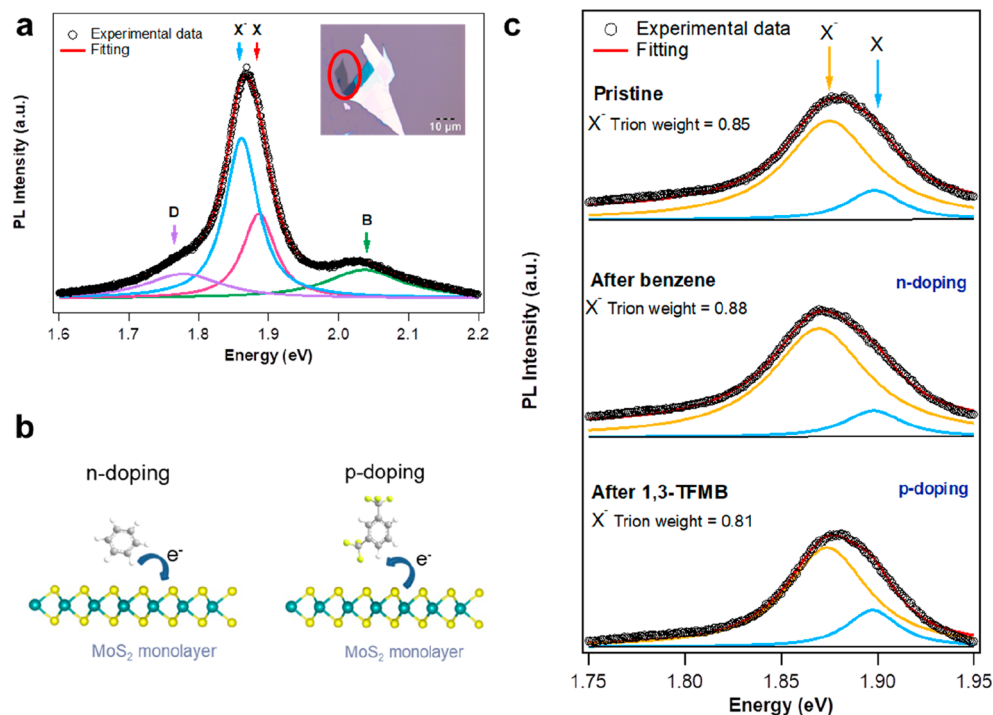


(trifluoromethyl)benzene (1,4-TFMB)] as aromatic molecules and MoS<sub>2</sub> and WSe<sub>2</sub> as representative TMD monolayers acting as platforms for physisorption. The functionality of aromatic molecules is therefore modified through simple substitution in the *meta* and *para* position with either more apolar methyl (–CH<sub>3</sub>) or more polar trifluoromethyl (–CF<sub>3</sub>) groups. The melting point, boiling point, and total dipole moment of each molecule are listed in Table S1. Importantly, benzene, *m*-xylene, and *p*-xylene are commonly used as solvents to dissolve complex organic molecules for functionalization of 2D materials and for device fabrication. These aromatic solvents could be unintentionally physisorbed on 2D materials resulting in changes in physical and chemical properties. Thus, understanding the effect of these molecules on monolayer TMDs is a crucial issue to be addressed not only for revealing molecule–TMD interactions but also for device optimization. We reveal here that the occurrence of charge transfer between physisorbed molecules and monolayer TMDs leads to change in electron/hole density and trion/exciton intensity ratio. Through low-temperature photoluminescence (PL) measure-

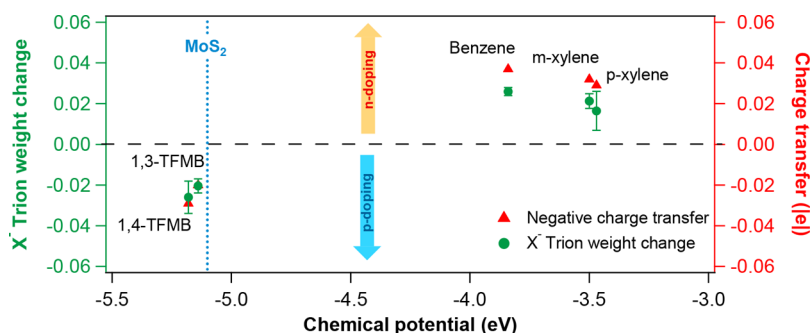
ments performed at 78 K corroborated with density functional theory (DFT) calculations, we quantify the n or p nature of doping for each molecule deposited on monolayer TMDs. Our results provide a textbook proof-of-concept of the use of physisorbed aromatic molecules on single-layer TMDs to dope the 2D material in a controlled fashion.

The photoluminescence behavior of monolayer MoS<sub>2</sub> is known to be strongly temperature-dependent.<sup>24–26</sup> A typical PL spectrum of monolayer MoS<sub>2</sub> at 78 K recorded in N<sub>2</sub> atmosphere is shown in Figure 1a. It exhibits four components in the range of 1.6–2.2 eV: neutral A exciton (X) at ~1.88 eV and negatively charged A trion (X<sup>−</sup>) at ~1.90 eV, B exciton (B) at ~2.05 eV and a defect-induced emission (D) at ~1.75 eV. The A excitons (X and X<sup>−</sup>) are blue-shifted of 23 meV compared to room temperature because of the bandgap enlargement upon decreasing the temperature.<sup>24</sup> Temperature dependent studies on the photoluminescence of monolayer MoS<sub>2</sub> (Figure S1 in the Supporting Information) reveals that localized emission caused by excitons bounded with defects (D) appears below 100 K. It is fair to indicate that such a D peak is not observed in every flake of our experiment likely because of the difference in defect density generated by mechanical exfoliation.

The physisorption of benzene and its derivatives [i.e., benzene, *m*-xylene, *p*-xylene, 1,3-bis(trifluoro)methylbenzene (1,3-TFMB), and 1,4-bis(trifluoromethyl)benzene (1,4-TFMB)] as aromatic molecules onto monolayer MoS<sub>2</sub> (Figure 1b) is explored by monitoring the photoluminescence spectra. Such study provides evidence for the occurrence of chemical doping as revealed by major changes in PL emission intensity and shape.<sup>9</sup> Representative PL spectra of monolayer MoS<sub>2</sub> tuned by aromatic molecules at 78 K are portrayed in Figure 1c, whereas the spectra, and related fitting, for MoS<sub>2</sub> with all



**Figure 1.** (a) Photoluminescence spectra of monolayer MoS<sub>2</sub> at 78 K (inset: optical image of as-exfoliated monolayer MoS<sub>2</sub> highlighted with a red circle). (b) Molecular representation of charge transfer between physisorbed aromatic molecules and monolayer MoS<sub>2</sub>. (c) Photoluminescence spectra of monolayer MoS<sub>2</sub> before and after physisorption of benzene and 1,3-TFMB with calculated X-trion weight.



**Figure 2.**  $X^-$  trion weight change and calculated Bader charge transfer (red) by physisorption of aromatic molecules from low-temperature PL measurement of  $\text{MoS}_2$  (green) as a function of chemical potential. Fluorinated molecules (1,3-TFMB and 1,4-TFMB) possessing lower chemical potential are easier to accept electrons from monolayer  $\text{MoS}_2$ ; in contrast, nonfluorinated molecules (benzene, *p*-xylene and *m*-xylene) with higher chemical potential donate electrons to monolayer  $\text{MoS}_2$ .

kinds of physisorbed molecules are displayed in Figure S3. We focus our attention on the A peak that tracks the population of the trion ( $X^-$ ) and neutral exciton (X). Given that at 78 K all aromatic molecules investigated are in their solid phase, the chemical doping process might differ from those reported in previous studies carried out at room temperature.<sup>9,10,17,18</sup> Our data reveal that upon treatment with benzene, the spectral weight of trion increases whereas the use of 1,3-TFMB induces a decrease in the trion weight. Such effects are caused by changes in charge carrier density in monolayer  $\text{MoS}_2$  induced by molecular doping. When monolayer  $\text{MoS}_2$  is n-doped, the increase in electron density promotes the formation of the negatively charged trion. Conversely, p-doping enables the recombination of neutral excitons into the positively charged trion. The trion weight can be quantified as

$$\gamma^- = \frac{I_{X^-}}{I_{\text{total}}} = \frac{I_{X^-}}{I_{X^-} + I_X} \quad (1)$$

where  $\gamma^-$  is the negative trion weight of monolayer  $\text{MoS}_2$ ,  $I_{X^-}$  the area of negative trion peak,  $I_X$  the area of neutral exciton peak, and  $I_{\text{total}}$  the area of total photoluminescence intensity. Therefore, we calculate the trion weight change ( $\Delta\gamma^-$ ).  $\Delta\gamma^- > 0$  indicates an increase in electron density upon molecular physisorption, implying an n-doping effect. Conversely,  $\Delta\gamma^- < 0$  denotes a decrease in electron density upon molecular physisorption, corresponding to p-doping.

To explore the origin of the trion weight change ( $\Delta\gamma^-$ ), we have discussed different mechanisms which lead to the exclusion of dipolar effects and the activation of defect-induced photoluminescence, as shown in the Supporting Information. This discussion made it possible to demonstrate that the observed induced doping in the TMDs is due to charge transfer between molecules and 2D materials. The trion weight change ( $\Delta\gamma^-$ ) quantifies the ability of doping, in this specific case of  $\text{MoS}_2$  as a result of the physisorption of the molecular monolayer, as determined by PL measurements. In the simplest approximation, when two species A and B (here the solvent molecules and the  $\text{MoS}_2$  solid) with chemical potentials  $\mu_A \neq \mu_B$  are brought into close contact, electron density  $N$  flows from the high-potential to the low-potential system until equilibration is reached:<sup>27</sup>

$$\Delta N = \frac{\mu_A - \mu_B}{\eta_A + \eta_B} \quad (2)$$

where  $\eta_A$  and  $\eta_B$  are the chemical hardness of A and B, respectively. To understand the difference in values of  $\Delta\gamma^-$

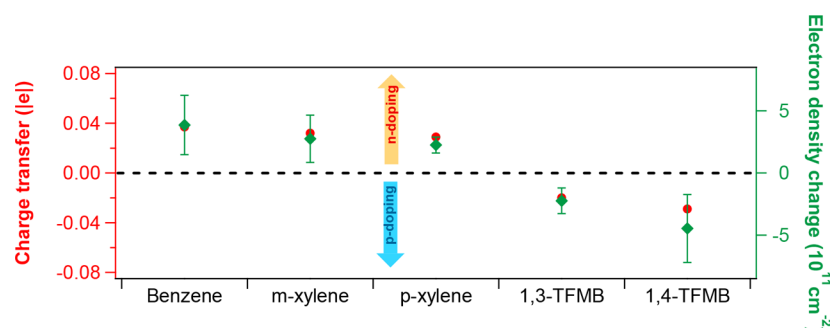
resulting from the physisorption of different molecules, we have calculated the chemical potential and hardness values of monolayer  $\text{MoS}_2$  and the investigated aromatic molecules.<sup>28</sup> In the frontier molecular orbital framework and using a finite difference approximation, these become

$$\mu = \frac{1}{2}(\epsilon_{\text{hole}} + \epsilon_{\text{electron}}); \quad \eta = \frac{1}{2}(\epsilon_{\text{electron}} - \epsilon_{\text{hole}}) \quad (3)$$

where  $\epsilon_{\text{hole}}$  ( $\epsilon_{\text{electron}}$ ) denotes the energy of the molecular HOMO (LUMO) level for the solvent and the energy of the top of the valence (conduction) band for the 2D solid.

As shown in Figure 2, fluorinated molecules have a smaller chemical potential due to the electron-withdrawing character of  $-\text{CF}_3$  groups. The chemical potential of monolayer  $\text{MoS}_2$  is calculated to be  $-5.1$  eV, which is closely above that of  $-5.15$  and  $-5.19$  eV for 1,3-TFMB and 1,4-TFMB, respectively. Therefore, electrons tend to transfer from  $\text{MoS}_2$  to 1,3-TFMB and 1,4-TFMB, so that electron density is decreased in monolayer  $\text{MoS}_2$  resulting in the p-doping effect. In contrast, benzene ( $-3.84$  eV), *m*-xylene ( $-3.50$  eV), and *p*-xylene ( $-3.47$  eV) have higher chemical potentials than monolayer  $\text{MoS}_2$ ; hence, it is easier to transfer electrons from the former to the latter, leading to n-doping to the 2D layer. While the chemical hardness weighted difference in chemical potential between aromatic molecules explains the type of doping of  $\text{MoS}_2$ , the quantitative agreement is limited. In other words, the sign of the calculated  $\Delta N$  in eq 2 is consistent with the measured  $\Delta\gamma^-$ , but the trend with the nature of the solvent is only qualitative (see Table S2). For example, benzene has smaller chemical potential than xylenes, but it gives larger trion weight change; 1,3-TFMB and 1,4-TFMB have much smaller chemical potential difference than other molecules, but  $\Delta\gamma^-$  values are comparable to benzene and xylenes. This is likely because of the thermodynamic stability driving to certain configurations of the self-assembled molecules on the surface of  $\text{MoS}_2$ , possibly sourcing a surface electrostatic potential, together with other effects not included in the simple model (hybridization with and electronic polarization in the solid). To better appraise these effects, we have assessed the charge transfer taking place from the solvents to the  $\text{MoS}_2$  ( $\text{WSe}_2$ ) sheet upon adsorption by applying the Bader charge analysis to the equilibrated interfaces.<sup>29</sup> The calculated charge transfer matches the trend of  $\Delta\gamma^-$  much better than the chemical potential-based model (Figure 2).

The correlation of  $\Delta\gamma^-$  and charge-transfer value can be estimated by a mass action model:<sup>9</sup>



**Figure 3.** Electron density change (green) after physisorption on MoS<sub>2</sub> of each aromatic molecule, as calculated by mass action model and provided by a DFT/Bader charge analysis (red).

$$\frac{N_X n_{el}}{N_{X^-}} = \left( \frac{4m_X m_e}{\pi \hbar^2 m_{X^-}} \right) k_B T \exp\left( -\frac{E_b}{k_B T} \right) \quad (4)$$

where  $N_X$  and  $N_{X^-}$  are the populations of excitons (X) and trions (X<sup>-</sup>).  $n_{el}$  is the electron density, and  $E_b$  is the binding energy of the trion ( $\sim 20$  meV).  $T$  is the temperature (78 K).  $m_X$ ,  $m_{X^-}$ , and  $m_e$  are effective masses of the exciton, trion, and electron, respectively. Considering  $m_e \approx 0.35m_0$  and  $m_h \approx 0.45m_0$ ,  $m_X$  and  $m_{X^-}$  can be calculated as  $m_X = m_e + m_h = 0.8m_0$  and  $m_{X^-} = 2m_e + m_h = 1.15m_0$ . Hence, the trion weight can be expressed as

$$\frac{I_{X^-}}{I_{total}} = \frac{\frac{\gamma_{tr} N_{X^-}}{\gamma_{ex} N_X}}{1 + \frac{\gamma_{tr} N_{X^-}}{\gamma_{ex} N_X}} \approx \frac{1.5 \times 10^{-15} n_{el}}{1 + 1.5 \times 10^{-15} n_{el}} \quad (5)$$

where  $\gamma_{tr}$  and  $\gamma_{ex}$  are radiative decay rates of the trion and exciton, respectively. Figure 3 demonstrates a good correspondence between the electron density change obtained from experimental data and the charge transfer predicted by DFT calculations, as the electron density change of monolayer MoS<sub>2</sub> after the physisorption of aromatic molecules is linearly proportional to the charge-transfer value per molecule. By carefully excluding the doping effect from the air atmosphere, our measurements show that the modulation of charge density by solvent trace in monolayer MoS<sub>2</sub> is largely decreased to  $10^{11}/\text{cm}^{-2}$  compared to previous studies that attain  $10^{13}/\text{cm}^{-2}$ .<sup>30</sup> Therefore, it is possible to estimate the number of physisorbed molecules on the surface of monolayer MoS<sub>2</sub> by considering

$$\Delta n_{el} \cdot |e| = n_{mol} \cdot \Delta \sigma^- \quad (6)$$

where  $\Delta n_{el}$  is the change in electron density,  $|e|$  the absolute value of elementary charge,  $n_{mol}$  the number of molecules adsorbed per unit area ( $\text{cm}^{-2}$ ), and  $\Delta \sigma^-$  the negative charge-transfer value. The estimated number of adsorbed molecules for each chemical is listed in Table 1. It is clear that xylenes have the smaller number of adsorbed molecules, whereas fluorinated molecules tend to self-assemble more densely on

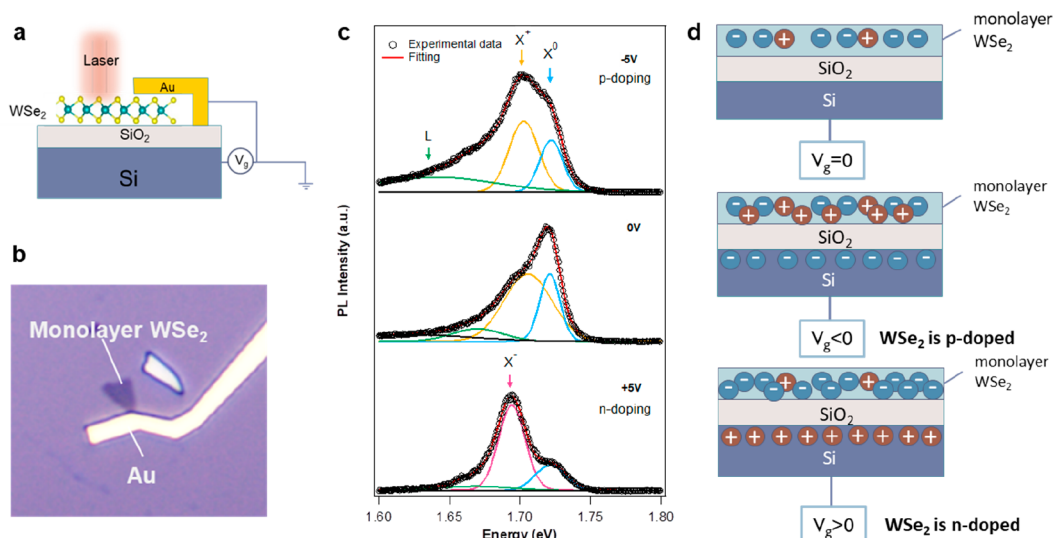
**Table 1.** Estimated Number of Physisorbed Aromatic Molecules on Monolayer MoS<sub>2</sub> Calculated from Electron Density Change and Transferred Charge

	benzene	m-xylene	p-xylene	1,3-TFMB	1,4-TFMB
number of physisorbed molecules ( $\times 10^{13}/\text{cm}^2$ )	1.04	0.86	0.77	1.12	1.54

the surface of the MoS<sub>2</sub>. This might be attributed to the preference of Coulomb interaction between the electron-withdrawing  $-\text{CF}_3$  group and the intrinsically n-doped monolayer MoS<sub>2</sub>. We have also studied at a theoretical level how charge transfer from the physisorbed molecules to the TMD monolayer is influenced by varying the density of molecules, interlayer distance, and inclination angle between molecules and monolayer MoS<sub>2</sub>, as discussed in the Supporting Information.

In contrast to monolayer MoS<sub>2</sub>, the photoluminescence behavior of monolayer WSe<sub>2</sub> exhibits multiple possibilities for trion recombination. At low temperature, the coexistence of neutral excitons and positively charged excitons or negatively charged excitons and biexcitons is observed by changing temperature (4, 10, 30, and 60 K) and device structure.<sup>31–33</sup> To investigate the influence of aromatic molecules deposited on monolayer WSe<sub>2</sub>, here we first study its excitonic characteristics at 78 K on the SiO<sub>2</sub> substrate. The temperature dependence of monolayer WSe<sub>2</sub> is similar to that of MoS<sub>2</sub> whereby, upon decreasing the temperature, the exciton peak is blue-shifted. Defect-induced emissions appear below 200 K as multiple peaks in the range of 1.60–1.65 eV in which the intensity is dependent on the quality of the flakes. Moreover, at room temperature, a single excitonic peak at 1.66 eV is observed. Below 100 K, this peak is split into two independent components at  $\sim 1.72$  and  $\sim 1.70$  eV (Figure S2).

To assign the origin of these two excitonic components, field-effect devices were fabricated by E-beam lithography to induce electrical tunable doping. Panels a and b of Figure 4 portray the schematic and optical image of the device, respectively. As displayed in the scheme in Figure 4d, by adding a gate voltage on the field-effect transistor (FET) device, it is possible to modulate charge carrier density in WSe<sub>2</sub> monolayers and therefore to tune the proportion of excitons and trions. Upon applying negative gate voltage, electrons in monolayer WSe<sub>2</sub> are attracted to the ground via the gold electrode; the electron density in the material is therefore largely decreased, and neutral excitons tend to capture excess holes to form positive trion. In contrast, when a positive gate voltage is applied, more electrons are attracted to the channel; therefore, negative trion recombination is facilitated. Figure 4c shows the PL spectra of monolayer WSe<sub>2</sub> at 78 K under  $-5$  V,  $0$  V, and  $+5$  V gate voltage. At  $0$  V, apart from the peak at  $\sim 1.65$  eV, two other peaks, namely, trion at  $\sim 1.70$  eV and exciton at  $\sim 1.72$  eV, are observed. At  $-5$  V, the intensity of the trion peak increases. Considering the increase in hole density by increasing negative gate voltage, we assign this peak as a positive trion (two holes and one electron combined). This



**Figure 4.** Gate-induced photoluminescence of monolayer WSe<sub>2</sub>. (a) Schematic representation of the device for gate-induced photoluminescence in monolayer WSe<sub>2</sub> and (b) its corresponding optical image. (c) PL spectra of monolayer WSe<sub>2</sub> at 78 K under  $-5$ ,  $0$ , and  $+5$  V, demonstrating the recombination of neutral exciton ( $X^0$ ) and positive trion ( $X^+$ ) at  $0$  and  $-5$  V and the existence of neutral exciton ( $X^0$ ) and negative trion ( $X^-$ ) at  $+5$  V. (d) Modulation of charge carrier by gating. The change in carrier density results in electrostatic doping on monolayer WSe<sub>2</sub> and affects the recombination of positive or negative triions.

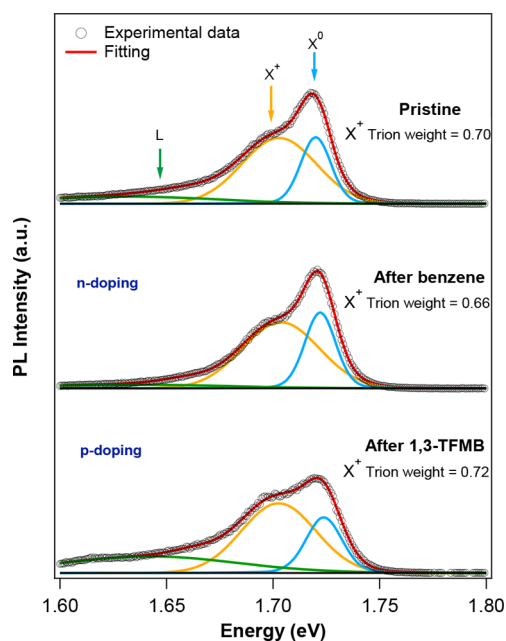
also implies that our material is intrinsically p-doped even after annealing, which is inconsistent with previous work in which either a protection layer of h-BN was added or the experiment was performed at lower temperature.<sup>32</sup> At  $+5$  V, when more electrons are injected into monolayer WSe<sub>2</sub>, the negative trion (two electrons and one hole combined) emission at  $\sim 1.69$  eV dominates. We extract the binding energy of the positive trion and negative trion at 78 K to be  $17\text{--}22$  meV and  $31\text{--}40$  meV, respectively, by calculating the difference of trion emission energy and neutral exciton energy. We have also observed quantum-confined Stark effect by applying higher gate voltage and a substrate-induced hysteresis in PL mapping under small gating steps; however, this is beyond the scope of this work.

After studying the peak position of each type of exciton emission in our experimental conditions, we focused our attention on the effect of chemical doping on monolayer WSe<sub>2</sub>. With no gate voltage applied to the flake, we discuss only the neutral exciton and positive trion. We quantify the positive trion as

$$\gamma^+ = \frac{I_{X^+}}{I_{\text{total}}} = \frac{I_{X^+}}{I_X + I_{X^+}} \quad (7)$$

where  $\gamma^+$  is the positive trion weight of monolayer WSe<sub>2</sub>,  $I_{X^+}$  the area of positive trion peak,  $I_X$  the area of neutral exciton peak, and  $I_{\text{total}}$  the area of total photoluminescence intensity. Therefore, we calculate the trion weight change ( $\Delta\gamma^+$ ) to evaluate the charge-transfer doping of molecules on monolayer WSe<sub>2</sub>.

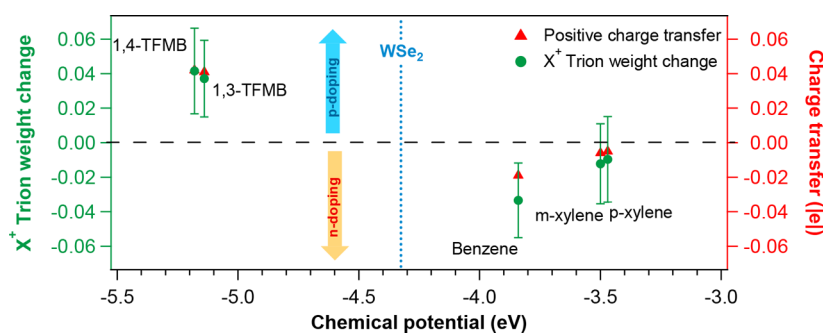
Figure 5 shows typical fitted PL spectra before and after doping with aromatic molecules at 78 K. The spectral weight change of the trion induced by benzene indicates n-doping and that 1,3-TFMB is a p-dopant. For other aromatic molecules, we find a similar type of doping as monolayer MoS<sub>2</sub> (detailed PL spectra are shown in Figure S4). To elucidate the consistency of doping, we have calculated the chemical potential of monolayer WSe<sub>2</sub> to be  $-4.33$  eV, which is nearly 1 eV above the fluorinated molecules but still 1 eV lower than the other molecules, suggesting that the charge-transfer



**Figure 5.** Photoluminescence spectra of monolayer WSe<sub>2</sub> before and after physisorption of benzene and 1,3-TFMB with calculated  $X^+$  trion weight.

direction between the molecules and monolayer WSe<sub>2</sub> should be the same as in MoS<sub>2</sub>. Both calculated charge transfers and trion weight changes reveal that benzene has the largest n-doping ability and that 1,4-TFMB p-dopes the monolayer WSe<sub>2</sub> the most. Both xylenes have trivial influence on carrier density change in monolayer WSe<sub>2</sub> (Figure 6).

In summary, our spectroscopic investigation provides unambiguous evidence of the full potential of the physisorption of small aromatic molecules on MoS<sub>2</sub> and WSe<sub>2</sub> to tune their optoelectronic properties via charge transfer. The PL study performed at 78 K demonstrated for the first time that tunable chemical doping can be achieved on both 2D materials



**Figure 6.**  $X^+$  trion weight change and calculated DFT/Bader charge transfer (red) by physisorption of aromatic molecules from low-temperature PL measurements of monolayer  $WSe_2$  (green) as a function of chemical potential. Fluorinated molecules (1,3-TFMB and 1,4-TFMB) possessing lower chemical potential are more prone to accept electrons from monolayer  $MoS_2$ ; in contrast, nonfluorinated molecules (benzene, *p*-xylene, and *m*-xylene) with higher chemical potential donate electrons to monolayer  $WSe_2$ .

through a subtle choice of simple aromatic molecules and their dosing on the surface. In particular, while fluorinated aromatics determined a p-doping, an n-doping was observed for the other methyl-substituted and non-substituted molecules. The calculated charge transfer of fluorinated solvent molecules on  $MoS_2$  could be comparable to traditional organic p-dopants tetracyanoquinodimethane (TCNQ) and tetracyanoethylene (TCNE), while typical n-dopants tetrathiafulvalene (TTF) and benzene viologen (BV) could donate electrons beyond the ability of aromatic solvents.<sup>17</sup> Charge carrier modulation by optical response is found to be 2 orders of magnitude lower in charge carrier density than previous studies at room temperature in air<sup>9,30</sup> by freezing the system at low temperature in inert atmosphere. The combined experimental analyses and DFT calculations utilized in this study represent a novel technique for estimating the density of physisorbed molecules in 2D surfaces. Furthermore, we have investigated for the first time the gate-tunable photoluminescence and defined binding energy of both positive and negative trions of monolayer  $WSe_2$  at 78 K through field-effect transistors.

Our results clearly indicate that care should be taken when choosing the solvent for the cleaning of the 2D materials or as “dispersant” for their chemical functionalization with larger (macro)molecules, because it can introduce strong electronic effects like doping. On the other hand, solvent molecules are clearly multifunctional systems because they both act as dispersant for the 2D materials and can enable the tuning of their optoelectronic properties.

Overall, our findings are instrumental both for fundamental and more applicative studies as many relevant solvents consist of small aromatic molecules, which therefore cannot be considered as inert media in the processing, but rather as a powerful tool for tuning the TMD properties and device optimization.

## EXPERIMENTAL SECTION

**Sample Preparation.** Monolayer  $MoS_2$  and  $WSe_2$  flakes were mechanically exfoliated from commercially available molybdenum disulfide (Furuchi, Japan) and tungsten disulfide (HQ Graphene) crystals using the scotch tape method. The flakes were transferred onto  $SiO_2$  (90 nm)/Si substrate, and their thickness was monitored by optical microscopy combined with Raman spectroscopy and atomic force microscopy. The samples were thermally annealed at 200 °C inside a vacuum chamber to desorb atmospheric adsorbates. Then, they were no longer exposed to air after the annealing and were

characterized only under inert atmosphere ( $N_2$ -filled glovebox). Anhydrous solvents from Sigma-Aldrich were opened in glovebox filled with  $N_2$ . To exclude the dielectric screening caused by environmental changes after depositing solvent molecules, we drop-cast each solvent molecule on monolayer TMDs and spin-dried at 2000 rpm for 60 s to guarantee the presence of a limited number of molecules physisorbed on the surface of the TMD.

**Low-Temperature Photoluminescence Spectroscopy.** Photoluminescence spectra were recorded in inert atmosphere ( $N_2$ ) by using a Renishaw inVia spectrometer equipped with 532 nm laser in aid of a Linkam TP95 temperature controller. Samples were mounted in the glovebox and immediately measured after annealing to avoid exposure to air. The spectra were taken at different temperatures, spanning from room temperature (298 K) to 78 K. The excitation power was kept below 1 mW to avoid local heating damage effects. The wavenumber (energy) resolution amounted to  $\sim 1$  meV.

**Device Fabrication and Electrical Characterization.** Back-gated FETs were fabricated on thermally oxidized heavily n-doped silicon substrates (Fraunhofer Institute IPMS,  $\rho_{Si} \approx 0.001 \Omega \cdot cm$ ,  $t_{ox} = 90$  nm) by means of E-beam lithography with poly(methyl methacrylate) (PMMA) resists, thermal evaporation of Au (80 nm), and lift-off in acetone. Devices were annealed in high vacuum ( $\sim 10^{-7}$  mbar) overnight at 200 °C (Plassys ME400B). Electrical characterization was carried out at room temperature under  $N_2$  atmosphere (glovebox) with source-measurement units from Keithley (model 2636A).

**Computational Details.** Our theoretical calculations were performed using density functional theory with the projector-augmented wave (PAW) scheme, as implemented in the Vienna Ab-Initio Simulation Package (VASP).<sup>34,35</sup> The exchange–correlation potentials were treated by Perdew–Burke–Ernzerhof<sup>36</sup> parametrization of the generalized gradient approximation (GGA), and the kinetic energy cutoff for the basis set was 600 eV. van der Waals interactions were taken into account using Grimme’s semiempirical DFT-D2 corrections.<sup>37</sup> To model the physisorption of solvents on  $MoS_2$  ( $WSe_2$ ) monolayer, a  $5 \times 5 \times 1$  supercell was constructed and a vacuum of 25 Å thickness was used to avoid any physical interactions in the stacking direction. The first Brillouin zone integration was performed using a  $2 \times 2 \times 1$  and  $4 \times 4 \times 1$  Monkhorst–Pack k-point mesh<sup>38</sup> for geometry optimizations and electronic structure calculations, respectively. The integral atomic positions were fully relaxed according to the Hellmann–Feynman forces until the residual forces and total

energy difference remained below 1 meV/Å and  $10^{-5}$  eV, respectively.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpcllett.8b03697.

Molecular density, distance, and orientation by theoretical calculation; molecular dipole and MoS<sub>2</sub> defects by experimental observation (PDF)

## ■ AUTHOR INFORMATION

### Corresponding Authors

\*E-mail: samori@unistra.fr.

\*E-mail: david.beljonne@umons.ac.be.

### ORCID

Jérôme Cornil: 0000-0002-5479-4227

David Beljonne: 0000-0002-2989-3557

Paolo Samori: 0000-0001-6256-8281

### Author Contributions

Y.W., S.B., and P.S. conceived the experiment. Y.W. worked on sample preparation, device fabrication, and optical and electrical characterization. Y.W., M.-A.S., and S.B. analyzed the data. A.S., J.C., and D.B. performed the modeling work. Y.W. and P.S. cowrote the manuscript. All authors discussed the results and contributed to the interpretation of data as well as to editing the manuscript.

### Notes

The authors declare no competing financial interest.

## ■ ACKNOWLEDGMENTS

We thank Y. Zhao (ISIS) for valuable discussions and precious support. Device fabrication was carried out in part at the nanotechnology facility eFab (IPCMS, Strasbourg). The authors are thankful to S. Siegwald for assistance with microfabrication. We acknowledge funding from the M-ERA.NET project MODIGLIANI, the European Commission through the Graphene Flagship (GA-785219) and the Marie-Curie IEF MULTI2DSWITCH (GA-700802), the Agence Nationale de la Recherche through the Labex projects CSC (ANR-10-LABX-0026 CSC) and NIE (ANR-11-LABX-0058 NIE) within the Investissement d'Avenir program (ANR-10-120 IDEX-0002-02), and the International Center for Frontier Research in Chemistry (icFRC). The work in Mons is supported by FNRS/F.R.S. J.C. and D.B. are FNRS Research Directors.

## ■ REFERENCES

- (1) Akinwande, D.; Petrone, N.; Hone, J. Two-dimensional flexible nanoelectronics. *Nat. Commun.* **2014**, *5*, 5678.
- (2) Fiori, G.; Bonaccorso, F.; Iannaccone, G.; Palacios, T.; Neumaier, D.; Seabaugh, A.; Banerjee, S. K.; Colombo, L. Electronics Based on Two-Dimensional Materials. *Nat. Nanotechnol.* **2014**, *9* (10), 768–779.
- (3) Mak, K. F.; Shan, J. Photonics and optoelectronics of 2D semiconductor transition metal dichalcogenides. *Nat. Photonics* **2016**, *10*, 216.
- (4) Hu, Y.; Huang, Y.; Tan, C.; Zhang, X.; Lu, Q.; Sindoro, M.; Huang, X.; Huang, W.; Wang, L.; Zhang, H. Two-dimensional transition metal dichalcogenide nanomaterials for biosensing applications. *Mater. Chem. Frontiers* **2017**, *1* (1), 24–36.
- (5) Li, S.-L.; Tsukagoshi, K.; Orgiu, E.; Samori, P. Charge transport and mobility engineering in two-dimensional transition metal chalcogenide semiconductors. *Chem. Soc. Rev.* **2016**, *45* (1), 118–151.
- (6) Yan, W.; Txoperena, O.; Llopis, R.; Dery, H.; Hueso, L. E.; Casanova, F. A two-dimensional spin field-effect switch. *Nat. Commun.* **2016**, *7*, 13372.
- (7) Anichini, C.; Czepa, W.; Pakulski, D.; Aliprandi, A.; Ciesielski, A.; Samori, P. Chemical sensing with 2D materials. *Chem. Soc. Rev.* **2018**, *47*, 4860–4908.
- (8) Bertolazzi, S.; Gobbi, M.; Zhao, Y.; Backes, C.; Samori, P. Molecular chemistry approaches for tuning the properties of two-dimensional transition metal dichalcogenides. *Chem. Soc. Rev.* **2018**, *47*, 6845–6888.
- (9) Mouri, S.; Miyauchi, Y.; Matsuda, K. Tunable photoluminescence of monolayer MoS<sub>2</sub> via chemical doping. *Nano Lett.* **2013**, *13* (12), 5944–5948.
- (10) Lee, I.; Rathi, S.; Li, L.; Lim, D.; Khan, M. A.; Kannan, E. S.; Kim, G.-H. Non-degenerate n-type doping by hydrazine treatment in metal work function engineered WSe<sub>2</sub> field-effect transistor. *Nanotechnology* **2015**, *26* (45), 455203.
- (11) Kiriya, D.; Tosun, M.; Zhao, P.; Kang, J. S.; Javey, A. Air-stable surface charge transfer doping of MoS<sub>2</sub> by benzyl viologen. *J. Am. Chem. Soc.* **2014**, *136* (22), 7853–7856.
- (12) Splendiani, A.; Sun, L.; Zhang, Y. B.; Li, T. S.; Kim, J.; Chim, C. Y.; Galli, G.; Wang, F. Emerging Photoluminescence in Monolayer MoS<sub>2</sub>. *Nano Lett.* **2010**, *10* (4), 1271–1275.
- (13) Mak, K. F.; Lee, C.; Hone, J.; Shan, J.; Heinz, T. F. Atomically Thin MoS<sub>2</sub>: A New Direct-Gap Semiconductor. *Phys. Rev. Lett.* **2010**, *105* (13), 136805.
- (14) Ross, J. S.; Wu, S.; Yu, H.; Ghimire, N. J.; Jones, A. M.; Aivazian, G.; Yan, J.; Mandrus, D. G.; Xiao, D.; Yao, W.; et al. Electrical control of neutral and charged excitons in a monolayer semiconductor. *Nat. Commun.* **2013**, *4*, 1474.
- (15) Tongay, S.; Zhou, J.; Ataca, C.; Liu, J.; Kang, J. S.; Matthews, T. S.; You, L.; Li, J.; Grossman, J. C.; Wu, J. Broad-range modulation of light emission in two-dimensional semiconductors by molecular physisorption gating. *Nano Lett.* **2013**, *13* (6), 2831–6.
- (16) Gobbi, M.; Orgiu, E.; Samori, P. When 2D Materials Meet Molecules: Opportunities and Challenges of Hybrid Organic/Inorganic van der Waals Heterostructures. *Adv. Mater.* **2018**, *30*, 1706103.
- (17) Jing, Y.; Tan, X.; Zhou, Z.; Shen, P. W. Tuning electronic and optical properties of MoS<sub>2</sub> monolayer via molecular charge transfer. *J. Mater. Chem. A* **2014**, *2* (40), 16892–16897.
- (18) Han, H. V.; Lu, A. Y.; Lu, L. S.; Huang, J. K.; Li, H.; Hsu, C. L.; Lin, Y. C.; Chiu, M. H.; Suenaga, K.; Chu, C. W.; Kuo, H. C.; Chang, W. H.; Li, L. J.; Shi, Y. Photoluminescence enhancement and structure repairing of monolayer MoSe<sub>2</sub> by hydrohalic acid treatment. *ACS Nano* **2016**, *10* (1), 1454–1461.
- (19) Lockhart de la Rosa, C. J.; Phillipson, R.; Teyssandier, J.; Adisojoso, J.; Balaji, Y.; Huyghebaert, C.; Radu, I.; Heyns, M.; De Feyter, S.; De Gendt, S. Molecular doping of MoS<sub>2</sub> transistors by self-assembled oleylamine networks. *Appl. Phys. Lett.* **2016**, *109* (25), 253112.
- (20) Kang, D. H.; Dugasani, S. R.; Park, H. Y.; Shim, J.; Gnappareddy, B.; Jeon, J.; Lee, S.; Roh, Y.; Park, S. H.; Park, J. H. Ultra-low Doping on Two-Dimensional Transition Metal Dichalcogenides using DNA Nanostructure Doped by a Combination of Lanthanide and Metal Ions. *Sci. Rep.* **2016**, *6*, 20333.
- (21) Fang, H.; Chuang, S.; Chang, T. C.; Takei, K.; Takahashi, T.; Javey, A. High-performance single layered WSe<sub>2</sub> p-FETs with chemically doped contacts. *Nano Lett.* **2012**, *12* (7), 3788–3792.
- (22) Fisher, A.; Blöchl, P. Adsorption and scanning-tunneling-microscope imaging of benzene on graphite and MoS<sub>2</sub>. *Phys. Rev. Lett.* **1993**, *70* (21), 3263.
- (23) Dong, X.; Fu, D.; Fang, W.; Shi, Y.; Chen, P.; Li, L. J. Doping single-layer graphene with aromatic molecules. *Small* **2009**, *5* (12), 1422–1426.

(24) Korn, T.; Heydrich, S.; Hirmer, M.; Schmutzler, J.; Schüller, C. Low-temperature photocarrier dynamics in monolayer MoS<sub>2</sub>. *Appl. Phys. Lett.* **2011**, *99* (10), 102109.

(25) Lanzillo, N. A.; Birdwell, A. G.; Amani, M.; Crowne, F. J.; Shah, P. B.; Najmaei, S.; Liu, Z.; Ajayan, P. M.; Lou, J.; Dubey, M.; Nayak, S. K.; O'Regan, T. P. Temperature-dependent phonon shifts in monolayer MoS<sub>2</sub>. *Appl. Phys. Lett.* **2013**, *103* (9), 093102.

(26) Mak, K. F.; He, K. L.; Lee, C.; Lee, G. H.; Hone, J.; Heinz, T. F.; Shan, J. Tightly bound trions in monolayer MoS<sub>2</sub>. *Nat. Mater.* **2013**, *12* (3), 207–211.

(27) Crispin, X.; Geskin, V.; Crispin, A.; Cornil, J.; Lazzaroni, R.; Salaneck, W. R.; Bredas, J.-L. Characterization of the interface dipole at organic/metal interfaces. *J. Am. Chem. Soc.* **2002**, *124* (27), 8131–8141.

(28) Pearson, R. G. The electronic chemical potential and chemical hardness. *J. Mol. Struct.: THEOCHEM* **1992**, *255*, 261–270.

(29) Tang, W.; Sanville, E.; Henkelman, G. A grid-based Bader analysis algorithm without lattice bias. *J. Phys.: Condens. Matter* **2009**, *21* (8), 084204.

(30) Choi, J.; Zhang, H.; Du, H.; Choi, J. H. Understanding solvent effects on the properties of two-dimensional transition metal dichalcogenides. *ACS Appl. Mater. Interfaces* **2016**, *8* (14), 8864–8869.

(31) Ross, J. S.; Klement, P.; Jones, A. M.; Ghimire, N. J.; Yan, J.; Mandrus, D.; Taniguchi, T.; Watanabe, K.; Kitamura, K.; Yao, W.; et al. Electrically tunable excitonic light-emitting diodes based on monolayer WSe<sub>2</sub> p–n junctions. *Nat. Nanotechnol.* **2014**, *9* (4), 268.

(32) Jones, A. M.; Yu, H.; Ghimire, N. J.; Wu, S.; Aivazian, G.; Ross, J. S.; Zhao, B.; Yan, J.; Mandrus, D. G.; Xiao, D.; et al. Optical generation of excitonic valley coherence in monolayer WSe<sub>2</sub>. *Nat. Nanotechnol.* **2013**, *8* (9), 634.

(33) You, Y.; Zhang, X.-X.; Berkelbach, T. C.; Hybertsen, M. S.; Reichman, D. R.; Heinz, T. F. Observation of biexcitons in monolayer WSe<sub>2</sub>. *Nat. Phys.* **2015**, *11* (6), 477.

(34) Kresse, G.; Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1996**, *54* (16), 11169.

(35) Kresse, G.; Joubert, D. From ultrasoft pseudopotentials to the projector augmented-wave method. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1999**, *59*, 1758.

(36) Perdew, J. P.; Burke, K.; Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **1996**, *77* (18), 3865.

(37) Grimme, S. Semiempirical GGA-type density functional constructed with a long-range dispersion correction. *J. Comput. Chem.* **2006**, *27* (15), 1787–1799.

(38) Monkhorst, H. J.; Pack, J. D. Special points for Brillouin-zone integrations. *Phys. Rev. B* **1976**, *13* (12), 5188.