



# Optical study of oriented double- $\text{Se}_8$ -ring clusters and luminescent $\text{Se}_2^-$ anions in LTA at extremely high selenium loading density

Vladimir V. Poborchii<sup>a,\*</sup>, Vitalii P. Petranovskii<sup>b</sup>, Igor A. Glukhov<sup>c,d</sup>, Andrei A. Fotiadi<sup>e</sup>

<sup>a</sup> National Institute of Advanced Industrial Science and Technology, Tsukuba, 305-8565, Japan

<sup>b</sup> Centro de Nanociencias y Nanotecnología, Universidad Nacional Autónoma de México, Ensenada, Baja California, Mexico

<sup>c</sup> Lab-STICC (UMR CNRS 6285), École Nationale d'Ingénieurs de Brest, France

<sup>d</sup> Ulyanovsk State University, 42 Leo Tolstoy Street, Ulyanovsk, 432970, Russia

<sup>e</sup> Optoelectronics and Measurement Techniques unit, University of Oulu, Oulu, Finland

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## ABSTRACT

Recently, LTA-Se(1–8) samples with 1–8 Se atoms per cavity (simplified unit cell, large cavity + sodalite cage) obtained via adsorption at the temperature of  $\sim 450$  °C were reported. It was shown that single  $\text{Se}_8$  or single  $\text{Se}_{12}$  ring are formed in the large LTA cavities,  $\text{Se}_8/\text{Se}_{12}$  ring concentration ratio decreasing with an increase in the Se loading density. Contrary, in the present work, using Se vapour adsorption at  $\sim 550$  °C, we succeeded in encapsulation of  $\sim 17$  Se atoms per cavity (LTA-Se(17)) with a significant increase in the  $\text{Se}_8/\text{Se}_{12}$  concentration ratio manifesting double  $\text{Se}_8$ -ring cluster formation in the most of the LTA large cavities, which is a step towards cluster crystal fabrication. According to our polarization/orientation Raman spectroscopic study of LTA-Se(17) single crystals, the orientations of the  $\text{Se}_8$  and  $\text{Se}_{12}$  appeared to be similar to those in previously investigated LTA-Se(1–8). Importantly, luminescent  $\text{Se}_2^-$  anions, oriented along the LTA 4-fold axes and located in the sodalite cages, are detected via Raman polarization/orientation dependencies of LTA-Se(17). Bright  $\text{Se}_2^-$  light emission with a maximum at  $\sim 1.56$  eV and vibronic structure is observed in the 1.3–1.8 eV spectral range. We show that the anions experience a compression in LTA which is slightly relaxing with a decrease in temperature producing an anomalous Raman band downshift. The compression of  $\text{Se}_2^-$  in LTA is weaker/stronger than that in sodalite/cancrinite, luminescence band photon energy depending on its strength. High concentration of regularly arranged  $\text{Se}_2^-$  in LTA suggests considering LTA-Se(17) as an important novel light-emitting material.

## 1. Introduction

Selenium is considered as an important material since the 19th century due to its photosensitivity [1] and, especially, from the mid-20th century due to its applications in xerography [2]. In the 21st century, Se and its compounds are employed in solar-energy devices [3], topological-insulator materials [4], lithium batteries [5] etc. A few years ago, Se nanoparticles were explored as anti-microbial/cancer/oxidant agents and tunable light-emitting bio-markers [6–8]. Zeolites themselves are important aluminosilicate materials with uniform sub-nanometer scale pores enabling molecular hosting and exhibiting unique chemical, optical and elastic properties [9–11]. Formation of LTA-based functional materials with a resonance electronic inter-cluster interaction at high loading densities is another important point. Therefore, LTA zeolite with regularly arranged highly-concentrated Se

particles is attractive subject for research nowadays.

Here, we focus on the optical properties of Se species confined and regularly arranged in the  $\sim 1.14$  nm size large cavities and  $\sim 0.63$  nm size sodalite cages of the LTA zeolite. Using Se vapour adsorption at the temperature  $t \sim 550$  °C, we succeeded in encapsulation of  $\sim 17$  Se atoms per simplified unit cell (large cavity + sodalite cage), which is significantly higher than the Se loading densities obtained in our previous works, where adsorption was done at  $t \sim 450$  °C [12–15]. In those works, we were able to univocally identify Raman and optical absorption bands of previously unknown  $\text{Se}_{12}$  rings. However, in order to make a step towards fabrication of a new functional material with the interacting Se clusters, we need to increase Se loading density, which is achieved in the present work. Polarized Raman spectra (RS) of so-prepared LTA-with-Se single crystals (LTA-Se(17)) show that the most of Se is stabilized in the form of double- $\text{Se}_8$ -ring clusters in the LTA

\* Corresponding author.

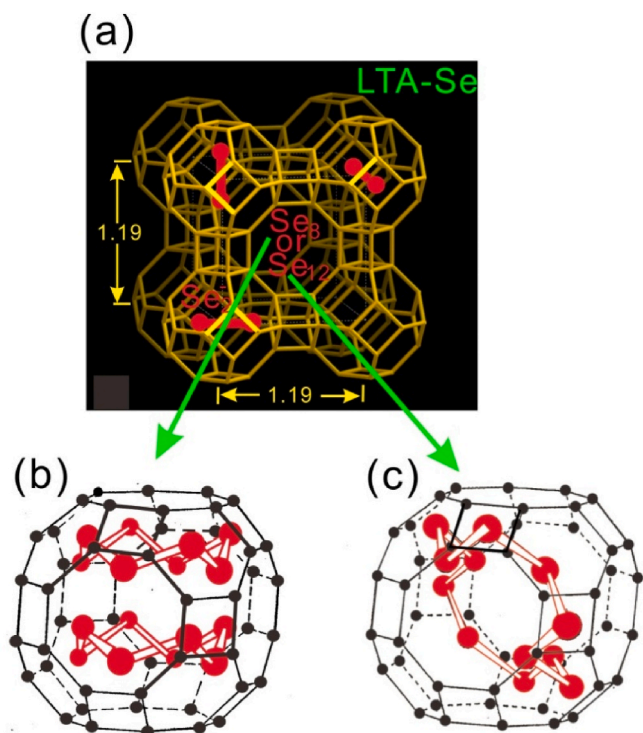
E-mail address: [Vladimir.Poborchii@gmail.com](mailto:Vladimir.Poborchii@gmail.com) (V.V. Poborchii).

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**Fig. 1.** Framework of LTA (a);  $\text{Se}_8$  double ring cluster (b) and  $\text{Se}_{12}$  (c) schematic view in the LTA large cavity.  $\text{Se}_2$  anions are shown in the sodalite cages (a).

large cavities, with the rest of the large cavities being occupied by single  $\text{Se}_{12}$  rings. Importantly, luminescent  $\text{Se}_2$  anions appeared to be formed in the LTA sodalite cages at  $t \sim 550^\circ\text{C}$ . The anions experience a compression in LTA which is slightly relaxing with a decrease in temperature producing an anomalous Raman band downshift, luminescence band photon energy depending on its strength. Their high concentration and regular arrangement suggests considering LTA-Se(17) as a promising light-emitting material. We should stress that our polarization/orientation Raman technique allows studying relatively small LTA single crystals compared to the size of  $>40\ \mu\text{m}$  required for X-ray diffraction (XRD). Another advantage of our technique is a possibility of identifying types and orientations of a variety of molecules in the same crystal, which is difficult for XRD due to partial occupancies of Se atom sites. Additionally, we have to note that such periodically arranged system of Se clusters and light-emitting centers could be an interesting system for high-spatial-resolution luminescence and Raman/Brillouin imaging techniques.

## 2. Experimental

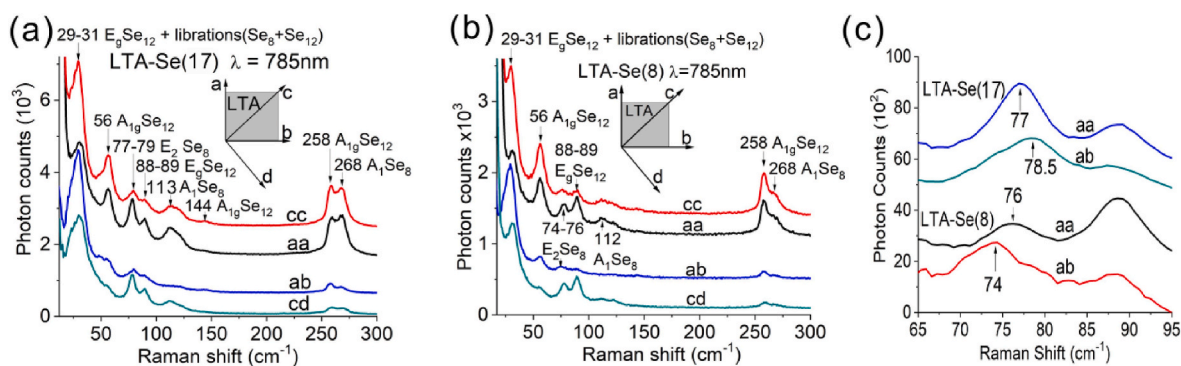
Synthetic LTA zeolites ( $\text{Na}_{12}\text{Al}_{12}\text{Si}_{12}\text{O}_{48}$ ) were used in this work. LTA has nearly spherical-shape large cavities with diameter of  $\sim 1.14\ \text{nm}$  which are connected through narrow windows of  $\sim 0.42\ \text{nm}$  diameter (Fig. 1). Each large cavity is accompanied by a small cavity, sodalite cage, with the diameter of  $\sim 0.63\ \text{nm}$ . The zeolite structures are accessible at the International Zeolite Association website [http://www.iza-structure.org/IZA-SC/ftc\\_table.php](http://www.iza-structure.org/IZA-SC/ftc_table.php). The sizes of the cubic LTA crystals were  $\sim 20\ \mu\text{m}$  along the edge of the cube. Dehydration of zeolites in vacuum was performed in pyrex ampoules at  $t \sim 550^\circ\text{C}$ . The selenium vapour adsorption was done at the same temperature during a week. Amount of adsorbed Se was determined via an accurate zeolite powder and selenium weighing before the adsorption and then controlled with the LTA-Se density measurement using single crystal flotation in Clerici solutions [16].

RS were studied using Renishaw micro-Raman spectrometer with  $\sim 1\ \mu\text{m}$  focused laser probe size. The  $514.5\ \text{nm}$  line of the  $\text{Ar}^+$  laser and  $785\ \text{nm}$  line of the infrared light-emitting-diode laser were used for the RS excitation. Photo-luminescence spectra excited with the  $514.5\ \text{nm}$  line were recorded using the same spectrometer. Optical absorption spectra (OAS) in the visible and near UV spectral ranges were studied using Carl Zeiss and CRAIC micro-optical spectrometers, the light probe size being  $<10\ \mu\text{m}$ . The samples were intentionally broken to minimize their optical density and placed into glycerol between two cover glasses to avoid surface light scattering.

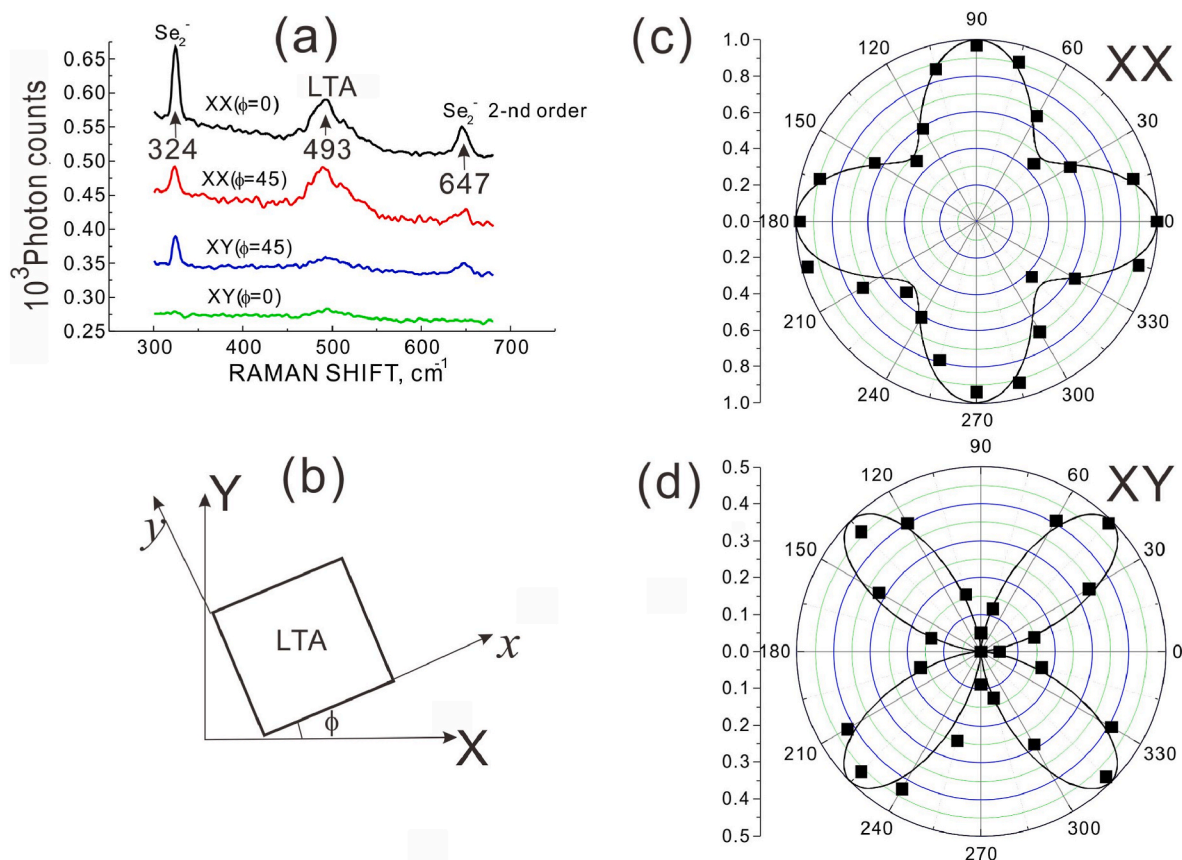
## 3. Double $\text{Se}_8$ ring clusters in LTA-Se(17): a step towards a cluster crystal

At the Se adsorption temperature of  $\sim 550^\circ\text{C}$ , we succeeded in significant increase in the Se loading density in LTA-Se up to  $\sim 17\ \text{at./cav.}$  (here cavity (cav.) means the simplified unit cell). No sign of the zeolite water was observed in RS and IR spectra of so-prepared LTA-Se(17). Formally, the obtained loading density of  $\sim 17\ \text{at./cav.}$  roughly corresponds to two  $\text{Se}_8$  rings per cavity. However, our spectroscopic data show that the situation is more complicated. At first, we have found that in addition to the Se ring molecules,  $\text{Se}_2$  anions are formed in the LTA-Se(17) (we discuss the  $\text{Se}_2$  anions in the following section). At second, still significant amount of  $\text{Se}_{12}$  rings is found in this sample like in LTA-Se prepared at  $t \sim 450^\circ\text{C}$  [12–15].

Fig. 2(a) shows RS of LTA-Se(17) for different polarization configurations (first/second index stands for the polarization of incident/scattered light) at the  $785\ \text{nm}$  excitation wavelength. Spectra for LTA-Se(8) from Ref. [13] are shown in Fig. 2(b). In contrast with the tendency observed for the increase of Se loading density from 1 to 8 at./cav. showing a gradual decrease in the  $\text{Se}_8/\text{Se}_{12}$  Raman band intensity ratio [13], the increase from 8 to 17 at./cav. clearly shows an increase in the  $\text{Se}_8/\text{Se}_{12}$  band intensity ratio Fig. 2(a and b). This fact is an argument for



**Fig. 2.** Raman spectra of LTA-Se(17) (a), LTA-Se(8) (b) for cc-, aa-, ab- and cd-polarization configurations at the  $785\ \text{nm}$  excitation wavelength with the axis directions shown in the insets;  $\text{Se}_8\ \text{E}_2$  mode band splitting for both samples (c).



**Fig. 3.** Raman spectra of  $\text{Se}_2^-$  in LTA-Se(17) for different polarization configurations at the 514.5 nm excitation wavelength (a); schematic view of the LTA-Se(17) crystal rotated in the laboratory co-ordinate XY-plane (b); polarization/orientation dependencies of the  $\text{Se}_2^-$  band intensity on the angle  $\varphi$  for XX (c) and XY (d) polarization configurations, lines standing for theoretical calculations and squares representing experimental points.

the double  $\text{Se}_8$  ring formation in the large LTA cavities. Taking into account an approximate  $\text{Se}_8/\text{Se}_{12}$  concentration ratio of  $\sim 3.5$  at the 8 at./cav. Se loading density [13] and the observed here a factor of  $\sim 2.3$  increase in the  $\text{Se}_8/\text{Se}_{12}$  RS band intensity ratio in LTA-Se(17), compared to LTA-Se(8), we conclude that  $\sim 20\%$  of the cavities are occupied with  $\text{Se}_{12}$  while the rest of the large cavities are filled with the double  $\text{Se}_8$  rings. Also taking into account  $\text{Se}_2^-$  anions, which are, probably, located in nearly every sodalite cage, we obtain  $\sim 17$  at./cav. in agreement with the experimental Se-loading-density value.

RS polarization dependencies in Fig. 2(a and b) are similar to each other. Therefore, the ring orientations in both samples are like those reported earlier for LTA-Se(8) [13,15] with the only difference that LTA-Se(17) has two  $\text{Se}_8$  rings in the large cavity (Fig. 1(b)) while LTA-Se(8) has only one  $\text{Se}_8$  [13]. The  $\text{Se}_{12}$  ring orientation is shown in Fig. 1(c).

In the restricted space of the LTA large cavity, two  $\text{Se}_8$  rings are slightly compressed compared to a single  $\text{Se}_8$  ring. One can see a confirmation for this in RS of LTA-Se(17). Indeed, the  $A_1$  symmetric bond-bending mode band of a double  $\text{Se}_8$  ring cluster in LTA-Se(17) at  $\sim 113 \text{ cm}^{-1}$  displays an upshift compared to  $\sim 112 \text{ cm}^{-1}$  for a single  $\text{Se}_8$  in LTA-Se(8). Both values are smaller than  $\sim 114 \text{ cm}^{-1}$  of the  $\text{Se}_8$  ring in AFI-Se indicating rather strong compression of the ring in the 0,73 nm wide AFI channel [13].

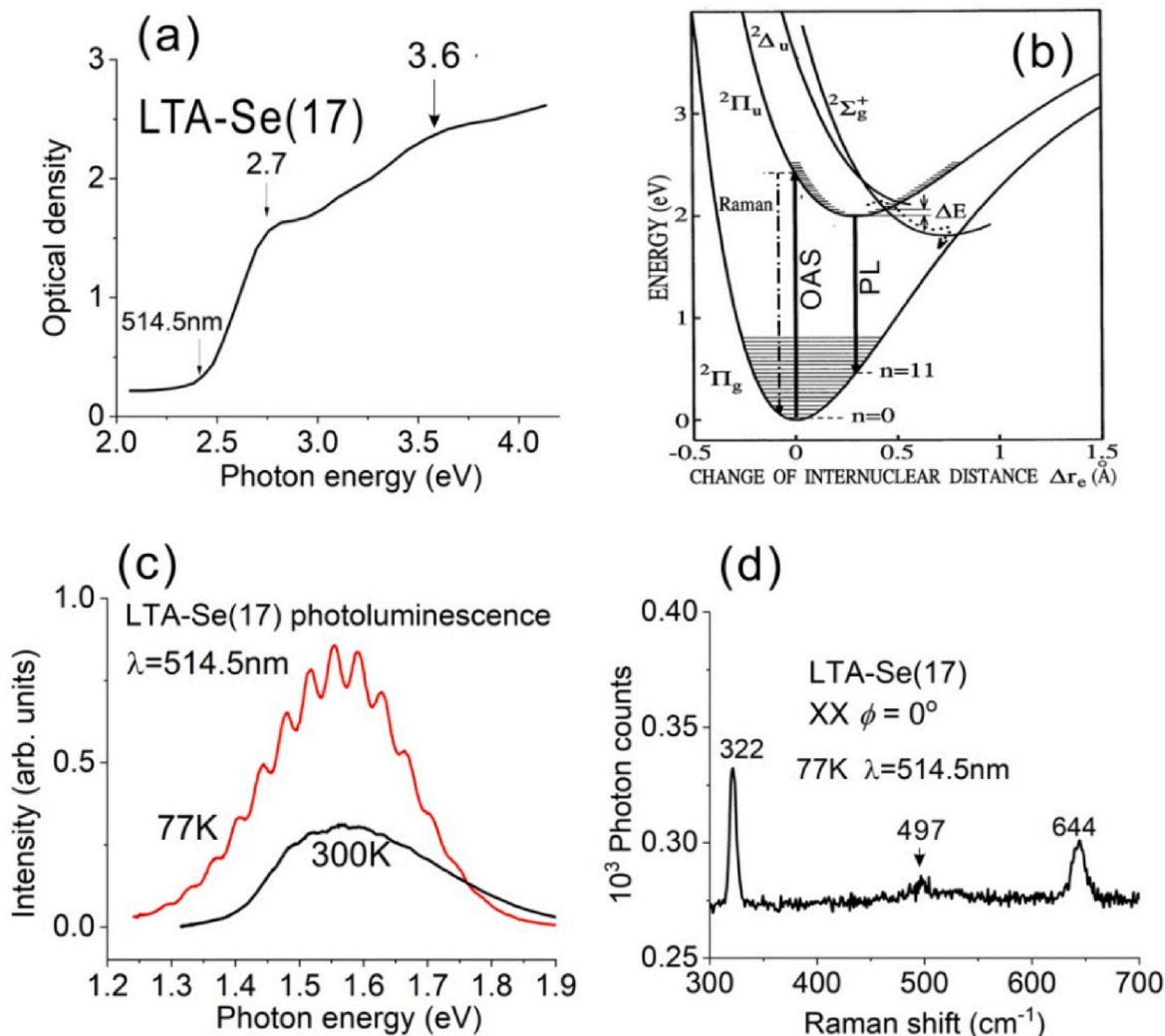
An interesting change is observed for the  $E_2$  bond-bending mode that is split off due to a zeolite-induced reduction of the  $\text{Se}_8$  ring  $D_{4d}$  symmetry (Fig. 2(c)). The  $E_2$  bond bending mode band of the double ring cluster displays both upshift and reverse split off compared to that of the single  $\text{Se}_8$  ring. For a single  $\text{Se}_8$  ring in LTA-Se(8), the **aa**-active component displays frequency of  $\sim 76 \text{ cm}^{-1}$  which is higher than that of the **ab**-active component of  $\sim 74 \text{ cm}^{-1}$ . However, for a double  $\text{Se}_8$  ring cluster in LTA-Se(17), the **aa**-active component displays frequency of

$\sim 77 \text{ cm}^{-1}$  which is lower than that of the **ab**-active component at  $\sim 78.5 \text{ cm}^{-1}$ . Probably, a single  $\text{Se}_8$  ring LTA-Se(8) is rather centrally located in the cavity while the  $\text{Se}_8$  location in LTA-Se(17) is more peripheral. Therefore, the symmetry distortion of the ring is different in these two cases. Anyway, our data definitely suggest that the dihedral and bond angles of double  $\text{Se}_8$  rings are decreased compared to a single  $\text{Se}_8$  ring to make the double rings more compact.

Turning back to the basic ideas of the cluster crystal fabrication [17–19], we stress that making a regular lattice of clusters in LTA is very important. The cluster crystal requirement includes uniformity of the cavity-confined clusters and their electron-resonance interaction. Compared to single  $\text{Se}_8/\text{Se}_{12}$  ring formation in LTA-Se(8) [13], the double  $\text{Se}_8$ -ring fabrication in the most of the LTA cavities is a big step towards this goal.

#### 4. Raman, optical absorption and luminescence spectra of $\text{Se}_2^-$ in LTA-Se(17): its compression and anomalous temperature dependence of the vibration frequency

RS of  $\text{Se}_2^-$  formed in LTA-Se(17) taken with the 514.5 nm wavelength laser are shown in Fig. 3(a) for four polarization configurations. ( $\varphi$  is the angle between the laboratory X-axis and the crystal [100] x-axis (Fig. 3 (b)). The 1st-order Raman band of  $\text{Se}_2^-$  is displayed at  $\sim 324 \text{ cm}^{-1}$  and the 2nd-order one at  $\sim 647 \text{ cm}^{-1}$ . The observed 1st-order frequency is close to the frequencies observed for the same anion in other zeolites such as SOD-Se ( $331\text{--}334 \text{ cm}^{-1}$ ) [20], CAN-Se ( $320 \text{ cm}^{-1}$ ) [21] and Nd-FAU-Se ( $328 \text{ cm}^{-1}$ ) [22], the frequencies being dependent on the interaction with zeolites. Taking into account the fact that the  $\text{Se}_2^-$  is Raman active at the 514.5 nm excitation only when both incident and scattered lights are polarized along the molecule axis [21], the LTA-Se



**Fig. 4.** Room temperature OAS of LTA-Se(17) (a); Se<sub>2</sub><sup>-</sup> electron structure [26] with schematically shown absorption (OAS) and luminescence (PL) transitions (b); LTA-Se(17) PL spectra at the temperatures of ~300 K (black) and ~77 K (red) (c); RS of LTA-Se(17) at ~77 K. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

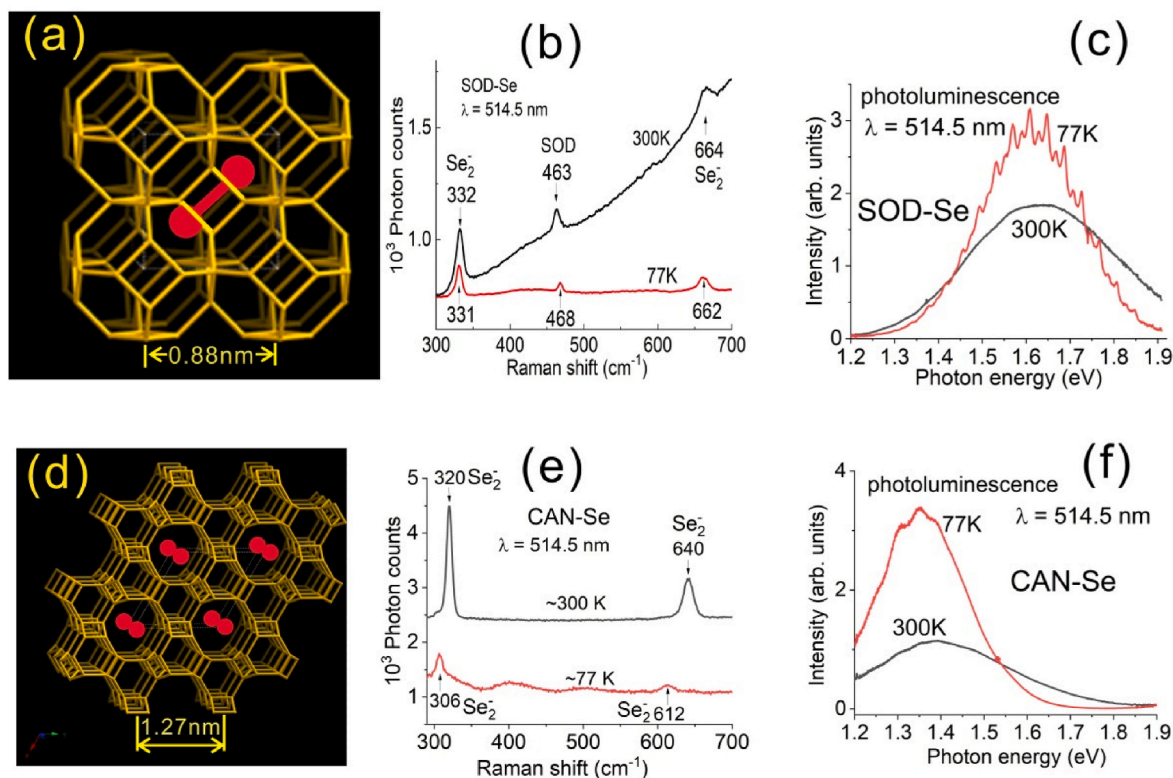
(17) polarization dependence clearly shows that the Se<sub>2</sub><sup>-</sup> anions are oriented along the 4-fold axes of LTA crystals. Small size and negative charge of the species suggest that they are, probably, located in sodalite cages of LTA (Fig. 1(a)) like S<sub>3</sub><sup>-</sup> [23]. Interestingly, in case of CAN-Se, luminescent Se<sub>2</sub><sup>-</sup> are formed when Se or Na<sub>2</sub>Se<sub>4</sub> was added to initial components used for the synthesis of cancrinite [18]. Contrary, only Se<sub>2</sub><sup>2-</sup> are formed in CAN-Se obtained via Se vapour adsorption [24,25].

Importantly, Se<sub>2</sub><sup>-</sup> is a very suitable example of the polarization-orientation dependencies of RS intensity of a linear molecule located in a cubic crystal. Fig. 3 shows such dependencies for the XX(c) and XY (d) polarization configurations vs. the angle φ between the X-direction of the laboratory system of co-ordinates and the x-axis of the LTA crystal (Fig. 3(b)). Using the method [16] based on a general formula for RS efficiency of vibrations, we can easily find that the contribution of Se<sub>2</sub><sup>-</sup> oriented along the x- and y-axis to the XX-configuration RS displays cos<sup>4</sup>(φ) and sin<sup>4</sup>(φ) dependencies, respectively. Summation gives us sin<sup>4</sup>(φ) + cos<sup>4</sup>(φ) dependence clearly reproduced experimentally in Fig. 3 (c). For the XY-configuration, both x-oriented and y-oriented molecules contribute like sin<sup>2</sup>(φ)cos<sup>2</sup>(φ) in good agreement with the experiment shown in Fig. 3(d). The Se<sub>2</sub><sup>-</sup> bands completely vanish in the XY-configuration at φ = 0° (Fig. 4(a,d)), which confirms the perfect dimer orientation along the 4-fold axis of LTA.

OAS measurement of LTA-Se(17) was rather difficult due to the high

absorption of the sample. Even ~0.1 μm thick pieces of LTA-Se(17) crystals showed very high optical densities. Typical OAS of LTA-Se (17) is shown in Fig. 4(a). The Se<sub>2</sub><sup>-</sup> band at ~2.7 eV due to the 2Π<sub>g</sub> → 2Π<sub>u</sub> electron transition [20,26] (Fig. 4(b)) strongly contributes to the spectrum. The Se<sub>12</sub> and Se<sub>8</sub> rings mainly contribute to the absorption at the photon energies >3 eV [13]. The absorption peak at ~3.6 eV belongs to the double Se<sub>8</sub> ring cluster, the energy being slightly lower than that of single Se<sub>8</sub> at 3.75–3.85 eV [13].

It is known that the Se<sub>2</sub><sup>-</sup> anions display bright photo-luminescence (PL) in the red-near-infrared spectral range [20]. Steady-state PL spectra of LTA-Se(17) excited with the 514.5 nm light at the temperatures of ~300 K and ~77 K are shown in Fig. 4(c). A broad luminescence band with a maximum at ~1.56 eV and a distinct vibronic structure is displayed in the LTA-Se(17) PL spectrum taken at ~77 K. The band is related to the 2Π<sub>u</sub> → 2Π<sub>g</sub> electron transition like that in SOD-Se [20], reverse to the absorption electron transition between the same molecular electronic states. The Se loading density of ~17 at./cav. suggests that the anions occupy nearly every sodalite cage. Otherwise, we need to admit that some additional optically invisible Se species are formed in LTA-Se. As we mentioned above, no Se<sub>2</sub><sup>-</sup> was formed at the adsorption temperature of ~450 °C [13]. Probably, higher concentration of Se<sub>2</sub> molecules in Se vapour and larger amplitude of sodalite-cage-window vibrations at ~550 °C play a crucial role in the encapsulation of Se<sub>2</sub> in



**Fig. 5.** Schematic view of  $\text{Se}_2^-$  in SOD-Se with the molecule orientation proposed in Ref. [20] (a); RS of SOD-Se at the temperatures of  $\sim 300$  K (black) and  $\sim 77$  K (red) (b); PL spectra of SOD-Se at the temperatures of  $\sim 300$  K (black) and  $\sim 77$  K (red) (c); schematic view of  $\text{Se}_2^-$  in CAN-Se (d); RS of SOD-Se at the temperatures of  $\sim 300$  K (black) and  $\sim 77$  K (red) for incident and scattered lights polarized along CAN channels (e); PL spectra of CAN-Se at the temperatures of  $\sim 300$  K (black) and  $\sim 77$  K (red) (f). (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

sodalite cages. A neutral  $\text{Se}_2$  molecule, after occupation of the sodalite cage borrows an electron from the zeolite framework. Theoretically, we can place  $\text{Se}_2^-$  in the window between two large cavities with the orientation along the 4-fold axis. This would be beneficial for the interaction of the clusters in the large cavities, which is important for the cluster crystal. However,  $\text{Se}_2^-$  location in the sodalite cage looks more reasonable.

Fig. 4(d) shows RS of LTA-Se(17) taken at the temperature of  $\sim 77$  K, other conditions being equal to those of RS in Fig. 3(a), black curve. Surprisingly, instead of the expected Raman upshift with a decrease in temperature (like the LTA band does), the  $\text{Se}_2^-$  band displays a noticeable downshift from  $324\text{ cm}^{-1}$  to  $\sim 322\text{ cm}^{-1}$ . This can be explained by the anion compression at  $\sim 300$  K and its partial relaxation at  $\sim 77$  K. It is confirmed by the significant enhancement of the  $\text{Se}_2^-$  Raman intensity compared to that of the zeolite one with a decrease in temperature. According to the  $\text{Se}_2^-$  electronic structure, an increase in the anion's interatomic distance causes a decrease in the electron energy gap between the  $^2\Pi_g$  and  $^2\Pi_u$  states and red shift of the  $\text{Se}_2^-$  absorption band. Therefore, the resonance Raman condition should be improved at the  $514.5\text{ nm}$  excitation when the  $\text{Se}_2^-$  absorption band positioned at  $\sim 2.7\text{ eV}$  at  $\sim 300$  K is downshifted at  $\sim 77$  K.

Let us focus on the environment influence on the  $\text{Se}_2^-$  properties in other zeolites. Fig. 5 shows positions, RS and PL spectra of  $\text{Se}_2^-$  in SOD-Se with  $\sim 0.63\text{ nm}$  diameter pores and CAN-Se with  $\sim 0.6\text{ nm}$  diameter channels. The room-temperature vibration frequency increase from  $\sim 320\text{ cm}^{-1}$  in CAN-Se (Fig. 5(e)) to  $\sim 332\text{ cm}^{-1}$  in SOD-Se (Fig. 5(b)) suggests a quite strong compression of  $\text{Se}_2^-$  anions in SOD. Indeed,  $\text{Se}_2^-$  is confined only in two dimensions in the CAN channels and it is not confined in the anion axis direction (Fig. 5(d)). Contrary,  $\text{Se}_2^-$  is confined in all three dimensions in the cages of SOD (Fig. 5(a)). Correspondingly, the frequency of  $\sim 320\text{ cm}^{-1}$  of the anion in CAN-Se indicates its larger inter-atomic distance than that in the compressed one in SOD-Se with

**Table 1**

Raman and photo-luminescence peak positions of  $\text{Se}_2^-$  anion formed in cancrinite, LTA and sodalite.

$\text{Se}_2^-$	CAN	LTA	SOD
frequency ( $\text{cm}^{-1}$ ) at 300/77 K	320/306	324/322	332/331
PL peak (eV) at 300/77 K	1.39/1.35	1.57/1.56	1.63/1.62

the frequency of  $\sim 332\text{ cm}^{-1}$ .  $\text{Se}_2^-$  in LTA-Se(17) displays an intermediate frequency of  $\sim 324\text{ cm}^{-1}$  (Fig. 3(a)) corresponding to a relatively slight compression of the anion in the sodalite cage of LTA. The photon energy positions of the  $\text{Se}_2^-$  luminescence bands in three zeolites (Figs. 4(c) and 5(c,f)) appear to be in agreement with the compression effect on the anion's electronic structure (Table 1).

Temperature dependence of  $\text{Se}_2^-$  RS in SOD-Se (Fig. 5(b)) is similar to that in LTA-Se(17), namely, its frequency of  $\sim 331\text{ cm}^{-1}$  at  $\sim 77$  K is lower than  $\sim 332\text{ cm}^{-1}$  at  $\sim 300$  K. Contrary, the sodalite band frequency of  $\sim 468\text{ cm}^{-1}$  at  $\sim 77$  K is higher than  $\sim 463\text{ cm}^{-1}$  at  $\sim 300$  K in accordance with regular temperature-induced expansion/contraction. Temperature dependence of  $\text{Se}_2^-$  RS in CAN-Se (Fig. 5(e)) suggests effect of combining of two or more  $\text{Se}_2^-$  molecules into chain at low temperatures [21].

Probably,  $\text{Na}^+$  cations play important role in a strong  $\text{Se}_2^-$  compression in SOD-Se. Orientation of the anion surrounded with three  $\text{Na}^+$  cations along the 3 fold axis of SOD (Fig. 5(a)) was proposed in Ref. [20]. Contrary, as we found via the polarization/orientation RS, LTA-confined  $\text{Se}_2^-$  are oriented along the 4-fold axis of the zeolite. The sodalite cage space in this direction is slightly larger than that in the 3-fold axis one implying less compression of the anion. The cation sites in LTA-Se(17) are, probably, same as those in dehydrated LTA mainly in the windows between the large cavity and sodalite cage [27] and a few other sites out of the cage leaving more space for  $\text{Se}_2^-$  in it than that in SOD-Se.

In CAN-Se, the cations located near the channel walls make nearly no impact on  $\text{Se}_2^-$  aligned in the center of the channel along its direction (Fig. 5(d)). As pointed out in Ref. [20], the stress around  $\text{Se}_2^-$  does not allow growing large SOD-Se crystals with a significant concentration of the anions. Contrary, it is not an obstacle in LTA, which is beneficial for LTA-Se(17) as a light-emitting material. Thus, the anion compression depends on several parameters, namely: the pore size, the anion orientation and the cation influence. The anomalous temperature dependence of its Raman shift is an important scientific discovery related to the adsorbed material properties.

## 5. Conclusions

Using Se vapour adsorption at  $\sim 550^\circ\text{C}$ , we succeeded in encapsulation of  $\sim 17$  Se atoms per unit cell of the LTA zeolite single crystals. By means of the polarized Raman spectroscopy of so-prepared LTA-Se(17), we demonstrate formation of mainly double- $\text{Se}_8$ -ring clusters with a minor quantity of single  $\text{Se}_{12}$  rings in the large LTA cavities.  $\sim 80\%$  of the large cavities are filled with  $\text{Se}_8$  while the rest  $\sim 20\%$  with  $\text{Se}_{12}$ . Despite quite long history of studies of LTA-Se fabricated by different methods (high-pressure Se melt injection [28] and vapour adsorption at different temperatures [12–15,29–33]), this work provides the first clear evidence for the double- $\text{Se}_8$ -ring-cluster formation. Slightly compressed luminescent  $\text{Se}_2^-$  anions, oriented along the LTA 4-fold axes and located in the sodalite cages, are detected via Raman polarization/orientation dependencies of LTA-Se(17), the compression being partially relaxed with a decrease in temperature producing an anomalous  $\text{Se}_2^-$  Raman downshift. Our conclusion about the formation of Se rings in the LTA large cavities and diatomic Se species in the sodalite cages is, basically, in agreement with the LTA-Se 85-micron-size single crystal XRD data [33]. Bright  $\text{Se}_2^-$  light emission with a maximum at  $\sim 1.56$  eV and vibronic structure is observed in the 1.3–1.8 eV spectral range. The compression of  $\text{Se}_2^-$  in LTA is weaker/stronger than that in sodalite/cancrinite, luminescence band photon energy depending on its strength. High concentration of regularly arranged  $\text{Se}_2^-$  emission centers suggests considering LTA-Se(17) as an important novel light-emitting material. Moreover, the obtained LTA-Se(17) crystal could be an extremely interesting system for testing high-spatial-resolution luminescence, Raman and even Brillouin imaging techniques.

## CRedit authorship contribution statement

**Vladimir V. Poborchii:** Writing – review & editing, Writing – original draft, Visualization, Validation, Supervision, Resources, Project administration, Methodology, Investigation, Formal analysis, Data curation, Conceptualization. **Vitalii P. Petranovskii:** Validation, Resources, Methodology, Conceptualization. **Igor A. Glukhov:** Visualization, Software, Formal analysis. **Andrei A. Fotiadi:** Validation, Methodology, Funding acquisition, Formal analysis.

## Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Data availability

Data will be made available on request.

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