

Spinel type MCo_2O_4 ($\text{M} = \text{Mn}, \text{Mg}, \text{Ni}, \text{Cu}, \text{Fe}$ and Zn) for chemoresistance gas sensors

Zichen Zheng^a, Kewei Liu^{a,b}, Yiwen Zhou^a, Zicong Zhang^a, Hongyuan Su^a, Xudong Nie^a, Marc Debliquy^b, Zexin Yu^c, Chao Zhang^{a,*}

^a College of Mechanical Engineering, Yangzhou University, Yangzhou, 225127, PR China

^b Service de Science des Matériaux, Faculté Polytechnique, Université de Mons, Mons, 7000, Belgium

^c Institute for Manufacturing Technologies of Ceramic Components and Composites, University of Stuttgart, Stuttgart, 70569, Germany

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ABSTRACT

Gas sensors present significant research value and broad applicability for environmental monitoring, medical diagnosis and agricultural production. As a p-type spinel ternary semiconductor metal oxide (SMOX), MCo_2O_4 ($\text{M} = \text{Mn}, \text{Mg}, \text{Ni}, \text{Cu}, \text{Fe}$ and Zn) chemoresistance gas sensors possess a satisfactory sensing performance to diverse hazardous gases. Owing to the superior potential and widespread applications, this work provides a critical review of the current development for MCo_2O_4 chemoresistance gas sensors. Basic information of MCo_2O_4 and evaluation criteria of corresponding gas sensors were described primarily. Then the synthesis, morphology, characterization and sensing properties of MCo_2O_4 gas sensors were elaborated on the basis of different microtopography dimensions under zero dimension (0D), one dimension (1D), two dimension (2D) and three dimension (3D). Various efficient tactics for improving sensing performance and relevant transducing mechanism were demonstrated as well. Finally, perspectives on developing MCo_2O_4 synthesis and applications in gas sensors were elaborated.

1. Introduction

In modern society, toxic, flammable and explosive gases are serious sources of air pollution all along, which have done great harm to environment, human and other living creatures. Thus, the gas sensor technology which possesses lower detection limits than human respiratory system has been attracted great attention in scientific, agricultural, industrial and medical fields [1–9]. Specifically, gas sensor, which is constitutive of a transducer and an active layer can be generally categorized under chemical detector branch, while the desired experimental data of which can be acquired by converting adequate chemical reaction messages into measurable electronic signals, including variation in resistance, frequency, current or voltage [10,11]. Moreover, the capacity of gas sensors can be assessed by diverse indexes, e.g. sensitivity, limit of detection (LOD), selectivity, repeatability, operating temperature, humidity resistance, recovery and response time [12,13]. In order to optimize gas sensor performance, a series of strategies have been widely used such as heterojunction construction [14], morphological regulation [15], catalyst doping and decorating [16], pH regulation [17]

and utilization of metal-organic frameworks (MOF) [18]. Among diverse materials, the ternary semiconductor metal oxide (SMOX), including CoFe_2O_4 [19], ZnSnO_3 [20], ZnWO_4 [21] and NiFe_2O_4 [22] provided a strategy to detect different gases in a rapid, economical, convenient and reliable way compared with some large scale equipment such as gas chromatography (GC) and ion mobility spectrometry (IMS), which have been received great attention [23–25]. Due to the prominent sensing properties, SMOX gas sensors have been extensively utilized in health care [26–28], industrial design [29,30], agricultural engineering [31], daily life [32] and so forth [25]. The sensitivity of SMOX gas sensors is desirable, as poor cross-selectivity and high working temperature are still two major problems under practical application scenarios. However, spinel metal oxides which own special chemical component and structure gradually entered into researchers' field of vision. The transition metal or post-transition metal cations hold the two diverse cation sites while the cations with different chemical properties and charge states are arranged between the cations and the ambient lattice oxygen at disparate bonding energies. The combination of abundant transition metals and post-transition metal cations in spinel metal oxides opens up

* Corresponding author. College of Mechanical Engineering Yangzhou University Huayang West Road 196 Yangzhou 225127, Jiangsu Province PR China.
E-mail address: zhangc@yzu.edu.cn (C. Zhang).

new opportunities for enhancing many features of sensitivity, selectivity and reproducibility [33,34]. Particularly, $M\text{Co}_2\text{O}_4$ ($M = \text{Mn}, \text{Mg}, \text{Ni}, \text{Cu}, \text{Fe}$ and Zn) (The following non-essential are referred to as $M\text{Co}_2\text{O}_4$ commonly) with molecular formula AB_2O_4 of two transition metals is a p-type spinel metal oxides transformed from Co_3O_4 , where M^{2+} are located in one-eighth of the tetrahedral position while Co^{3+} occupied in a half of the octahedral position of the face-centered cubic oxygen lattice, which possesses the merits of high conductivity, excellent electrochemical properties, robust structural stability and low synthesis cost [35,36], is extensively applied in gas sensor [37], lithium battery [38], electrocatalyst [39] and supercapacitor [40], as shown in Fig. 1a and listed in Table 1. As a result, many researchers have published reviews based on $M\text{Co}_2\text{O}_4$ materials. Cheng et al. summarized the latest researches of core-shell structured NiCo_2O_4 based electrochemical capacitors, and substantiated that 1D NiCo_2O_4 prepared by electrospinning could be used as the matrix to construct diverse novel materials to improve the performance of capacitors [41]. Because of the rapid development of supercapacitor devices and batteries, Goncalves et al. reviewed the application of low-budget and vast multi-functional

MnCo_2O_4 materials [42].

Furthermore, as a typical spinel oxide and p-type semiconductor which owns suitable chemical structure and composition, $M\text{Co}_2\text{O}_4$ based gas sensors has strong response and excellent selectivity at low operating temperature for massive gases [63,64]. When detecting the same gas, the response value of p-type SMOX gas sensor is equivalent to the square root of which to common n-type SMOX gas sensor under the same testing circumstances, thus the n-type SMOX are preferred as gas sensor materials chiefly [65–67]. However, chemoresistance p-type $M\text{Co}_2\text{O}_4$ gas sensors with excellent property are still effective in monitoring diverse target gas and applying in many other fields, which has great prospects for fabricating high performance gas sensors which is displayed in Fig. 1b.

Despite there are abundant reviews about diverse SMOX based gas sensors, reviews on the chemoresistance $M\text{Co}_2\text{O}_4$ gas sensors are still lacking. Therefore, this review demonstrates the status of $M\text{Co}_2\text{O}_4$ and their expandable materials oriented towards the development of SMOX gas sensors with exquisite gas sensitive mechanism analysis. In detail, we focused on the morphology control of $M\text{Co}_2\text{O}_4$ microscopic

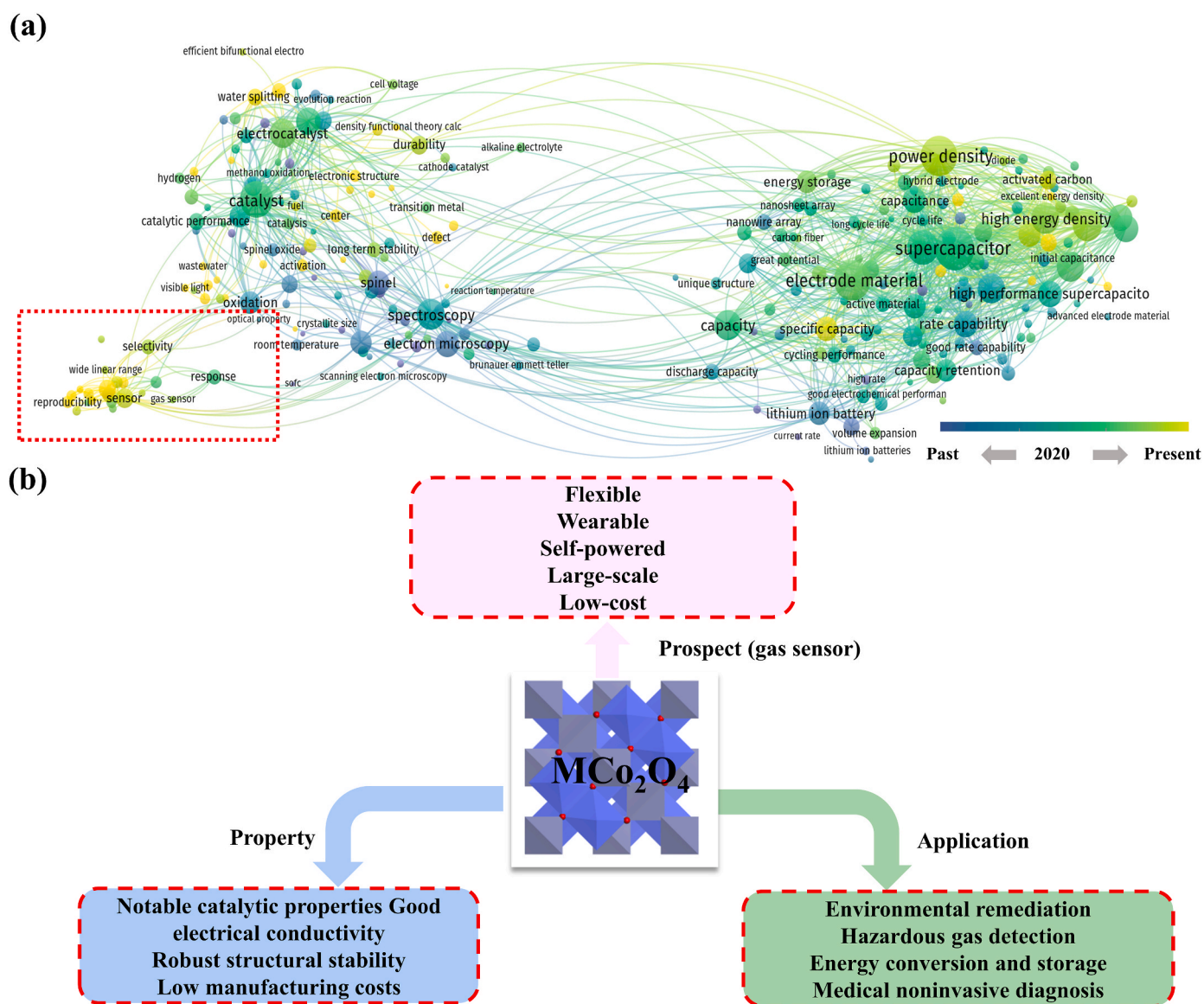


Fig. 1. (a) Visualization of current trends in research on $M\text{Co}_2\text{O}_4$, along with co-occurring keywords. Lines connecting two keywords, size of circle of each keyword, and color of circle indicate the relevance of the keywords, the frequency of keywords appearance, and average publication date, respectively. Data were collected from web of science. (b) The property, application and prospect of gas sensor based on $M\text{Co}_2\text{O}_4$.

Table 1The application in other fields based on MCo_2O_4 .

Dimension	Composites	Microstructures	Growth reagents	Synthesis method	Application	Ref.
3D	$\text{CeO}_2/\text{MnCo}_2\text{O}_4$	Microspheres	$\text{Mn}_4\text{H}_6\text{O}_4 \cdot 4\text{H}_2\text{O}$, $\text{C}_4\text{H}_6\text{CoO}_4 \cdot 4\text{H}_2\text{O}$, KMnO_4 , $\text{Ce}(\text{NO}_3)_3 \cdot 4\text{H}_2\text{O}$	Hydrothermal and gas bubble	Chemical looping combustion	[43]
0D	MgCo_2O_4	Nanoparticles	$\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, NaOH	Hydrothermal	Catalytic degradation	[44]
3D	$\text{O-NiCo}_2\text{O}_4$	Octahedral nanoblocks	Ethanol, ammonium hydroxide, Na_2CO_3 , $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, ethylene glycol, NaOH	Solvothermal	Catalytic degradation	[45]
1D	$\text{NiCo}_2\text{O}_4/\text{NC}$	Nanowires	$\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, DMF , PTA , NaOH , Ni foam	Hydrothermal/In-situ growth	Energy storage and conversion	[46]
1D	$\text{NiCo}_2\text{O}_4@\text{CC}$	Nanowires	$\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, 68 % wt. HNO_3 , CC	Hydrothermal	Energy storage and catalytic degradation	[47]
3D	$\text{NiCo}_2\text{O}_4@\text{RuO}_2$	Nanocones	$\text{RuCl}_3 \cdot x\text{H}_2\text{O}$, $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$, $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$, HMT , SDS , PSS , ethanol, KOH	Hydrothermal	Overall water splitting.	[48]
1D	$\text{In}@\text{NiCo}_2\text{O}_4$	Nanoneedles	$\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{In}(\text{NO}_3)_3 \cdot 4\text{H}_2\text{O}$, Carbon fiber	Pulsed laser ablation	Energy storage and conversion	[49]
2D	$\text{Co}_3\text{O}_4/\text{FeCo}_2\text{O}_4$	Hierarchical nanosheets	Foamed nickel, HCl , NH_4F , $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, CON_2H_4 , $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$, $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$, $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$	Hydrothermal	Energy storage and conversion	[50]
2D	FeCo_2O_4	Macaroon-like structure	Glucose, KOH , $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, NH_4F , PTFE hexamethylenetetramine, AC powder (YEC-8A)	Solvothermal	Energy storage and catalytic degradation	[51]
3D	$\text{Ce}@\text{ZnCo}_2\text{O}_4$	Nanospheres	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$, glycerol, isopropanol	Solvothermal	Glutathione detection	[52]
2D	ZnCo_2O_4	Sheet-shaped	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, isopropanol, ethanol	Solvothermal	Electromagnetic wave absorption	[53]
3D	$\text{Ce}@\text{ZnCo}_2\text{O}_4$	Core-shell microspheres	$\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$, $\text{Co}(\text{CH}_3\text{COO})_2 \cdot 4\text{H}_2\text{O}$, ethylene Glycol, $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$, hexamethylenetetramine	Reflux	Catalytic propane decomposition	[54]
3D	$\text{ZnO}/\text{ZnCo}_2\text{O}_4$	Spherical particles	$\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$, $\text{Co}(\text{CH}_3\text{COO})_2 \cdot 4\text{H}_2\text{O}$, glycyrrhiza glabra extract aqueous solution	Hydrothermal	Bisphenol A decomposition	[55]
1D	ZnCo_2O_4	Nanowire arrays	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{CO}(\text{NH}_2)_2$, nickel foam	Hydrothermal	Non-enzymatic glucose detection	[56]
3D	$\text{ZnCo}_2\text{O}_4/\text{C}$	Micro-hydrangeas	ZnCl_2 , $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, PVP , $\text{Na}_3\text{C}_6\text{H}_5\text{O}_7$, ethanol, ethylene glycol	Solvothermal	Lithium ions batteries	[57]
3D	$\text{ZnM}_x\text{Co}_{2-x}\text{O}_4$ ($\text{M} = \text{Mn}$)	Agglomerated nanoblocks	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$	Sol-gel combustion	H_2O_2 determination	[58]
3D	$\text{Pt}@\text{ZnCo}_2\text{O}_4$	Nanoparticles decorated on the microflowers	H_2PtCl_6 , NaBH_4 , PVP , phthalic acid, $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, acetone, DMF	Hydrothermal	Electrochemical Caffeine detection	[59]
2D	ZnCo_2O_4	Nanoplate arrays	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, 2-methylimidazole	Static synthesis	Dihydroxy benzene isomers detection	[60]
3D	ZnCo_2O_4	Spherical shape-like microparticles	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, citric acid, KOH	Sol-gel method	Energy storage and conversion	[61]
3D	$\text{ZnO}/\text{ZnCo}_2\text{O}_4$	Microflowers	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, isopropanol, ethanol, $\text{D}(+)\text{-Glucose}$	Hydrothermal/Solvothermal	Electromagnetic wave absorption	[62]
2D	ZnCo_2O_4	Wrinkled-paper-like nanoflakes	$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, PVP , HMT	Hydrothermal	Energy storage and conversion	[38]

Abbreviation: $\text{O-NiCo}_2\text{O}_4$: Octahedral NiCo_2O_4 ; NC : Porous Ni/C substrate; CC : carbon cloth; SDS : sodium dodecyl sulfate; PSS : polystyrene sulfonic acid sodium salt; Triblock copolymer pluronic F127: $(\text{HO}(\text{CH}_2\text{CH}_2\text{O})_{106}(\text{CH}_2\text{CH}(\text{CH}_3)\text{O})_{70}(\text{CH}_2\text{CH}_2\text{O})_{106}\text{H})$; PTFE : polytetrafluoroethylene; PrGO : porous reduced graphene oxide; PTA : Terephthalic acid; NC-CNT : nitrogen-rich carbon nanotube; PVP : polyvinyl pyrrolidone; DMF : $\text{N,N-Dimethylformamide}$; HMT : hexamethylene-tetramine.

materials, reviewed the performance of related gas sensors and summarized the related preparation process, gas-sensitive enhancement mechanism and important parameters of gas sensors based on the literature of the last five years as shown in Scheme 1. Furthermore, the test system and sensing parameters of gas sensors are introduced in section 2. Different synthesis strategies were used to fabricate MCo_2O_4 with various morphologies in diverse dimensions (0D, 1D, 2D, and 3D), which is summarized in Tables 2 and 3. Diverse effective tactics have been adopted to strengthen sensing performance and the concrete sensing results is displayed in Table 4. This is expected to set up a framework of MCo_2O_4 spinel oxide gas sensors and emphasize the significance of morphology control in novel material synthesis, further improving the logic and rationality of resistive gas sensor design.

2. Test system and basic gas sensing performance parameters

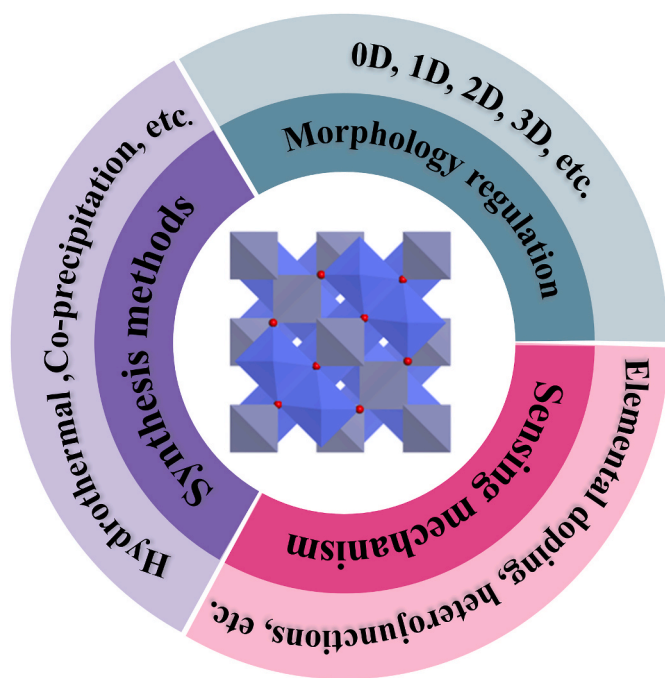
2.1. Test system

Generally, MCo_2O_4 based SMOX gas sensor system belongs to static system, comprising the valve system, flow control device, gas sensors and chamber, power supply and signal processing system [96,144]. In a typical testing process, the synthetic sample with an appropriate amount

of deionized water will be ground evenly in a grinding bowl. Then the as-prepared slurry is uniformly applied to the sensor substrate. The corresponding sensor is obtained by aging under the specific condition. Next, the sensor will be installed in a test chamber in preparation for test. Under specific operating conditions, high-purity air is introduced to obtain a steady resistance value and the tested gas is injected for gas sensing test to obtain experimental data. After the resistance value has stabilized, high purity air is introduced again to restore the resistance value to its initial state. Above all, the entire testing circumstance is an uncontaminated chamber with consistent relative humidity (RH) and operating temperature while the raw sensing data can be acquired and the crucial sensing parameters can be evaluated. In addition, the sensor can be designed as a ceramic on-tube structure or be fabricated on an aluminum oxide substrate.

2.2. Basic gas sensing performance parameters

In most instances, optimum working temperature, gas response, response and recovery time, selectivity, LOD, long-term stability, reproducibility and moisture resistance properties can be applied to evaluate sensing performance of MCo_2O_4 based gas sensors [143,145].



Scheme 1. Diagram of the synthesis strategies and sensing mechanism of $M\text{Co}_2\text{O}_4$ sensors.

Table 2
Summary of the design and synthesis approaches.

Synthesis approaches	Advantages	Disadvantages	Ref.
Hydrothermal and solvothermal methods	Facile preparation, broad resource of raw materials, high crystallinity of the samples, excellent morphology control	High temperature and high pressure, high equipment requirements, long reaction time	[68–70]
Co-precipitation	Low energy consumption, low temperature of synthesis, control over particle composition, facile and rapid synthesis process, uniform chemical composition, easy access to small particle size products	Reproducibility issues, difference in precipitation rate for certain reactants, precipitation of impurities	[71–73]
Self-sacrificing template	Excellent and precise morphology and size control, obtained products without agglomeration	High cost, relatively complex synthesis process, difficult to prepare products at large scale	[74–76]

2.2.1. Optimum operating temperature

Operating temperature served as a significant role in sensing mechanism whereas it controls chemical interactions between target gas molecules and the $M\text{Co}_2\text{O}_4$ junction [146]. As high working temperature leads to excessive energy consumption, hindering the development of flexible semiconductor gas sensor, sensor miniaturization and IC integration, lowering the long-term stability and safety feature, room temperature or subzero temperature operated gas sensors are the focus of massive researchers [147,148].

2.2.2. Gas response

Gas response, also known as sensitivity, is the ratio of gas sensor signal output to target gas concentration, reflecting the intensity of the

physical and chemical reaction between the target gas molecules and sensing material, which can be illustrated by several patterns as discussed below [149,150].

In detail, the resistance value of SMOX sensor reaches steady state in the target gas and in high purity air circumstances can be abbreviated as R_g and R_a respectively. Thus, the response is defined as R_a/R_g under oxidized gas detection circumstances and R_g/R_a can be used to calculate for reducing gas detection circumstances. Furthermore, the response is also demonstrated as the rate of resistance variation during sensing process which can be defined as $\Delta R/R_a$. The superior sensing performance is usually accompanied by a high response value [151].

2.2.3. Response and recovery time

Response time is when electrode resistance changed 90 % during target gas adsorption process while the recovery time represents which changed 90 % during target gas desorption process, through which the adsorption and desorption rate of target gas molecules onto the surface of sensing material could be reflected [152,153].

2.2.4. Selectivity

The selectivity demonstrates the ability of sensors to resist the influence of interfered gas while the strong sensitivity to target gas and obviously weak sensitivity to other interfered gas (including water molecules) are what we pursue. Based on the working principle of chemically resistive SMOX gas sensor, cross-selectivity is inevitable. Thus, electronic nose based on SMOX gas sensors combined with pattern recognition algorithm and Eley-Rideal model on account of kinetic parameters during gas sensing can effectively improve the selectivity under the situation of multi-gas detection [31,154,155].

2.2.5. Limit of detection

LOD refers to lowest detectable concentration of the target gas which can be monitored in the background signal that is three times to noise. The optimization of the lower detection limit is highly essential in some special areas such as microenvironmental monitoring and medical diagnosis. The minimum detection limit is expressly crucial to expand application of gas sensors while the ppb and ppt level gas sensors are still lacking [156].

2.2.6. Stability

The stability of gas sensor can be divided into two parts: reproductivity and long-term stability. In specific, the excellent reproductivity of sensor represents small fluctuations in response of several common dynamic response-recovery cycles. The long-term stability is an index for evaluating sensor sensitivity fluctuation for a relative long period of ≥ 15 days. Stability affects the reliability of gas sensing data and the service life of sensor, which plays a decisive role in the practical application [157].

2.2.7. Moisture resistance

In general, water molecules in the environment are similar to other gas molecules which will adsorb on the sensing material and undergo physical and chemical reactions with oxygen species, leading to changes of reference resistance and gas response [158]. In particular, water molecules will contest adsorption site with target gas molecules and thus reducing sensitivity in most instances in the process of gas sensing. Therefore, excellent moisture resistance is the main performance of high-quality SMOX gas sensors at present.

2.2.8. Gas flow rate

Imprecise gas flow regulation will greatly affect the response value of the gas sensor. On the one hand, in the process of gas dynamic detection, the relationship between sensitivity and target gas concentration can be revealed through statistical analysis to eliminate the impact of gas flow error on detection results [159]. On the other hand, by constructing a more stable gas sensitive matrix and sensing film, the influence of large

Table 3
MCo₂O₄ microstructures: morphologies, methods of preparation and other growth parameters.

Dimension	Composites	Microstructures	Growth reagents	Synthesis method	Anneal temperature (°C)	Anneal time	Ref.
0D	MnCo ₂ O ₄ /Polypyrrole	Nanoparticles	Mn(NO ₃) ₂ ·4H ₂ O, CoCl ₂ ·6H ₂ O, NaOH, DMF	In situ chemical polymerization	500 °C	6 h	[77]
	MnCo ₂ O ₄	Spherical shape nanoparticles	Mn(NO ₃) ₂ ·4H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, ammonium hydroxide, urea	Hydrothermal	–	–	[78]
	MgCo ₂ O ₄	Nanoparticles	Co(NO ₃) ₂ ·6H ₂ O, Mg(NO ₃) ₂ ·6H ₂ O, AOT, ethylene glycol	Microwave-assisted colloidal	400, 500, 600, 700, 800 °C	5 h	[79]
	MoS ₂ /NiCo ₂ O ₄	Nanoparticles	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, MoS ₂ , NaOH	Hydrothermal	–	–	[37]
	Ni _x Co _{3-x} O ₄	Nanoparticles	Co(II) and Ni(II) oxalates, ammonium oxalate, H ₂ C ₂ O ₄	Co-precipitation/Thermal decomposition	300 °C	24 h	[80]
	g-C ₃ N ₄ /NiCo ₂ O ₄	Spherical nanoparticles	Melamine, Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, NaOH	Hydrothermal	–	–	[81]
	In ₂ O ₃ /YSZ/NiCo ₂ O ₄	Nanoparticles	In(NO ₃) ₃ , glycine, NiCo ₂ O ₄ , ethyl cellulose, ethanol, α-terpineol	Self-propagating high-temperature	900 °C	3 h	[82]
	NiCo ₂ O ₄	Nanoparticles	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea, ammonium hydroxide	Hydrothermal	–	–	[83]
	FeCo ₂ O ₄ /graphene	Nanoparticles	Co(NO ₃) ₂ ·6H ₂ O, Fe(NO ₃) ₂ ·6H ₂ O, urea, NaOH	Hydrothermal	–	–	[84]
	WO ₃ /FeCo ₂ O ₄	Nanoparticles	Na ₂ WO ₄ ·2H ₂ O, NaCl, HCl, FeSO ₄ ·7H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, carbamide, ethylene glycol	Hydrothermal	350 °C	2 h	[85]
	ZnO/ZnCo ₂ O ₄	Nanodots	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, NaCl, ZnO	Hydrothermal/Co-precipitation	600 °C	2 h	[86]
	ZnO/ZnCo ₂ O ₄	Nanoparticles	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, PVP, DMF	Hydrothermal	450 °C	3 h	[87]
	ZnCo ₂ O ₄	Nanoparticles	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, K ₂ CO ₃	Co-precipitation/digestion	600 °C	5 h	[88]
	ZnCo ₂ O ₄ /Polypyrrole	Nanoparticles	Zn(NO ₃) ₂ ·H ₂ O, CoCl ₂ ·6H ₂ O, NaOH, DMF	In situ chemical polymerization	100 °C	5 h	[77]
	1D	Co ₃ O ₄ /NiCo ₂ O ₄ /CC	Nanowires	CC, HCl, acetone, CTAB, Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea, 2-methylimidazole	Hydrothermal	350 °C	2 h
NiCo ₂ O ₄ /rGO		Nanowires	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea, ethanol, GO	Hydrothermal	300 °C	2 h	[90]
Pd/NiCo ₂ O ₄		Nanoneedles	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea, Pd thin film	Thin film deposition/ Hydrothermal	350 °C	2 h	[91]
NiCo ₂ O ₄ /rGO		Nanorods	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, CTAB	Hydrothermal	350 °C	3 h	[92]
NiCo ₂ O ₄		Nanorods	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, ammonium fluoride, urea	Hydrothermal	600 °C	5 h	[93]
NiCo ₂ O ₄		Hierarchical hollow microtubules	Na ₂ WO ₄ ·H ₂ O, AgNO ₃ , Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, PVP, urea, HNO ₃ , ammonium hydroxide	Template-assisted process/Etching treatment	300 °C	2 h	[94]
NiCo ₂ O ₄ /SnO ₂		Nanofiber	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, ethanol, DMF, PVP, SnCl ₂ ·2H ₂ O, NaOH	Electrospinning/ Hydrothermal	400 °C	3 h	[95]
Zn@NiCo ₂ O ₄		Mesoporous rods	Ni(SO ₄) ₂ ·7H ₂ O, Zn(SO ₄) ₂ ·7H ₂ O, CoSO ₄ ·7H ₂ O, oxalic acid, ethylene glycol	Hydrothermal	450 °C	2 h	[96]
CuO/CuCo ₂ O ₄		Nanotubes	Co(NO ₃) ₂ ·6H ₂ O, CuCl ₂ ·2H ₂ O, PVP, DMF, ethanol	Electrospinning	350 °C	3 h	[97]
TiO ₂ /ZnCo ₂ O ₄		Porous nanorods	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, ammonium acid carbonate, Titanium (IV) oxyacetylacetonate	Dual-oxalate sacrificial template	500 °C	2 h	[98]
ZnCo ₂ O ₄		Porous nanorods	Zn(NO ₃) ₂ ·6H ₂ O, Co(CH ₃ COO) ₂ ·4H ₂ O, ethanol, oxalic acid	Co-precipitation	350 °C	2 h	[99]
ZnCo ₂ O ₄		Single-layer nanochain			500 °C	2 h	
ZnCo ₂ O ₄		Microtubes	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, absorbent cotton	Biomorphic template	500, 600, 700 °C	2 h	[100]
ZnO/ZnCo ₂ O ₄		Nanotubes	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, DMF, ethanol, PVP	Capillary electrospinning	450 °C	100 min	[101]
ZnO/ZnCo ₂ O ₄		Tube in tube nanostructure	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, DMF, ethanol, PVP	Capillary electrospinning	450 °C	100 min	[102]
Zn _x Co _{2-x} O ₄	Nanorods	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, K ₂ CO ₃	Co-precipitation	500 °C	2 h	[103]	
2D	MgCo ₂ O ₄	Nanosheets	Co(NO ₃) ₂ ·6H ₂ O, Mg(NO ₃) ₂ ·6H ₂ O, urea, NH ₄ OH	Hydrothermal	–	–	[104]
	NiCo ₂ O ₄ /WO ₃	Nanosheets	Na ₂ WO ₄ ·2H ₂ O, HCl, NaCl, Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea	Hydrothermal/Co-precipitation	350 °C	2 h	[105]
	ZnO/NiCo ₂ O ₄	Nanosheets with nanofibers	Zn(NO ₃) ₂ ·6H ₂ O, Na ₂ CO ₃ , SDSN, Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea	Hydrothermal	350 °C	2 h	[106]
	NiCo ₂ O ₄ /MWCNTs	Nanoflakes	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, MWCNTs, HNO ₃	Hydrothermal	450 °C	3 h	[107]
	NiCo ₂ O ₄	Nanoflakes	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, ammonium hydroxide	Co-precipitation	450 °C	4 h	[108]

(continued on next page)

Table 3 (continued)

Dimension	Composites	Microstructures	Growth reagents	Synthesis method	Anneal temperature (°C)	Anneal time	Ref.
3D	NiCo ₂ O ₄	Nanoplates	Poly ethylene glycol-6000, Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O	Hydrothermal	300, 500, 700 °C	3 h	[109]
	NiCo ₂ O ₄ /α-MoO ₃	Porous nanosheets	Mo powder, H ₂ O ₂ , Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea	Hydrothermal/Co-precipitation	350 °C	2 h	[110]
	NiCo ₂ O ₄	Nanosheets	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, Sodium dodecyl sulfate, NaOH	Hydrothermal	–	–	[111]
	NiCo ₂ O ₄ /rGO	Nanoplates	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, CTAB	Hydrothermal	400 °C	2 h	[112]
	CuCo ₂ O ₄	Nanosheets	Cu(NO ₃) ₂ ·3H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, citric acid	Solution combustion method	–	–	[113]
	CuCo ₂ O ₄	Nanoplatelets	Cu(NO ₃) ₂ ·3H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, urea	Hydrothermal	350 °C	2 h	[114]
	MoS ₂ -ZnCo ₂ O ₄ -ZnCo ₂ O ₄ /CC	Nanosheets	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, CO(NH ₂) ₂ , (NH ₄)F, Na ₂ MoO ₄ ·6H ₂ O, CH ₄ N ₂ S	Hydrothermal	200 °C	5 h	[115]
	ZnCo ₂ O ₄	Porous nanosheets	Graphene sheets, ethanol, ammonia solution, Co(CH ₃ COO) ₂ ·4H ₂ O, Zn(CH ₃ COO) ₂ ·2H ₂ O	Hydrothermal	450 °C	3 h	[116]
	ZnCo ₂ O ₄	Nanosheet arrays	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, CO(NH ₂) ₂ , (NH ₄)F	Hydrothermal	400 °C	2 h	[117]
	Co ₃ O ₄ /NiCo ₂ O ₄	Nanocages	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, 2-Methylimidazole, methanol	Hydrothermal/Co-precipitation	550 °C	2 h	[118]
	NiO/NiCo ₂ O ₄	Hierarchical microspheres	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, ethanol	Hydrothermal	500 °C	3 h	[119]
	NiCo ₂ O ₄	Nanospheres	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, urea	Hydrothermal	350 °C	2 h	[120]
	PdO-NiO/NiCo ₂ O ₄	Truncated Nanocages	Ni(NO ₃) ₂ ·6H ₂ O, Na ₃ C ₆ H ₅ O ₇ ·2H ₂ O, K ₃ [Co(CN) ₆], PdCl ₂ , methanol, ammonia solution	Co-precipitation/etching/wet impregnation	350 °C	2 h	[121]
	NiO/NiCo ₂ O ₄	Hollow microspheres	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, H ₃ BTC, PVP, DMF	Self-sacrificing template	450 °C	2 h	[76]
	Co ₃ O ₄ /NiCo ₂ O ₄	Nanocages	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, 2-Methylimidazole, methanol	Co-precipitation	450 °C	2 h	[122]
	NiCo ₂ O ₄ /WO ₃	Nanoflowers	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, HCl, Na ₂ WO ₄ ·2H ₂ O	Hydrothermal	350 °C	2 h	[123]
	NiCo ₂ O ₄	Hollow microspheres	CoCl ₂ ·6H ₂ O, NiCl ₂ ·6H ₂ O, urea	Hydrothermal	300, 400, 500 °C	3 h	[124]
	NiCo ₂ O ₄	Hierarchical microflowers	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, 2-Methylimidazole, ethanol	Hydrothermal	400 °C	2 h	[125]
	NiCo ₂ O ₄	Double-shelled hollow spheres	Ni(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, xylose, isopropanol	Self-template	450 °C	2 h	[126]
	PANI/NiCo ₂ O ₄	Nanospheres	Polyimide film, Co(CH ₃ COO) ₂ ·6H ₂ O, gelatin granules, Ni(CH ₃ COO) ₂ ·4H ₂ O, NH ₄ HCO ₃ , ethylene glycol, ammonium persulphate, aniline, HCl	Hydrothermal	350 °C	3 h	[127]
	Cu _{0.3} Co _{2.7} O ₄	Hollow microspheres	Co(CH ₃ COO) ₂ ·6H ₂ O, CuSO ₄ ·5H ₂ O, NH ₃ ·H ₂ O, n-propanol	Hydrothermal	–	–	[128]
	ZnO/ZnCo ₂ O ₄	Hexahedral-structure	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, PVP; K30	Hydrothermal	600 °C	4 h	[129]
	ZnCo ₂ O ₄	Hierarchical porous architectures	Co(CH ₃ COO) ₂ ·4H ₂ O, Zn(CH ₃ COO) ₂ ·2H ₂ O, CO(NH ₂) ₂ , ethanol	Hydrothermal	300 °C	10 min	[130]
	Co ₃ O ₄ /ZnCo ₂ O ₄	Hollow nanostructures	2-methylimidazole, methanol, Co(NO ₃) ₂ ·6H ₂ O, ethanol, Zn(NO ₃) ₂ ·6H ₂ O	Self-sacrificing template	450 °C	2 h	[131]
	Ag-ZnCo ₂ O ₄	Hollow microspheres	Zn(CH ₃ COO) ₂ ·2H ₂ O, Co(CH ₃ COO) ₂ ·4H ₂ O, AgNO ₃ , NaOH, polyethylene glycol 1000, ethanol	Solvent-thermal	500 °C	3 h	[132]
	ZnCo ₂ O ₄	Hierarchical microflowers	2-methylimidazole, methanol, Co(NO ₃) ₂ ·6H ₂ O, Zn(NO ₃) ₂ ·6H ₂ O	Solvothermal	405 °C	55 min	[133]
	ZnCo ₂ O ₄	Yolk-shell spheres	Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, D-Glucose, DMF, CO(NH ₂) ₂ , PVP	Hydrothermal	450 °C	2 h	[134]
	Ag@ZnCo ₂ O ₄	Hollow spheres	2-methylimidazole, methanol, Co(NO ₃) ₂ ·6H ₂ O, ethanol, Zn(NO ₃) ₂ ·6H ₂ O, glucose, NaBH ₄ , AgNO ₃	Self-sacrificing template	400 °C	1 h	[135]
ZnCo ₂ O ₄	Yolk-shell microspheres	Zn(CH ₃ COO) ₂ ·2H ₂ O, Co(CH ₃ COO) ₂ ·4H ₂ O, ethylene glycol	Co-precipitation	350 °C	5 h	[136]	
ZnO/ZnCo ₂ O ₄	Hierarchical structure	NH ₄ HCO ₃ , Zn(CH ₃ COO) ₂ ·2H ₂ O, Co(CH ₃ COO) ₂ ·4H ₂ O	Hydrothermal	500 °C	2 h	[137]	
Pd-ZnO/ZnCo ₂ O ₄	Hollow spheres	K ₂ PdCl ₄ , Zn(NO ₃) ₂ ·6H ₂ O, Co(NO ₃) ₂ ·6H ₂ O, methanol, ethanol, 2-methylimidazole, PS latex microspheres (1 μm)	MOF templated synthesis	450 °C	1 h	[138]	
PdO-ZnO/ZnCo ₂ O ₄	Microspheres	K ₃ [Co(CN) ₆], Zn(NO ₃) ₂ ·6H ₂ O, HCl, PVP, PdCl ₂ , ethanol	Co-precipitation	700 °C	2 h	[139]	
ZnCo ₂ O ₄	Hollow polyhedral superstructure	2-methylimidazole, methanol, Co(NO ₃) ₂ ·6H ₂ O, Zn(NO ₃) ₂ ·6H ₂ O, ethanol	Co-precipitation	500 °C	2 h	[140]	
ZnCo ₂ O ₄	Multishelled Hollow Twin Spheres	Co(NO ₃) ₂ ·6H ₂ O, Zn(NO ₃) ₂ ·6H ₂ O, xylose	Solvothermal	450 °C	7 h	[36]	

(continued on next page)

Table 3 (continued)

Dimension	Composites	Microstructures	Growth reagents	Synthesis method	Anneal temperature (°C)	Anneal time	Ref.
	ZnO/ZnCo ₂ O ₄	Hierarchical macroporous honeycomb-like	Co(NO ₃) ₂ ·6H ₂ O, Zn(NO ₃) ₂ ·6H ₂ O, glycerol, 2-propanol, ethanol	Self-template	350 °C	2 h	[141]
	Zn _x Co _{3-x} O ₄	Nanocages	2-methylimidazole, Co(NO ₃) ₂ ·6H ₂ O, CTAB, Zn(NO ₃) ₂ ·6H ₂ O	Co-precipitation	350 °C	2 h	[142]
	ZnCo ₂ O ₄	Cabbage-shaped	Co(NO ₃) ₂ ·6H ₂ O, Zn(NO ₃) ₂ ·6H ₂ O, <i>m</i> -phthalic acid, acetone, DMF	Co-precipitation	500 °C	10 min	[143]

Abbreviation: AOT: Dioctyl sulfosuccinate sodium salt; YSZ: Yttria stabilized zirconia; rGO: Reduced graphene oxide; GO: Graphene oxide; CTAB: Cetyl trimethyl ammonium bromide; MWCNTs: multi-walled carbon nanotubes; CC: Carbon cloth.

or small gas flow rate on the response value can be eliminated from the root.

3. MCo₂O₄ based gas sensors: synthesis, growth, structure and characterization

In an upcoming section, we summarized the common synthesis strategies and controllable methods to fulfill featured morphologies on the basis of dimensionality of MCo₂O₄ gas sensors in Tables 2 and 3. Yet, for further enhance the sensing properties of MCo₂O₄ for practical applications, studies on sensing materials with unique morphologies and microstructures are still in progress.

3.1. Synthesis of MCo₂O₄ semiconductor metal oxides

At present, there are many methods for the synthesis of MCo₂O₄ SMOX, which can be divided into two categories: gas phase method and liquid phase method on the basis of the preparation system, of which the latter accounts for a larger proportion. In addition, liquid phase methods can be divided into hydrothermal/solvothermal method, co-precipitation method and self-sacrificing template method.

3.1.1. Hydrothermal and solvothermal methods

Hydrothermal and solvothermal methods are typical wet chemistry methods which mainly study the chemical behavior and law of substances in solution under relatively high pressure and high temperature [160]. By means of simply changing the hydrothermal/solvothermal reaction parameters, such as temperature, concentration, pH value, solvent and surfactant, the degree of crystallization, size, morphology and substance equality of the product can be controlled [161,162]. Collectively, our previous summary had shown that the primary wet chemical methods used to prepare MCo₂O₄ are hydrothermal and solvothermal methods.

3.1.2. Co-precipitation

Co-precipitation method can be adopted to prepare solid products by means of the action of precipitating agent and more than two kinds of metal salt solution in the form of hydroxide, carbonate, citrate or oxalate. The final powder is obtained after a heat treatment of calcination if required [71].

3.1.3. Self-sacrificing template

Sacrificial template method is a method to convert template materials into corresponding compounds through displacement reaction or redox reaction by using the vacancy formation principle of mutual diffusion [74,75], which provides a practical method to prepare hollow MCo₂O₄ materials.

3.2. Morphology regulation for MCo₂O₄ preparation

In this section, we will discuss the enhancement for detecting specific gases concentrating on the morphology regulation of MCo₂O₄ materials

in different dimensions (0D, 1D, 2D and 3D). In particular, X dimensional material can be defined as 3-X dimensional scales of which in space was at nanoscale, for example, one dimensional material is identify with that there are two dimensions in space at nanoscale, such as nanofibers and nanowires.

3.2.1. 0-dimensional (0D) MCo₂O₄

Currently, most of the researches of 0D MCo₂O₄ gas sensors are focus on nanoparticles. Tiny nanoparticles usually possess large surface area, which is more beneficial for adsorbing abundant oxygen molecules and frequently contacting with the target gas. However, small nanoparticles are more likely to form giant and dense second-ordered particles as the van der Waals' force is inversely proportion to the size of particles. Therefore, the macro-aggregates will prevent target gas from diffusing to the entire material surface, which will decrease the gas response [163]. Furthermore, undersized nanoparticles will bring out the severe aggregation and sintering at high operating temperatures (250–450 °C) in solid-state gas devices [164], leading to the aging of sensors. Nevertheless, some typical 0D MCo₂O₄ nanomaterials are successfully applied for gas detection. In general, hydrothermal method is the most frequently-used way to obtain the corresponding nanoparticles. For distance, 0D ZnO/ZnCo₂O₄ and pure ZnCo₂O₄ nanoparticles synthesized via a facile hydrothermal way was performed by Liu et al. as shown in Fig. 2a and b. The sensitivity of ZnO/ZnCo₂O₄ sensor (15.6) is around four times than that of pure ZnCo₂O₄ gas sensor (3.7) in 100 ppm ethylene glycol vapor (Fig. 2c–e) while the better anti-interference property of ZnO/ZnCo₂O₄ sensor was achieved as shown in Fig. 2f. The high chemisorbed oxygen content (55 %) on the surface and formation of p-n heterojunction are two primary avails for ethylene glycol sensing [87]. Spherical shaped MnCo₂O₄ nanoparticles which possessed a mean grain size of 35–55 nm with large BET surface area (65.3 m²·g⁻¹) was obtained via urea-assited hydrothermal method without post-calcination process by Vadivel's group and the as-prepared MnCo₂O₄ gas sensor exhibited superior sensing performance towards ethanol and acetone [78]. Akhtar et al. employed one-step hydrothermal method for synthesizing the NiCo₂O₄/MoS₂ nanocomposites. Using sodium hydroxide as acid-base regulator, the round and biscuit-shaped NiCo₂O₄ nanoparticles with a size of around 270–280 nm were modified onto the MoS₂ lamellar structure [37]. Meanwhile, Balaji and co-workers reported that the extremely small size NiCo₂O₄ particles were fabricated via hydrothermal and post-annealing possess. The uniform spherical nanoparticles increased the number of chemisorbed oxygen molecules while adequate BET surface area and rich oxygen vacancies enhanced the sensitivity synergistically [83]. Akhtar et al. applied hydrothermal method to decorate g-C₃N₄ on NiCo₂O₄ spherical nanoparticles for ethanol sensing. In this report, the enhanced sensing response (19.2 @ 100 ppm) and extremely low experimental LOD (10 ppb) are derived from enlarged BET surface area and incremental oxygen species absorbed onto the NiCo₂O₄/g-C₃N₄ surface [81]. As an appropriate substrate for different structured-composites, g-C₃N₄ occupies a homogeneous nitrogen group of the structure, which supplies more coordination sites and increases the quantity of adsorbed

Table 4
Sensing performances of gas sensors based on MCo₂O₄.

Dimension	Composites	Microstructures	Synthesis method	Target gas	Concentration (ppm)	T (°C)	Response	Response/Recovery time	LOD	Ref.
0D	MnCo ₂ O ₄ /Polypyrrole	Nanoparticles	In situ chemical polymerization	Acetone	2.5	RT	^b 1.06	47/100 s	–	[77]
	MnCo ₂ O ₄	Spherical shape nanoparticles	Hydrothermal	Ethanol	500	RT	^e 55	65/45 min	–	[78]
	MgCo ₂ O ₄	Nanoparticles	Microwave-assisted colloidal	Acetone	50	250	^a 17.4	20.1/97.3 s	0.5 ppm	[79]
	MoS ₂ /NiCo ₂ O ₄	Nanoparticles	Hydrothermal	Ethanol	100	170	^a 9	21/10 s	0.1 ppm	[37]
	Ni _x Co _{3-x} O ₄	Nanoparticles	Co-precipitation/Thermal decomposition	CO	20	120	^b 2.5	–	0.66 ppm	[80]
	g-C ₃ N ₄ /NiCo ₂ O ₄	Spherical nanoparticles	Hydrothermal	Ethanol	100	110	^a 19.2	21/11 s	10 ppb	[81]
	In ₂ O ₃ /YSZ/NiCo ₂ O ₄	Nanoparticles	Self-propagating high-temperature	Ammonia	50	500	^e 87.19	–	–	[82]
	NiCo ₂ O ₄	Nanoparticles	Hydrothermal	Acetone	500	RT	^f 72 × 10 ⁻³	20/26 s	–	[83]
	FeCo ₂ O ₄ /graphene	Nanoparticles	Hydrothermal	Acetone	500	RT	^e 46	31/19 s	–	[84]
	WO ₃ /FeCo ₂ O ₄	Nanoparticles	Hydrothermal	Ammonia	100	300	^a 33.4	32.5/20.2 s	–	[85]
	ZnO/ZnCo ₂ O ₄	Nanodots	Hydrothermal/Co-precipitation	Acetone	10	240	^a 15.3	7.7/10.3 s	5 ppm	[86]
	ZnO/ZnCo ₂ O ₄	Nanoparticles	Hydrothermal	Ethylene glycol	100	160	^a 15.5	90/45 s	1.59 ppm	[87]
	ZnCo ₂ O ₄	Nanoparticles	Co-precipitation/digestion	Liquefied petroleum gas	50	350	^a 72	90/80 s	–	[88]
	ZnCo ₂ O ₄ /Polypyrrole	Nanoparticles	In situ chemical polymerization	Acetone	0.7	RT	^b 0.39	85/34 s	–	[77]
1D	Co ₃ O ₄ /NiCo ₂ O ₄ /CC	Nanowires	Hydrothermal	Ethanol	200	180	^a 35	7/10 s	–	[89]
	NiCo ₂ O ₄ /rGO	Nanowires	Hydrothermal	H ₂ S	100	R	^a 3.51	2/449 s	–	[90]
	Pd/NiCo ₂ O ₄	Nanoneedles	Thin film deposition/ Hydrothermal	H ₂	300	300	^b 533 %	26/23 s	131 ppb	[91]
	NiCo ₂ O ₄ /rGO	Nanorods	Hydrothermal	Ammonia	100	RT	^a 1.068	57/185 s	–	[92]
	NiCo ₂ O ₄	Nanorods	Hydrothermal	Ammonia	25	RT	^a 3.4	33/249 s	–	[93]
	NiCo ₂ O ₄	Hierarchical hollow microtubules	Template-assisted process/Etching treatment	Xylene	100	220	^a 9.25	20/9 s	1 ppm	[94]
	NiCo ₂ O ₄ /SnO ₂	Nanofiber	Electrospinning/ Hydrothermal	Ethanol	100	160	^a 8.87	114/115 s	5 ppm	[95]
	Zn@NiCo ₂ O ₄	Mesoporous rods	Hydrothermal	NO _x	10	200	^b 7.4	28/30 s	–	[96]
	CuO/CuCo ₂ O ₄	Nanotubes	Electrospinning	n-Propanol	10	RT	^a 14	6.3/4.1 s	–	[97]
	TiO ₂ /ZnCo ₂ O ₄	Porous nanorods	Dual-oxalate sacrificial template	HCHO	100	130	^a 4.1	120/200	1.04 ppm	[98]
				TEA	100	220	^a 15	9/77 s	1.12 ppm	
	ZnCo ₂ O ₄	Porous nanorods	Co-precipitation	TEA	100	200	^a 2.7	10/120 s	–	[99]
		Single-layer nanochain					^a 14	7/57 s	–	
	ZnCo ₂ O ₄	Microtubes	Biomorphic template	H ₂ S	10	90	^a 6.37	100/104 s	50 ppb	[100]
	ZnO/ZnCo ₂ O ₄	Nanotubes	Capillary electrospinning	Acetone	100	175	^a 34	3.2/3.4 s	–	[101]
	ZnO/ZnCo ₂ O ₄	Tube in tube nanostructure	Capillary electrospinning	Ethanol	100	150	^a 58	5.6/4.8 s	–	[102]
			Acetone	100	150	^a 38	6.4/8.2 s	–		
			NH ₃	100	150	^a 21	9.3/11.7 s	–		
			Methanol	100	175	^a 25	6.7/9.5 s	–		
Zn _x Co _{2-x} O ₄	Nanorods	Co-precipitation	Liquefied petroleum gas	10	250	^a 77.5	80/65 s	–	[103]	
2D	MgCo ₂ O ₄	Nanosheets	Hydrothermal	Ethanol	500	RT	^f 81 × 10 ⁻³	15/19 s	–	[104]
	NiCo ₂ O ₄ /WO ₃	Nanosheets	Hydrothermal/Co-precipitation	Xylene	100	300	^a 15.69	–	5 ppm	[105]
	ZnO/NiCo ₂ O ₄	Nanosheets with nanofibers	Hydrothermal	Methanol	100	250	^a 6.77	37/175 s	–	[106]
	NiCo ₂ O ₄ /MWCNTs	Nanoflakes	Hydrothermal	Ammonia	100	RT	^g 25.7	–	–	[107]
	NiCo ₂ O ₄	Nanoflakes	Co-precipitation	Ammonia	10	RT	^b 109 %	44/18 s	–	[108]
	NiCo ₂ O ₄	Nanoplates	Hydrothermal	Triethylamine	10	220	^a 2.53	33/42 s	500 ppb	[109]
	NiCo ₂ O ₄ /α-MoO ₃	Porous nanosheets	Co-precipitation	Ethanol	1	350	^a 20	–	50 ppb	[110]
	NiCo ₂ O ₄	Nanosheets	Hydrothermal	Formaldehyde	100	225	^a 70	22/34 s	–	[111]

(continued on next page)

Table 4 (continued)

Dimension	Composites	Microstructures	Synthesis method	Target gas	Concentration (ppm)	T (°C)	Response	Response/Recovery time	LOD	Ref.
3D	NiCo ₂ O ₄ /rGO	Nanoplates		Alcohol	–	280	–	–	–	[112]
	CuCo ₂ O ₄	Nanosheets	Solution combustion method	Ozone	1	90	^a 27	–	3.9 ppb	[113]
	CuCo ₂ O ₄	Nanoplatelets	Hydrothermal	Ammonia	400	RT	^b 7.9 %	–	–	[114]
	MoS ₂ -ZnCo ₂ O ₄ -ZnCo ₂ O ₄ /CC	Nanosheets	Hydrothermal	H ₂ S	1 mM	37	3.8 mA	–	5 nM	[115]
	ZnCo ₂ O ₄	Porous nanosheets	Hydrothermal	H ₂ S	5	120	^a 6.27	35/- s	100 ppb	[116]
	ZnCo ₂ O ₄	Nanosheet arrays	Hydrothermal	Methanol	100	400	^a 9	2/23 s	–	[117]
	Co ₃ O ₄ /NiCo ₂ O ₄	Nanocages	Hydrothermal/Co-precipitation	H ₂ S	100	250	^a 57	153/40 s	–	[118]
	NiO/NiCo ₂ O ₄	Hierarchical microspheres	Hydrothermal	Xylene	50	RT	^a 4	8/4 s	0.5 ppm	[119]
	NiCo ₂ O ₄	Nanospheres	Hydrothermal	Xylene	100	260	^a 16.4	26/27 s	–	[120]
	PdO-NiO/NiCo ₂ O ₄	Truncated Nanocages	Coprecipitation/etching/wet impregnation	Acetone	100	210	^a 6.7	19/28 s	–	[121]
	NiO/NiCo ₂ O ₄	Hollow microspheres	Co-precipitation	Acetone	100	160	^a 17.86	11/13 s	25 ppb	[76]
	Co ₃ O ₄ /NiCo ₂ O ₄	Nanocages	Co-precipitation	Acetone	100	238.9	^a 3.1	8/20 s	–	[122]
	NiCo ₂ O ₄ /WO ₃	Nanoflowers	Hydrothermal	NO ₂	20	150	^a 116.9	13/16 s	52 ppb	[123]
	NiCo ₂ O ₄	Hollow microspheres	Hydrothermal	Triethylamine	500	180	^a 2.26	30/37 s	145 ppb	[124]
	NiCo ₂ O ₄	Hierarchical microflowers	Hydrothermal	n-butanol	100	165	^b 240 %	68/107 s	–	[125]
	NiCo ₂ O ₄	Double-shelled hollow spheres	Self-template	Xylene	100	240	^a 23.3	15.4/30.5 s	0.5 ppm	[126]
	PANI/NiCo ₂ O ₄	Nanospheres	Hydrothermal	Ammonia	20	RT	^a 4.67	22/62 s	216 ppb	[127]
	Cu _{0.3} Co _{2.7} O ₄	Hollow microspheres	Hydrothermal	Ethanol	50	190	^a 2.09	–	–	[128]
	ZnO/ZnCo ₂ O ₄	Hexahedral-structure	Hydrothermal	Formaldehyde	20	200	^a 17	390/174 s	–	[129]
	ZnCo ₂ O ₄	Hierarchical porous architectures	Hydrothermal	Xylene	200	260	^a 50	1/12 s	1 ppm	[130]
	Co ₃ O ₄ /ZnCo ₂ O ₄	Hollow nanostructures	Self-sacrificing template	Acetone	100	255	^a 16.3	41/47 s	0.51 ppm	[131]
	Ag-ZnCo ₂ O ₄	Hollow microspheres	Solvent-thermal	Formaldehyde	100	99.3	^a 11.18	76/127 s	–	[132]
	ZnCo ₂ O ₄	Hierarchical microflowers	Solvothermal	Acetone	100	200	^a 32.32	2.6/8.8 s	50 ppb	[133]
ZnCo ₂ O ₄	Yolk-shell spheres	Hydrothermal	Acetone	500	200	^a 38.2	19.72/500 s	0.5 ppm	[134]	
Ag@ZnCo ₂ O ₄	Hollow spheres	Self-sacrificing template	Acetone	20	220	^a 12	12/54 s	0.25 ppm	[135]	
ZnCo ₂ O ₄	Yolk-shell microspheres	Co-precipitation	Ozone	890 ppb	200	^b 71 %	37.4/1260 s	80 ppb	[136]	
ZnO/ZnCo ₂ O ₄	Hierarchical structure	Hydrothermal	TEA	100	220	^a 5.12	8/65 s	800 ppb	[137]	
Pd-ZnO/ZnCo ₂ O ₄	Hollow spheres	MOF templated synthesis	Acetone	5	250	^b 69 %	–	–	[138]	
PdO-ZnO/ZnCo ₂ O ₄	Microspheres	Co-precipitation	Formaldehyde	100	139	^a 26.9	9/14 s	0.2 ppm	[139]	
ZnCo ₂ O ₄	Hollow polyhedral superstructure	Co-precipitation	Ethanol	20	200	^b 14.2	22/62 s	1 ppm	[140]	
ZnCo ₂ O ₄	Multishelled Hollow	Solvothermal	Formaldehyde	0.6	170	^b 12	15/266 s	–	[36]	
ZnCo ₂ O ₄	Twin Spheres	Solvothermal	Propanal	0.6	190	^b 16	3/100 s	–		
ZnO/ZnCo ₂ O ₄	Hierarchical macroporous honeycomb-like	Self-template	CO	10	300	^a 17.3	300/50 s	–	[141]	
Zn _x Co _{3-x} O ₄	Nanocages	Co-precipitation	Acetone	200	170	^a 35.6	43/51 s	0.5 ppm	[142]	
ZnCo ₂ O ₄	Cabbage-shaped	Co-precipitation	Formaldehyde	100	180	^a 7.4 ± 0.5	9/12 s	–	[143]	

RT: Room temperature.

CC: Carbon cloth.

^aGas response $S = R_g/R_a$.^bGas response $S = (R_a - R_g)/R_a$.^cGas response $S = \Delta V_a - \Delta V_b$ (ΔV is the electric potential difference of the sensor).

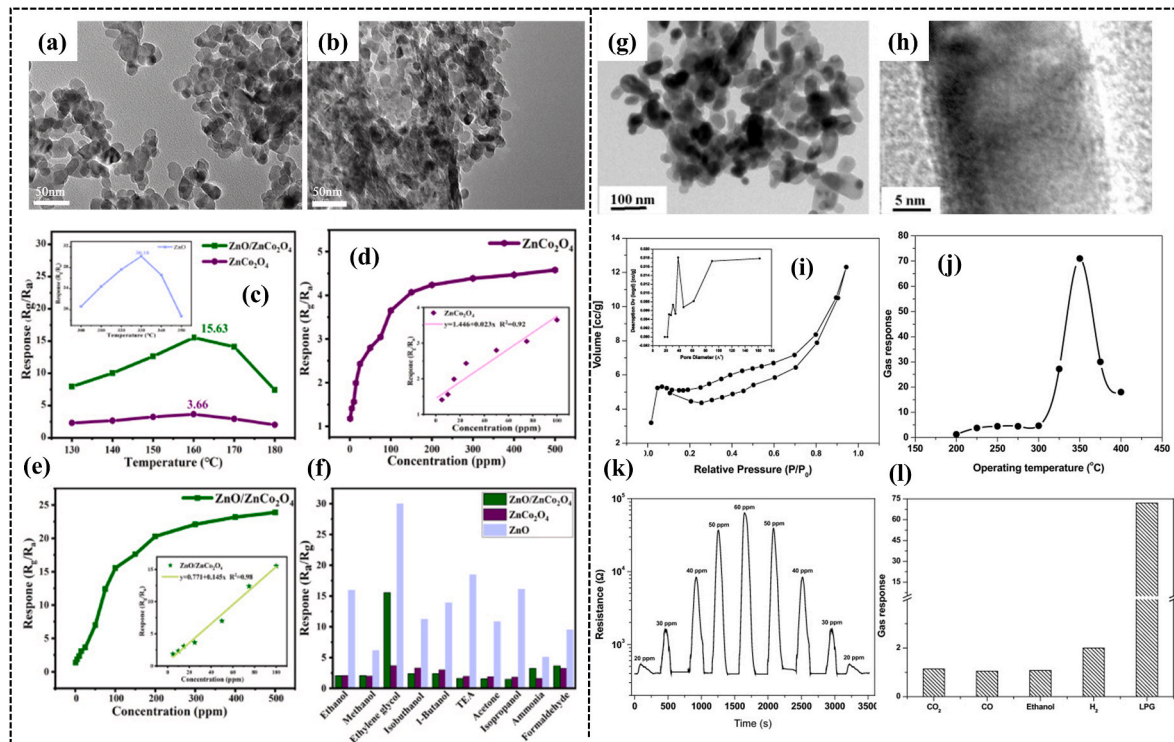
^dGas response $S = \text{peak current}$.^eGas response $S = \text{counts/ppm}$.^fGas response $S = \text{output voltage/gas concentration}$.^gGas response $S = C_{\text{gas}}/C_{\text{air}}$ (C_{air} represents the capacitance of the sensor in the air and C_{gas} is the capacitance in testing gas).

Fig. 2. TEM images of (a) ZnCo_2O_4 and (b) $\text{ZnO}/\text{ZnCo}_2\text{O}_4$. (c) The responses to 100 ppm ethylene glycol at different operating temperatures from 130 to 180 °C, (d and e) the responses to different concentrations of ethylene glycol ranging from 5 to 500 ppm at 160 °C, (f) the response values of the sensors to 100 ppm different gases at the optimum operating temperature. (Reproduced with permission from Ref. [87], Copyright© 2022, Elsevier). TEM (g) and HRTEM (h) images of nanostructured ZnCo_2O_4 . (i) Nitrogen adsorption–desorption isotherm of nanostructured spinel ZnCo_2O_4 . Inset shows the BJH pore size distribution obtained from the desorption branch of isotherm. (j) Effect of operating temperature on the gas response of nanostructured spinel ZnCo_2O_4 to 50 ppm LPG. (k) Response of nanostructured spinel ZnCo_2O_4 upon sequential exposure to LPG with concentrations varying from 20 to 60 ppm at 350 °C. (l) Bar chart showing the gas response of nanostructured spinel ZnCo_2O_4 for different gases. The gas concentration and operating temperature in all cases were 50 ppm and 350 °C, respectively. (Reproduced with permission from Ref. [88], Copyright© 2011, Elsevier).

molecules [165]. Polymer polypyrrole (PPy) has shown the potential gas sensor application for its tunable electrical and nonlinear optical properties [166,167]. By constructing $\text{ZnCo}_2\text{O}_4/\text{PPy}$ and $\text{MnCo}_2\text{O}_4/\text{PPy}$ heterojunction nanocomposites via a hydrothermal method, a highly sensitive acetone gas sensor was obtained as the ternary spinel SMOX served on effective catalysts for gas molecules oxidation due to the combination of couples $\text{M}^{3+}/\text{M}^{2+}$ (Zn and Mn) and $\text{Co}^{3+}/\text{Co}^{2+}$ [36,77]. Yuan et al. reported that they loaded FeCo_2O_4 nanoparticles on the WO_3 nanosheets to form the p-n heterogenous junction for constructing the NH_3 gas sensors with superior performance. WO_3 oxide nanosheets coated with FeCo_2O_4 nanoparticles were synthesized via regulating the content of HCl in precursor solution. The corresponding gas sensor synthesized by adding 2 mL HCl displayed highest response (34 @ 100 ppm), which is primarily due to the special binding form of Co^{2+} , Co^{3+} and NH_3 . Meanwhile, the extra nanostructure and p-n heterojunction benefits the sensing performance as well [85].

On the other hand, in addition to the traditional hydrothermal method, other methods, including co-precipitation [80], microwave-assisted colloidal [79], self-propagating high-temperature synthesis [82], solution combustion [168] et al. are employed to fabricating OD MCo_2O_4 gas sensors. The nanostructured ZnCo_2O_4 were obtained utilizing the co-precipitation and digestion methods while the ZnCo_2O_4 nanoparticles with the size of 30–40 nm and the BET surface area of 16.2 m^2/g were obtained as shown in Fig. 2a-c. As evident in

Fig. 2d-f, the as-fabricated sensor demonstrated the outstanding sensitivity of 72 towards 50 ppm LPG at the optimal operating temperature, excellent dynamic response condition and superior selectivity, which shed some light on combustible and explosive energy gas detection [88]. Via regulating the Ni/Co content in nanocrystalline NiCo_2O_4 , Vladimirova et al. obtained the $\text{Ni}_x\text{Co}_{3-x}\text{O}_4$ product by co-precipitation strategy for exploring the impact of cationic distribution on sensing performance of carbon monoxide. The $\text{Ni}_x\text{Co}_{3-x}\text{O}_4$ samples were formed from aggregated nanoparticles with the wide size distribution interval from 20 nm to 100 nm). In detail, the nanoparticles own a cellular nanostructure with the average pore diameter of about 8–15 nm, which is due to the gas emission (CO , CO_2) in time of oxalate precursor decomposition. With the increase in Ni proportion, the gas sensitivity of CO decreases, owing to the decline of adsorbed oxygen concentration on $\text{Ni}_x\text{Co}_{3-x}\text{O}_4$ surface [80]. Morán-Lázaro et al. reported a microwave-assisted colloidal technique to obtain MgCo_2O_4 nanoparticles and the size of nanoparticles are 6–33 nm while their average size is around 16 nm, which aggregate to form the irregular porous microstructure. Dioctyl sulfosuccinate sodium salt (AOT), an effective surfactant, made the porous structure synthesized successfully. Exploring on the acetone sensing performance revealed that the response can reach 17.4 under 100 ppm target gas at 250 °C under low-humidity conditions. Nevertheless, on the account of competition of water molecules on the MgCo_2O_4 , response drops below 2 when the ambient RH is 50 % [79].

Due to the high RH environment, some hydroxyl groups are prone to absorb on the surface of sensing material. Because of the existence of hydroxyl groups, the highly stable adsorption states can be formed. Thus, the hydroxyl groups interferes with the regular sensing process by preventing it from reacting with the reaction sites on the surface [169, 170]. It can be seen that the sample synthesized by microwave hydrothermal method possessed high purity and reaction conditions is mild, which should be widely used in the synthesis of 0D MCo_2O_4 . Furthermore, co-deposition and hydrothermal method were combined to successfully modify ZnO nanorods with ZnCo_2O_4 nanodots. The $\text{ZnCo}_2\text{O}_4/\text{ZnO}$ precursor was obtained appropriately by adding certain amount of NaCl for inhibiting deposition of crystallites. Furthermore, based on the thermogravimetric (TG) curve, the 600°C was selected as the optimal calcination temperature to obtain the best substance. Thus, the gas sensor exhibited intensive response of 15.3 and rapid response process of 7.7 s at 240°C towards 10 ppm acetone, which was attributed to the ZnO nanorods surface modifications by adding ZnCo_2O_4 nanodots [86]. As stated above, 0D MCo_2O_4 synthesized by most researchers typically are nanoparticles or clusters, which have established the basis for subsequent assembly into 3D MCo_2O_4 materials.

3.2.2. 1-dimensional (1D) MCo_2O_4

Up to now, a wide variety of 1D MCo_2O_4 , containing nanorods, nanofibers, nanowires, nanotubes, nanoneedles and nanochains are explored for improving sensing performance. Due to inherent quantum confinement properties of 1D material, they have unique shapes and size dependent features, considered as an exciting class of materials for many nano-technology related sensors [171]. Among the developed synthesis methods, hydrothermal and solvothermal methods are extensively used because of high yield, facile control and operation simplicity [70]. For example, a $\text{Co}_3\text{O}_4/\text{NiCo}_2\text{O}_4/\text{Carbon cloth}$ (CC) multi-layered nanostructure was prepared for ethanol sensing application. Specifically, CC is applied for rock-bottom template while cobaltate, nickelate and cetyltrimethyl ammonium bromide (CTAB) were added during

hydrothermal reaction to form $\text{ZIF-67}/\text{NiCo}_2\text{O}_4/\text{CC}$. Finally, $\text{Co}_3\text{O}_4/\text{NiCo}_2\text{O}_4/\text{CC}$ was successfully obtained by calcination at 350°C during 1 h with the heating rate of 1°C per minute. By this ordered synthesis route as demonstrated in Fig. 3a, uniform Co_3O_4 nanocubes grown on the NiCo_2O_4 nanowires of around 80 nm in diameter based on the CC substrate (Fig. 3b and c). Surprisingly, the as-developed sensors demonstrated the excellent sensing performance ($35 @ 200$ ppm ethanol) at 180°C as depicted in Fig. 3d and e. The strengthened sensing performance is derived from the porous nanostructure and facilitated carrier transportation between CC and Co_3O_4 nanocube group [89]. In Ref. [90], H_2S sensing with $\text{NiCo}_2\text{O}_4/\text{reduced graphene oxide}$ (rGO) nanocomposites electronic devices was firstly reported by Wu et al. From SEM results, NiCo_2O_4 nanowires with the diameter of $5\text{--}7\ \mu\text{m}$ tightly fixed on the rGO layered structure, coming into being a great number of active sites. Furthermore, the excellent response (3.51) and ultrafast response process (2 s) towards 100 ppm H_2S were achieved which was attributed to the catalytic action of sensing layer. The group of Marimuthu et al. employed rGO catalyst for functionalizing of NiCo_2O_4 . The composites are made up of rGO nanosheets and NiCo_2O_4 nanorods which are attached to a certain center forming a bunch-like nanostructure. The sensors exhibited high sensitivity, excellent reproducibility of ten days period and fast response/recovery reaction to NH_3 [92]. Furthermore, the Co_3O_4 nanoparticles and NiCo_2O_4 nanorods were prepared respectively to fabricate the ammonia sensor by Marimuthu et al. while the sensing performance at room temperature of the latter one was better [93]. Kumar et al. adopted a facile hydrothermal strategy for mesoporous Zn doped NiCo_2O_4 rods preparation. The length of log-linked $\text{Ni}_{0.8}\text{Zn}_{0.2}\text{Co}_2\text{O}_4$ rod is $5\text{--}10\ \mu\text{m}$ while the BET surface area of which is as high as $76\ \text{m}^2/\text{g}$. Compared to the pure NiCo_2O_4 and ZnCo_2O_4 nanorods, the $\text{Ni}_{0.8}\text{Zn}_{0.2}\text{Co}_2\text{O}_4$ sensor exhibited higher response ($7.4 @ 10$ ppm) and short response time towards NO_x [96].

The MCo_2O_4 sensor owns better performance by combining the traditional hydrothermal method with different deposition methods. For fabricate an more effective hydrogen sensor, the NiCo_2O_4 nanoneedle

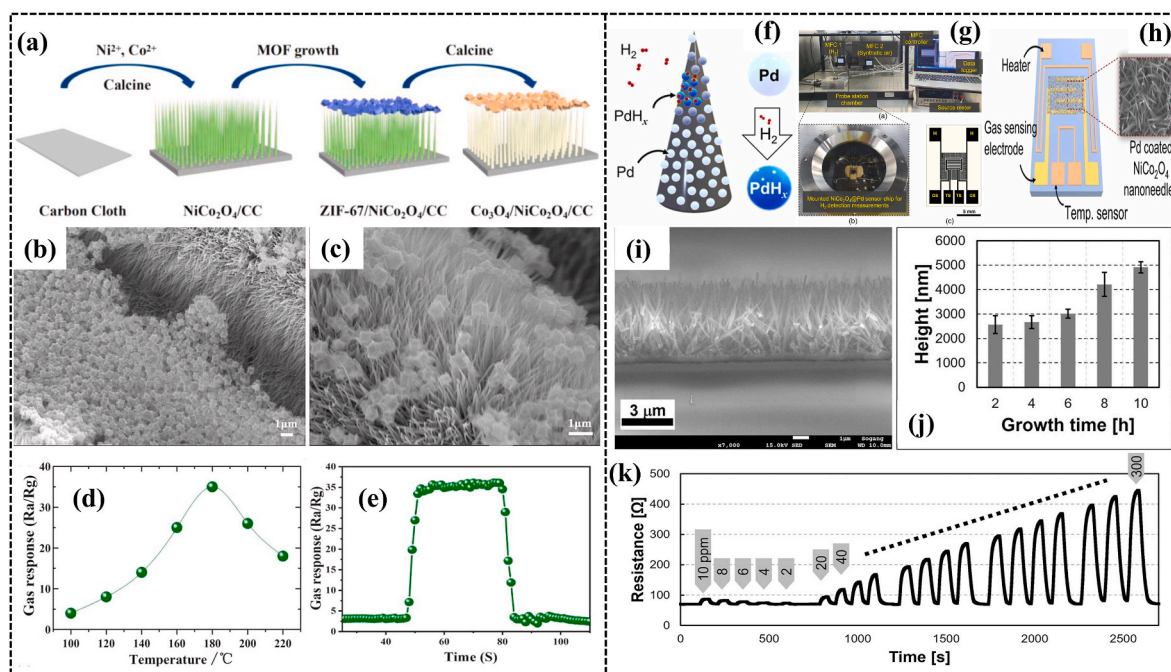


Fig. 3. (a) Growth mechanism diagram of synthesis process of $\text{Co}_3\text{O}_4/\text{NiCo}_2\text{O}_4/\text{CC}$. (b–c) SEM images of $\text{Co}_3\text{O}_4/\text{NiCo}_2\text{O}_4/\text{CC}$ under different magnification. (d) Gas response of $\text{Co}_3\text{O}_4/\text{NiCo}_2\text{O}_4$ exposed to 200 ppm ethanol at different temperature. (e) Response and recovery curves under 200 ppm of ethanol at 180°C . (Reproduced with permission from Ref. [89], Copyright© 2021, Elsevier). (f) The process of H_2 adsorption by Pd coated on the surface of NiCo_2O_4 NNF. (h) Experimental setup for target gas sensing measurements. (g) Schematic illustration of proposed H_2 sensor. (i–j) SEM images of cross-sections of structure with 10 h for NNF and comparison graph of structure layer height according to growth time. (k) Comparison of responses of the fabricated device to various H_2 concentrations (Reproduced with permission from Ref. [91], Copyright© 2023, Elsevier).

forest which deposited on Pd thin film was applied as shown in Fig. 3f–h. Through field emission scanning electron microscope (FESEM) results, the height of nanoneedle forest is proportional to the synthesis time and which can reach 4.9 μm after 10 h heating as illustrated in Fig. 3i and j. Meanwhile, the diameter of the single nanoneedle also becomes larger after prolonging the hydrothermal time. By depositing an ultrathin Pd film on the sensors, the intricate reaction between H_2 molecules and sensing material surface can be motivated while the fabrication costs can be reduced. Furthermore, the high BET surface area and appropriate porosity also improve the sensitivity a lot (Fig. 3k) [91].

Furthermore, Wang et al. combined electrospinning and hydrothermal methods to synthesize the $\text{NiCo}_2\text{O}_4/\text{SnO}_2$ heterostructure. The SnO_2 nanosheets were adhered uniformly to the NiCo_2O_4 nanofibers, leading to the loose and poly porous structure and increased exposed active sites as shown in Fig. 4a–g, which improved the response values 8 times than that of pure NiCo_2O_4 towards 100 ppm ethanol at optimum working temperature of 160 $^\circ\text{C}$ (Fig. 4h). Interestingly, a typical logarithmic relationship between response and ethanol concentration was discerned in Fig. 4i [95].

Besides adopting traditional hydrothermal method, electrospinning, coprecipitation and template method were also commonly used for the synthesis of 1D MCo_2O_4 . For example, Alali et al. established $\text{CuO}/\text{CuCo}_2\text{O}_4$ p-p heterojunction employing electrospinning strategy for n-propanol detection at room temperature. The $\text{CuO}/\text{CuCo}_2\text{O}_4$ nanotubes with hollow and uniform features were obtained and the diameter of which was calculated to be 60–70 nm [97].

Meanwhile, a p-n $\text{ZnO}/\text{ZnCo}_2\text{O}_4$ heterojunction was reported by the preparation of single capillary electrospinning technology and post-annealing process. Obviously, the diameter of hollow $\text{ZnO}/\text{ZnCo}_2\text{O}_4$ nanotubes cross profile is about 100–150 nm and the wall thickness is about 20 nm. The excellent gas sensitivity (34 @ 100 ppm) and ultra-

short response/recovery time (3.2/3.4 s) towards acetone are acquired which is derived from the distinct morphology and p-n heterostructure [101]. Interestingly, TiO_2 can replace ZnO totally to form heterojunction with 1D MCo_2O_4 . A p-n $\text{TiO}_2/\text{ZnCo}_2\text{O}_4$ heterojunction was constructed by sacrificial template method and acquired the best relevant gas sensor through changing the amount of TiO_2 . The $\text{TiO}_2/\text{ZnCo}_2\text{O}_4$ nanorods consisting of abundant irregular nanoparticles (5–20 nm) are discerned and the adding of TiO_2 exerts no distinct impact on morphology. Specifically, the ZnCo_2O_4 nanorods with the optimum TiO_2 content (4 wt%) discriminate HCHO and triethylamine (TEA) effectively. The developed gas sensor has the high response (15 @ 100 ppm) towards TEA at 220 $^\circ\text{C}$ and the response time is decreased to 9 s [98]. Meanwhile, TEA detection was also carried out using ZnCo_2O_4 by Luo et al. By adjusting the calcination time (350 $^\circ\text{C}$ and 500 $^\circ\text{C}$), porous ZnCo_2O_4 nanorod and single-layer ZnCo_2O_4 nanochain can be obtained, respectively. The ZnCo_2O_4 calcined at 500 $^\circ\text{C}$ exhibited better TEA sensing performance and the larger available surface area should be the primary reason [99]. The ZnCo_2O_4 microtubes as shown in Fig. 4j and k adopting absorbent cotton as template was synthesized by Xu et al. The immersion tactic in metal salt solution and calcination treatment in ambient atmosphere were combined to target material. By regulating calcination temperature (500 $^\circ\text{C}$, 600 $^\circ\text{C}$ and 700 $^\circ\text{C}$), the percentage of chemisorbed oxygen in ZCO600 declined from 37.81 % to 31.77 % and the oxygen vacancy increased after the H_2S molecules adsorption, indicating H_2O and SO_2 were generated after H_2S reacted with O_2 as shown in Fig. 4l. Furthermore, the gas sensor fabricated under 600 $^\circ\text{C}$ annealing temperature demonstrated rapid recovery process of 104 s to H_2S (5 @ 10 ppm) and the lowest LOD of 50 ppb at 90 $^\circ\text{C}$ as shown in Fig. 4m and n. In particular, XPS analysis were employed to evaluate the valence state of element on the surface of ZnCo_2O_4 when the as-prepared sensor contacted with H_2S at 90 $^\circ\text{C}$ [100]. The research provided us a more

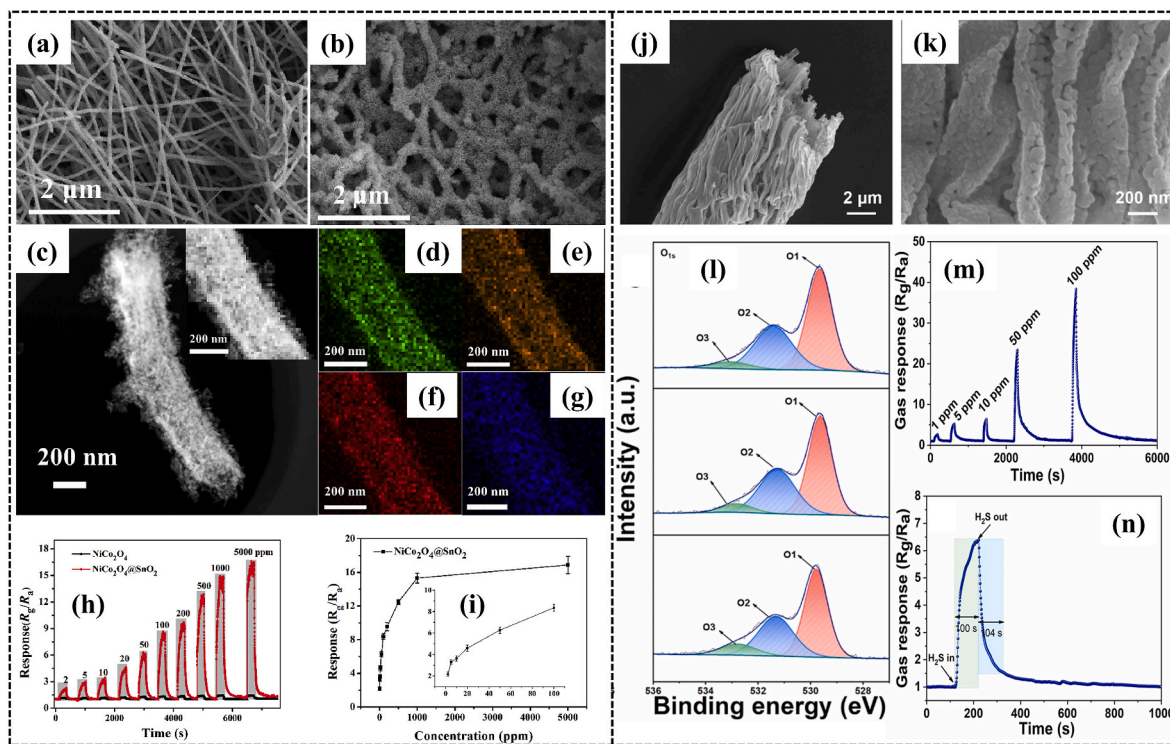


Fig. 4. SEM images of (a) NiCo_2O_4 and (b) $\text{NiCo}_2\text{O}_4/\text{SnO}_2$. (c) HAADF-STEM image of $\text{NiCo}_2\text{O}_4/\text{SnO}_2$ composites. (d–g) EDX elemental mappings of Ni, Co, O and Sn, respectively. (h) Dynamic transient for response of NiCo_2O_4 nanowires and $\text{NiCo}_2\text{O}_4/\text{SnO}_2$ composites to 2–5000 ppm ethanol at optimal working temperature of 160 $^\circ\text{C}$. (i) $\text{NiCo}_2\text{O}_4/\text{SnO}_2$ sensor response vs. ethanol concentration (2–5000 ppm) at 160 $^\circ\text{C}$, inset is the corresponding responses at low ethanol concentrations. (Reproduced with permission from Ref. [95], Copyright© 2019, Elsevier). (j, k) SEM images of ZCO-500, ZCO-600 and ZCO-700. (l) O 1s XPS spectrum of ZCO-500, ZCO-600 and ZCO-700. (m) Dynamic sensing response to different concentrations of H_2S at 90 $^\circ\text{C}$. (n) Dynamic response-recovery curve to 10 ppm H_2S . (Reproduced with permission from Ref. [100], Copyright© 2022, Elsevier).

effective strategy to verify the physicochemical change before and after the adsorption process. Furthermore, the calcination temperature has displayed a crucial impact on the surface activity of the MCo_2O_4 , while the high surface activity will increase the active site and surface adsorption capacity, thus improving the adsorption performance.

By modifying the elemental ratio of zinc to cobalt from 1 : 1 to 1 : 2.5 in ZnCo_2O_4 nanorods synthesized via co-precipitation/digestion strategy, Gawande et al. fabricated the gas sensor with optimum zinc-cobalt ratio (1 : 2.5) for liquefied petroleum gas (LPG) detection. The $\text{ZnCo}_{2.5}\text{O}_4$ nanorods were observed with diameter of 6–8 nm and length of 30–50 nm. Towards the LPG gas (10–80 ppm), the developed sensors exhibited high response, good selectivity and repeatability for 40 cycles [103]. Du et al. successfully synthesized a hierarchical tubular NiCo_2O_4 nanostructures which assembled by nanosheets via template method and etching treatment for xylene detection. In detail, the Ni–Co precursor was anchored on $\text{Ag}_8\text{W}_4\text{O}_{16}$ templates firstly and the $\text{Ag}_8\text{W}_4\text{O}_{16}/\text{NiCo}_2\text{O}_4$ composites can be obtained after annealing process. Then nitric acid and ammonia are used for the removal of $\text{Ag}_8\text{W}_4\text{O}_{16}$ templates and the thickness of NiCo_2O_4 nanosheets obtained is around 5 nm and the average length of assembled microtubes is 1.1 μm [94]. As described above, large specific surface area, fast carrier diffusion rate and strong structural stability compared with other dimensional materials, the three leading advantages which 1D MCo_2O_4 owns, endow them excellent gas sensing performance. However, due to the characteristics of the synthesis method, the reagent needs extremely high purity to ensure the quality of the product, therefore the large-scale production and widespread practical application of 1D MCo_2O_4 materials still need a lot of attempts and efforts.

3.2.3. 2-dimensional (2D) MCo_2O_4

At present, many researchers have been motivated to find more 2D MCo_2O_4 materials which possess special thickness tied properties in both chemistry and physics [4]. However, there is the tendency of abundant low dimensional nanomaterials taking shape in dense stacked nanostructure in time of establishment of the 2D conductive network, which is not beneficial to support sufficient contact between the lamellae inside the network and target gas, resulting in the decrease of sensitivity and response rate [172]. Therefore, some 2D materials with extremely ordered morphology can be obtained by synthesizing on some hard templates like Ni foam which possesses 3D grid microstructure and its large porosity can provide more location for the orderly growth of materials. The research group of Yin has fabricated special hexagonal ZnCo_2O_4 nanosheets arrays of single crystalline developed on nickel foam by a two-step synthesis method including hydrothermal and air calcination process, which own high specific area (94.1 m^2/g) and mesoporous structure properties contributing to the enhancing sensing performance. From FESEM results, the prepared hexagonal ZnCo_2O_4 nanosheets with the thickness of 84 nm attached to Ni foam densely. The response value of ZnCo_2O_4 sensor was 9 towards 100 ppm methanol at 400 °C. As the gas concentration increased from 100 to 1000 ppm, the sensor displayed the outstanding sensitivity, good linearity and adequate selectivity to target gas over interfering gas [117]. MgCo_2O_4 nanosheets were firstly obtained with the thickness of 10–15 nm while the diameter of 200–250 nm employing a hydrothermal strategy. The as-prepared sample owned the high BET surface area of 98.5 m^2/g and the according sensor displayed high response to ethanol [104]. Xu et al. reported a novel $\text{NiCo}_2\text{O}_4/\text{WO}_3$ gas sensor for xyene detection. In detail, the 2D NiCo_2O_4 nanosheets are observed to stick to every single 1D WO_3 nanofiber uniformly. Residual NiCo_2O_4 nanoparticles still adhere on WO_3 nanofibers, which indicated low crystallinity of NiCo_2O_4 nanosheets. By exploring the xyene sensing performance, $\text{NiCo}_2\text{O}_4/\text{WO}_3$ gas sensors exhibited high response ($R_g/R_a = 15.69$), excellent selectivity and magnificent stability toward 100 ppm xyene at 300 °C on account of the unique and exquisite morphology and p-n heterojunction [105]. Liang et al. synthesized a novel ZnO-core/ NiCo_2O_4 -shell nanocomposite chemical bath deposition for methanol detection. Porous NiCo_2O_4

nanosheets anchored on the ZnO nanofiber. Apparently, the $\text{ZnO}/\text{NiCo}_2\text{O}_4$ sensor exhibited better sensing performance than both pristine ZnO and NiCo_2O_4 . It should be noted that the complex and porous surface is of great benefit to sensing performance owing to the swift and adequate gas diffusion including the material. The $\text{ZnO}/\text{NiCo}_2\text{O}_4$ gas sensor owned outstanding response of 6.77 towards 100 ppm methanol at 250 °C [106]. Marimuthu et al. synthesized an exquisite $\text{NiCo}_2\text{O}_4/\text{multi-walled carbon nanotubes (MWCNTs)}$ nanocomposite for fabricating ammonia gas sensor via a hydrothermal method. The NiCo_2O_4 2D nanoflakes wrap together with MWCNTs were observed and the pure NiCo_2O_4 exhibited the outstanding repeatability to ammonia whereas the $\text{NiCo}_2\text{O}_4/\text{MWCNTs}$ nanocomposites do not achieve the same expectation, which may attribute to the loose combination of the composition [107]. As shown in Fig. 5a-d, a novel tortoise shell-like NiCo_2O_4 nanoplates for TEA detection was reported by the group of Zhao et al. The porous NiCo_2O_4 nanoplates which possessed large BET surface area (102.6 m^2/g) can be synthesized by annealing at the optimal temperature of 500 °C whereas the nanoplates are damaged and the larger pores can be observed when annealed at 700 °C. The developed gas sensors owned the high response towards TEA (2.58 @ 10 ppm) at the optimal temperature of 220 °C as shown in Fig. 5e and f. The Ni (II)/Ni (III) ratio of NiCo_2O_4 can be estimated to be 0.25 whereas the value is 0.32 for NiCo_2O_4 after TEA contact. The augment of Ni (II) contents in TEA sensing demonstrated reduction reaction occurred at Ni site [109]. Xu et al. reported a $\text{NiCo}_2\text{O}_4/\alpha\text{-MoO}_3$ nanostructure based gas sensor for highly sensitive ethanol detection. Abundant porous 2D NiCo_2O_4 nanosheets with the width of 500 nm anchoring uniformly on the $\alpha\text{-MoO}_3$ 1D nanorods. The as-prepared gas sensors possessed the ultralow LOD (50 ppb) and the excellent response (20 @ 1 ppm) towards ethanol [110]. Jain et al. synthesized the CuCo_2O_4 nanoplatelets with mixed-valent by a hydrothermal strategy for NH_3 detection at room temperature. The size of porous CuCo_2O_4 nanoplatelets assembled by massive uniform nanoparticles is 10–15 nm. Furthermore, the gas sensing performance of CuCo_2O_4 sample which added 10 at.% Co revealed that the high response (7.9 % @ 400 ppm) was obtained in 57 % RH at room temperature [114]. Mani et al. reported a novel $\text{MoS}_2/\text{ZnCo}_2\text{O}_4/\text{ZnCo}_2\text{O}_4/\text{CC}$ gas sensor for real-time H_2S detection in live cells. The MoS_2 nanosheets grew and evenly distributed on the double-layered ZnCo_2O_4 nanosheets. Owing to the excellent surface properties, porous nanostructure, roughened catalytic sites generated by MoS_2 , the $\text{MoS}_2/\text{ZnCo}_2\text{O}_4/\text{ZnCo}_2\text{O}_4/\text{CC}$ sensor exhibited outstanding sensitivity, good selectivity, wide detection range and low LOD [115]. Combinations of heterojunctions may affect sensing performance, and more research is needed to optimize the assembly. Meanwhile, there are a number of studies on the detection of specific gases, such as methanol, ethanol, ammonia, and hydrogen sulfide, suggesting that these 2D MCo_2O_4 materials have potential applications in gas sensors.

At present, in addition to the traditional hydrothermal method, only solution combustion method and co-precipitation strategy were used separately for the synthesis of 2D MCo_2O_4 sensors. Sharma et al. applied co-precipitation method for $\text{NiCo}_2(\text{OH})_6$ and NiCo_2O_4 gas sensor fabrication. Mesoporous $\text{NiCo}_2(\text{OH})_6$ gas sensor displayed extraordinary response, fast response/recovery process to 2 ppm ammonia. Interestingly, the p-type NiCo_2O_4 gas sensor displayed the abnormal n-type sensing behavior after annealing process which may due to the reducing of charge depletion width and incremental chemisorbed oxygen species, resulting in restoring the dominant charge carriers of NiCo_2O_4 [108]. Zhao et al. synthesized CuCo_2O_4 nanosheets as shown in Fig. 5g and h with regulable oxygen vacancy contents by controlling the amounts of citric acid in precursor employing solution combustion method. The as-prepared CuCo_2O_4 sample, with appropriate oxygen vacancy contents and high specific surface area of 53.7 m^2/g , which will supply sufficient adsorption sites and applicable passages for gas molecules absorption and transportation. As shown in Fig. 5i-l, the developed gas sensors operated at a relative low temperature of 90 °C demonstrated high response (27 @ 1 ppm) towards ozone at 70 % RH, Computable

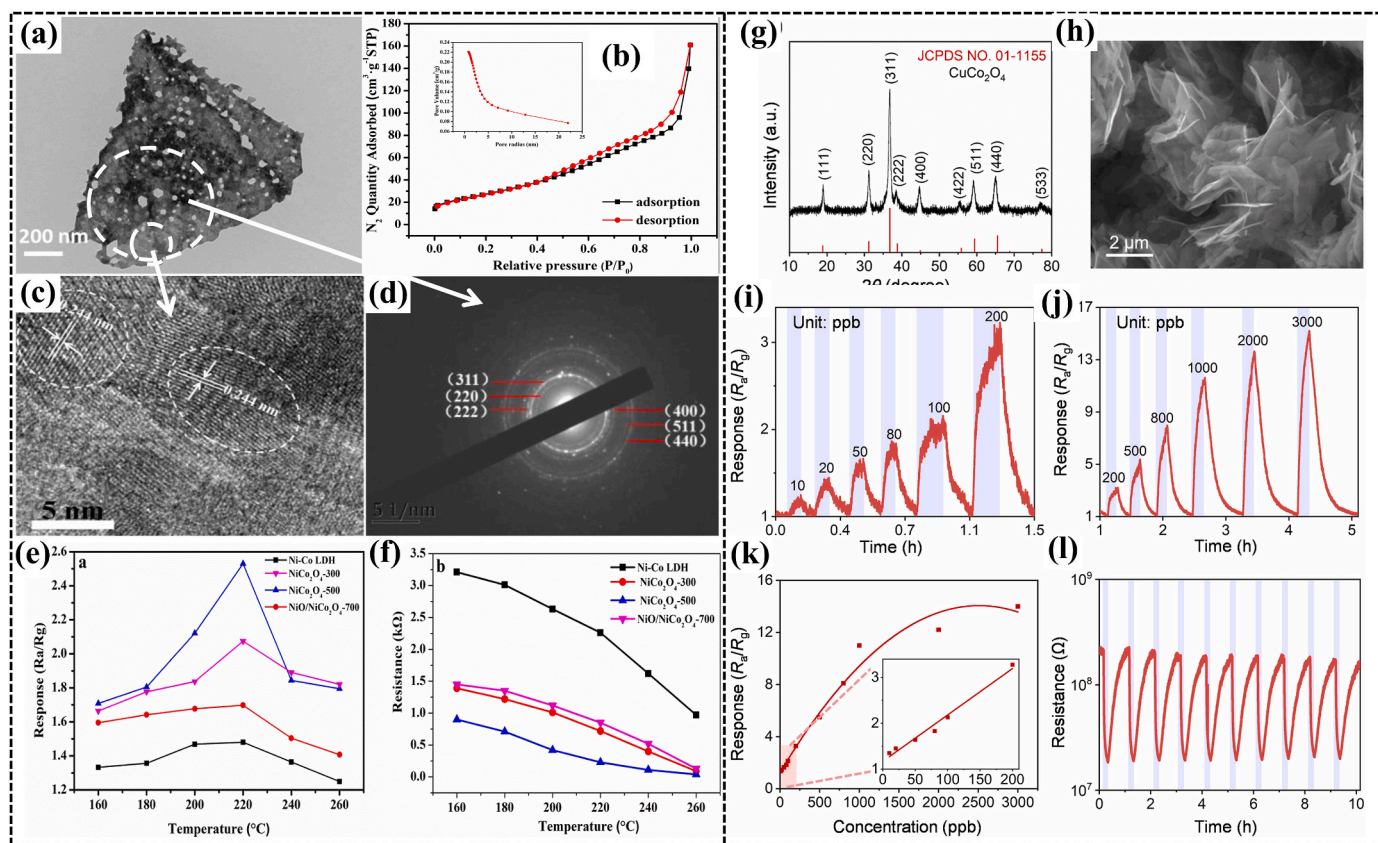
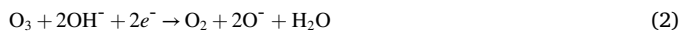


Fig. 5. (a) The TEM images of NiCo₂O₄ samples calcined at 500 °C. (b) Typical N₂ adsorption-desorption isotherm of NiCo₂O₄-500. Inset is pore-size distribution curve. (c) HRTEM and (d) SAED images of NiCo₂O₄-500. (e) The response of as-prepared samples at different temperature to 10 ppm TEA, (f) the change in R_a with operating temperature. (Reproduced with permission from Ref. [109], Copyright© 2020, Elsevier). (g) XRD pattern and (h) SEM image of the as-synthesized CuCo₂O₄. Dynamic response-recovery curves of CuCo₂O₄ sensor to (i) 10–200 ppb O₃ and (j) 200–3000 ppb O₃. (k) The relationship of response versus O₃ concentration in the range of 10–3000 ppb and the inset: enlarged view of 10–200 ppb. (l) The continuous response to cyclic 1 ppm O₃. (Reproduced with permission from Ref. [113], Copyright© 2023, Elsevier).

response to concentration relationship, stable dynamic loop and superior reproducibility. In surprise, the response and the baseline resistance of CuCo₂O₄ gas sensor are positively correlated with the ambient humidity, contrary to the common situation [113]. When the developed sensors are placed at humidity environment, the absorption and dissociation of water molecule will generate electrons to CuCo₂O₄ while the recombination of hole carriers can be discerned, which is illustrated by Eq. (1) [173]:



This reaction possessed a donor feature and reduced hole concentration, the baseline resistance thus rising with the increase of RH. There are two opposite effects on gas sensing application of p-type sensors as shown in Eq. (1). On the one hand, the active oxygen species O₂⁻ can be reduced which is indicated in Eq. (2). However, the O₃ response mostly depends on Eq. (3), which results negative influence of humidity negligible. On the other hand, the hole carriers can be reduced as well which benefits the sensitivity [174].



3.2.4. 3-dimensional (3D) MCo₂O₄

MCo₂O₄ nanoparticles, nanowires and other low dimensional nanostructures are prone to spontaneously take shape in 3D MCo₂O₄ collective structures with periodic holes, named hierarchical nanostructures, which can maximize gas accessibility instead of

sacrificing the hole accumulation layer onto the nanostructured block. In this way, excellent gas sensitivity and fast response are achieved simultaneously, which have made the 3D MCo₂O₄ sensors been reported the most.

Specifically, during the self-assembly process, the polymeric part will take shape in the center while the inorganic part will form on the outer surface. Then the 3D MCo₂O₄ nanostructures with hollow feature can be obtained by thermal decomposition of the core [175]. In several physicochemical methods, solvothermal and hydrothermal reactions are the most widely applied chemical routes for the preparation of highly crystalline grade and hollow 3D MCo₂O₄ nanostructures with periodic and well-defined pores. For instance, the hollow Cu_{1-x}Co_{2+x}O₄ (x = 0.7) microspheres were synthesized by employing a template-free hydrothermal strategy. The diameter of the microspheres is 3.6–4.0 μm while a detailed explanation of the time-dependent crystallization process indicated that the formation mechanism of Cu_{0.3}Co_{2.7}O₄ microspheres was mostly based on the Ostwald ripening mechanism [176]. Using this material for gas sensing testing, the response to ethanol can reach 2.09 at 190 °C [128]. Except 3D hollow MCo₂O₄ structure, the group of Ma find that an additional amount of sodium citrate (0.75 mmol) in the hydrothermal reaction processing of ZnCo₂O₄ with 3D hexahedral morphology can generate some ZnO phase, which presents better formaldehyde sensing properties compared with pure ZnCo₂O₄ [129]. To effectively test for TEA, ZnO/ZnCo₂O₄ microflowers with the diameter of 4 μm that comprised of abundant nanosheets with the average thickness of 15 nm were obtained. By adjusting the loading amount of zinc oxide, the overall carrier properties (p type/n type) of the ZnO/ZnCo₂O₄ SMOX and the content of adsorbed oxygen can be

changed while the better TEA gas sensor can be acquired. Under 220 °C, the ZnO/ZnCo₂O₄ gas sensor with 5%wt ZnO additive amount exhibited the response of 5.12 with the response/recovery time of 8/65 s to 100 ppm TEA [137]. Three diverse hydrothermal paths were adopted to synthesize the ZnCo₂O₄ nanotubes, nanosheets and yolk-shell spheres for acetone sensing. Among them, 3D ZnCo₂O₄ yolk-shell spheres with multi-shelled characteristics displayed outstanding response owing to the promoted gas molecule accessibility, multi-shell morphology, decreased crystalline size and large specific surface area [134]. Li et al. reported hollow porous tube assembled ZnCo₂O₄ porous architectures (Fig. 6a and b) combined solvothermal strategy and post-annealing treatment. The end of the ZnCo₂O₄ hollow tube joined together to form a concentric micro-flower with a diameter of 10–20 μm while the cross-section diameter of the hollow tube is around 500–600 nm confirmed by FESEM and transmission electron microscopy (TEM), which contributes the high surface area (35.6 m²/g). Thus the extremely stable sensing sensitivity and ultrafast response and recovery process were achieved as shown in Fig. 6c and d [130]. A porous hierarchical ZnCo₂O₄ microstructure which has a feature of micro-flower composed of abundant nanosheets (Fig. 6e and f) was performed by Li et al. The micro-flower ZnCo₂O₄ sensor exhibited prominently excellent ppm-level

acetone response (32.32), low LOD (2.51 @ 50 ppb), ultrafast response and recovery time (2.6/8.8 s) and excellent long-term stability towards 100 ppm acetone at 200 °C as shown in Fig. 6g and h. The superior acetone sensing properties was based on the abundant transport gas channels, more active sites derived from higher specific surface area of 95.04 m²g⁻¹, massive triangular and polygonal cone hollow spaces in ZnCo₂O₄ micro-flower [133].

As its low cost and high efficiency, template synthesis method based on precursor pyrolysis is considered to be the most appropriate one for obtaining 3D hollow nanostructure and microstructure [177,178]. Thus, the self-sacrificing template method was applied to prepare Co₃O₄/ZnCo₂O₄ hollow nanostructure derived from MOF. Specifically, ZIF-67 was fabricated through aging processing of Co(NO₃)₂ and methanol solution of 2-MeIm. As a template, ZIF-67 was added to obtain Co/Zn-ZIF@Co-Zn LDH while annealing is the last procedure to obtain Co₃O₄/ZnCo₂O₄ hollow nanostructure. The FESEM results displayed the Co₃O₄/ZnCo₂O₄ nanostructure owned the irregular rhombus decahedron morphology while the surface was kind of rough owing to the degradation of organic ligands. The excellent surface properties entitled Co₃O₄/ZnCo₂O₄ with an amplified target-receptor interface was attributed to the large BET surface area (83.9 m²/g), rendering it a

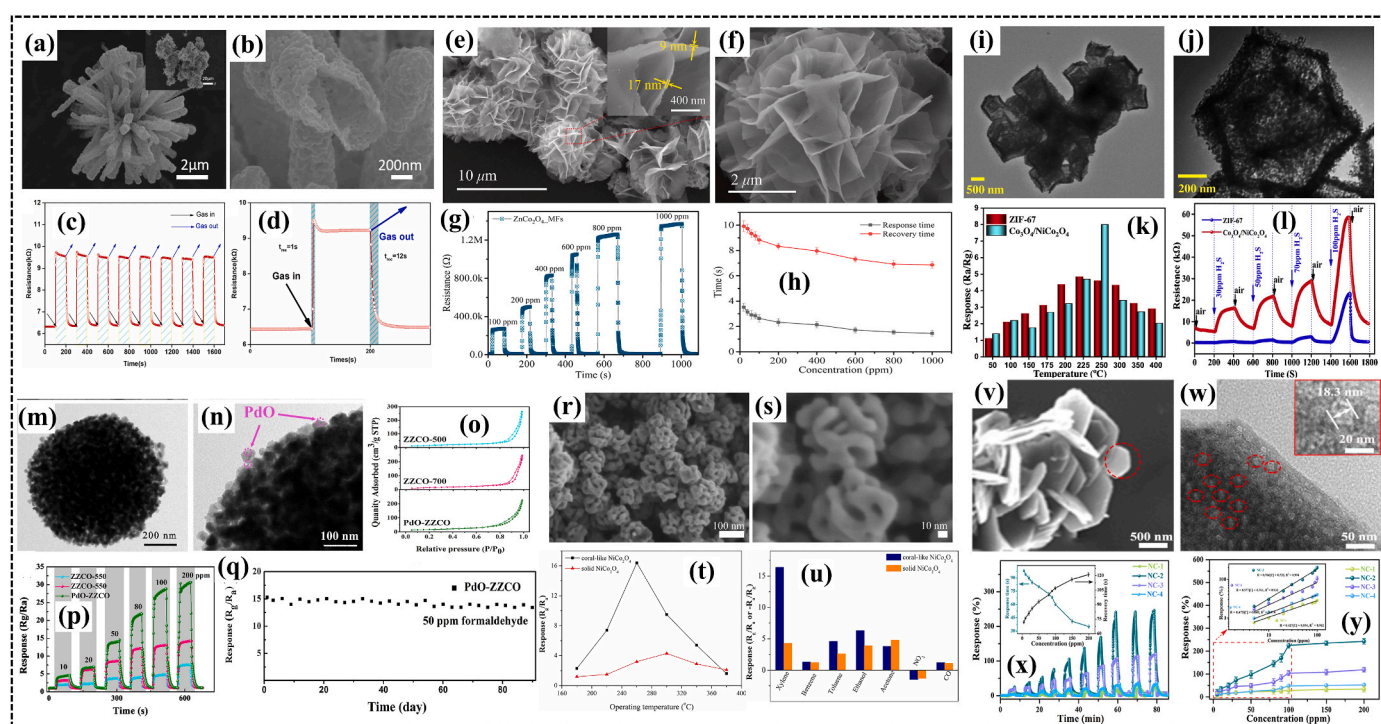


Fig. 6. (a) A typical FESEM image and (b) enlarged FESEM image of ZnCo₂O₄ hierarchical porous architectures (HPA). The inset is a low-magnification FESEM image of ZnCo₂O₄ HPA. (c) The ZnCo₂O₄ HPA sensor switches in the air and xylene atmosphere for eight cycles; (d) typical response and recovery behavior of ZnCo₂O₄ HPA to 200 ppm xylene at 260 °C. (Reproduced with permission from Ref. [130], Copyright© 2022, Elsevier). SEM images of (e) and (f): ZnCo₂O₄ microflowers (MFs). (g) Sensing curves of the sensor based on ZnCo₂O₄ MFs to 100–1000 ppm acetone gas at 200 °C. (h) Response/recovery time of ZnCo₂O₄ MFs based sensor to acetone gas at concentration range of 20–1000 ppm. (Reproduced with permission from Ref. [133], Copyright© 2021, Elsevier). (i, j) TEM images of Co₃O₄/NiCo₂O₄ composite. (k) Responses of ZIF-67 and Co₃O₄/NiCo₂O₄ to 50 ppm H₂S at different operating temperatures, (l) response curves at 100 ppm for H₂S at 30–100 ppm. (Reproduced with permission from Ref. [118], Copyright© 2020, Elsevier). TEM images (m, n) of PdO–ZnO/ZnCo₂O₄ microspheres. (o) N₂ adsorption-desorption isotherms of ZnO/ZnCo₂O₄-500 microsphere, ZnO/ZnCo₂O₄-700 microsphere and PdO–ZnO/ZnCo₂O₄ microsphere. (p) Dynamic response–recovery curves of the ZnO/ZnCo₂O₄-500, ZnO/ZnCo₂O₄-700 and PdO–ZnO/ZnCo₂O₄ MSs sensors to formaldehyde in the ranges of 10–200 ppm under 139 °C. (q) Long-term stability of the PdO–ZnO/ZnCo₂O₄ microsphere sensor to 50 ppm formaldehyde at 139 °C. (Reproduced with permission from Ref. [139], Copyright© 2020, Elsevier). (r) Low-magnification SEM image of the as-prepared, porous coral-like NiCo₂O₄ nanospheres, and (s) a representative SEM image of two samples of NiCo₂O₄ nanospheres. (t) Response of the sensors based on porous coral-like NiCo₂O₄ nanospheres and solid NiCo₂O₄ nanospheres to 100 ppm xylene. The response is plotted as a function of the operating temperature. (u) Response of the sensors based on porous coral-like NiCo₂O₄ nanospheres and solid NiCo₂O₄ nanospheres to 100 ppm of various gases at 260 °C, and 300 °C, respectively. (Reproduced with permission from Ref. [120], Copyright© 2018, Elsevier). (v) SEM and (w) TEM image of NC-2 (adding in 18 mmol 2-MIM). (x) Dynamic sensing transient curves under different concentrations (5, 10, 20, 30, 50, 80, 90, 100, 150, 200 ppm) n-butanol, inset is the response and recovery times of NC-2 sensors at different n-butanol concentrations. (y) Dot-line graph of concentration-dependent (5–200 ppm) response curves and the linear curves of concentration-dependent (5–100 ppm) response. (Reproduced with permission from Ref. [125], Copyright© 2020, Elsevier).

promising candidate material for gas detection [131]. The Ag/ZnCo₂O₄ microspheres combining self-template with annealing methods displayed superb formaldehyde sensing performance. The diameter of porous Ag/ZnCo₂O₄ microspheres assembled by massive nanoparticles ranged from 300 nm to 1 μm. The as-developed Ag/ZnCo₂O₄ sensors exhibited outstanding response of 11.18 which was better than pure ZnCo₂O₄ sensors towards 100 ppm formaldehyde at 99.3 °C. The enhanced sensing properties are derived from chemical sensitization and spillover effect. In detail, the reaction activation energy is reduced while oxygen species are prone to absorb on Ag nanoparticles which improved the formaldehyde sensitivity [132,179]. A novel double-shelled Co₃O₄/NiCo₂O₄ nanocage was synthesized by employing a template strategy. The as-developed products had a laminate nestlike shell which consists of massive nanosheets and the thickness of NiCo₂O₄ shell is about 100 nm while the thickness of internal Co₃O₄ shell is incommensurate as shown in Fig. 6i and j. The contrast experiment of sensing performance between ZIF-67 and Co₃O₄/NiCo₂O₄ was carried out. Owing to the hollow, penetrating and polyhedral nanostructure, high BET surface area (103 m²/g), low dissociation energy and high gas molecule entropy, the developed Co₃O₄/NiCo₂O₄ sensors have a better response towards 30–100 ppm H₂S as shown in Fig. 6k and l [118]. Koo et al. prepared Pd@ZnO/ZnCo₂O₄ hollow spheres on the basis of polystyrene sphere template and MOF mold methods. The Pd nanoparticles (2–3 nm) were loaded on the ZnO/ZnCo₂O₄ spheres with the diameter of 400 nm successfully. Outstanding response (69 % @ 5 ppm) towards acetone at 250 °C is revealed [138], which demonstrates that MOF-derived nanosized catalyst functionalized semiconductor oxide is a novel candidate for fabricating superior performance chemoresistance gas sensor. A novel ZnCo₂O₄ superstructure with hollow polyhedral characteristic was obtained via a ZIF-67/ZIF-8 sacrificial template route. The as-developed ZnCo₂O₄ sensor possessed a response of 14.2–20 ppm ethanol with a rapid response/recovery process (22/62 s) at 200 °C while the LOD can reach 1 ppm. Furthermore, the ZnCo₂O₄ sensor still had a high response of 17.6 with good repeatability to 200 ppm ethanol at 95%RH condition [140]. Based on the same research idea, Zhang et al. synthesized solid PdO/ZnO/ZnCo₂O₄ microspheres (Fig. 6m and n) for formaldehyde detection. Based on the catalytic impact of PdO, large BET surface area (Fig. 6o) and massive active adsorption sites, the PdO/ZnO/ZnCo₂O₄ possessed the excellent response (26.9 @ 100 ppm), fast response/recovery time (9/14 s), prime selectivity and low LOD (0.2 ppm) at 139 °C as shown in Fig. 6p and q [139]. Kaneti et al. reported a ZnO/ZnCo₂O₄ CO gas sensor with highly sensitivity and selectivity by template method. The hierarchical macroporous ZnO/ZnCo₂O₄ honeycomb shaped composites can be discerned. Unique macroporous structure constructed, high surface area, rapid electron transfer and stable p-n heterostructure synergistically improved the CO sensing properties [141].

Due to the high temperature stability, high electrical conductivity and excellent electrochemical properties, three-dimensional structure of NiCo₂O₄ is widely used in the preparation of high-performance gas sensors. For example, novel coral-like NiCo₂O₄ nanospheres with porous properties were synthesized by the group of Zhang et al. as shown in Fig. 6r and s. The coral-like NiCo₂O₄ nanospheres with the diameter of 60–80 nm owns the BET surface area of 83.8 m²/g. Towards 100 ppm xylene, the coral-like NiCo₂O₄ nanospheres displayed better response (4 times) than solid NiCo₂O₄ samples at 260 °C–300 °C while more superior selectivity property were shown in Fig. 6t and u [120]. Zhou et al. adopted PdO as a catalyst for synthesizing hollow NiO/NiCo₂O₄ nanocages with truncated feature by co-precipitation and etching-wet impregnation methods. The PdO–NiO/NiCo₂O₄ are hollow nanocubes with their interior expanding diagonally and OD PdO nanoparticles are uniformly dispersed on their surface. With the deopillant and porous nanostructure and catalytic action of PdO, PdO–NiO/NiCo₂O₄ truncated nanocage sensors exhibited the higher response of 6.7 towards 100 ppm acetone at 220 °C than NiO/NiCo₂O₄ solid nanocubes sensors [121]. The group of Hu et al. fabricated a highly sensitive nitrogen dioxide sensor

employing NiCo₂O₄/WO₃ p-n heterostructure. From the SEM results, rectangular-shaped WO₃ nanoflake and flower-shaped NiCo₂O₄ microspheres stick tightly to each other [123]. Meanwhile, a novel hierarchical NiCo₂O₄ microsphere was obtained by regulating the calcination temperature (300 °C, 400 °C and 500 °C) through a facile hydrothermal strategy and the NiCo₂O₄ microspheres were assembled by massive nanorods [124]. In addition to the NiCo₂O₄ microsphere, Dang et al. synthesized NiCo₂O₄ microflower with hierarchical feature applying for n-butanol detection as shown in Fig. 6v–y. By regulating the amount of 2-methylimidazole in synthesis, the NiCo₂O₄ microflower with the optimal morphology (adding 18 mmol 2-methylimidazole) have possessed high BET surface area (62.465 m²/g) and the numerous chemisorbed oxygen species of 58.78 %, while the corresponding gas sensor exhibited the high response of 240 %, fast response/recovery reaction (68/107 s) and brilliant long term stability [125]. The group of Zheng et al. reported a self-sacrificial template method to prepare NiCo₂O₄ spheres with different morphology by regulating the heating rate of calcination. The NiCo₂O₄ sample with a calcination rate of 10 °C/min had a double-shell hollow sphere structure while the diameter is 800–900 nm. Meanwhile, the gas sensor exhibited high sensitivity (23.3), fast response/recovery time (15.4/30.5 s) and excellent long-term stability towards 100 ppm xylene at the optimum operating temperature (240 °C). Compared with the single-shell hollow NiCo₂O₄ sphere and solid NiCo₂O₄ sphere, the better sensing performance is discerned. The higher specific surface area (93.5 m²/g) and unique nanostructure are beneficial to gas molecule transportation and diffusion [126]. Zhang et al. reported a NiCo₂O₄/PANI self-powered sensor for NH₃ detection. The NiCo₂O₄ solid microspheres with smooth surface are coated with PANI nanorods. The NiCo₂O₄/PANI sensor configured with eco-friendly triboelectric nanogenerator had the better response of 4.67 towards 20 ppm NH₃ than previous sensors conducted by self-powered voltage, which is due to the p-p heterojunction predominantly modified the sensor resistance as a hole pump, thus improving the sensitivity [127,180].

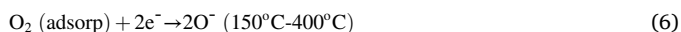
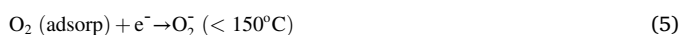
A novel hollow MCo₂O₄ (M = Mn, Ni, and Zn) multishelled twin nanospheres for formaldehyde and acetone detection was reported recently. The morphology of all products is formed by the adhesion of two multishelled spheres with the diameter of 400–500 nm to each other. Compared with the MnCo₂O₄ and NiCo₂O₄ gas sensors, ZnCo₂O₄ gas sensor have better response towards formaldehyde (203 @ 50 ppm) and acetone (318 @ 50 ppm) at 170 °C and 190 °C, respectively, which is attributed to the higher Co³⁺/Co²⁺ ratio and larger chemisorbed oxygen content [36]. Above all, the current research trend is applying MOF as template or precursors to synthesize 3D MCo₂O₄ based material on the account of its large surface area and easily tuned morphology by choosing various organic bridging ligands and metal ions [181–184], which provides effective technology for constructing exquisite structure like porous carbon, SMOX and SMOX/MOFs composites [185]. Meanwhile, as two efficient surfactants, PVP and DMF can be assembled into aggregates in the bulk precursor solution, which is conducive to the formation of 3D MCo₂O₄ based materials. By thermally decomposing the core during the self-assembly process, where the polymer part is located at the center and the inorganic part is developed onto the external surface, diverse 3D hollow nanostructures MCo₂O₄ based materials can be obtained. It is worth noting that solvothermal, hydrothermal reaction and template strategy are the most efficient strategies for producing the products possessing the high crystallinity, well-defined pores and large specific area. In detail, sensitivity of MCo₂O₄ gas sensor can be further improved by modifying crystal shapes by changing the preparation parameters, including reaction temperature and duration, pH value, solvent type, precursor concentration and the utilizing of catalysts. The specific gas sensitivity enhancement mechanism will be discussed in the next section.

4. Enhanced sensing properties and mechanism of MCo_2O_4 based gas sensor

While the synthetic material controls its morphology, the gas-sensing material can be suitably optimized to further enhance its response performance. Under the circumstance of impedance gas sensors and the specific detection process, the signal of the MCo_2O_4 gas sensor is mostly shown by the resistance value, while the analysis of gas sensing mechanism mainly depends on hole accumulation layer model [186,187], bulk resistance regulation theory [188] and gas diffusion control theory [189]. So essentially the gas-sensing performance can be enhanced by improving the carrier transport between the multiple materials in heterostructures. The group of Xu et al. constructed the grain size effect model, which indicated that gas sensitivity increased remarkably as the diameter of the crystal (D) equivalent to or smaller than 2L (L refers to the depth of surface space charge layer) [190]. However, the undesired increment of resistance aggravates the complexities of signal processing, suggesting an under-functioning gas sensor, which is known as receptor-transducer mismatch. The phenomenon limits the full use of semiconductor nanocrystals in practical gas sensors [191]. Elemental doping, functionalization of noble metals, construction of heterojunctions and tuning of synthesis parameters can all effectively modulate the efficiency and mode of electron transportation, resulting in substantial changes in the resistance of sensing materials and avoiding the mismatch between receptor and transducer. Researchers are currently using a combination of these approaches to create synergistic effects to enhance sensing performance.

4.1. The enhanced sensing strategies for pure MCo_2O_4 gas sensors

The sensing mechanism of MCo_2O_4 gas sensor is on the basis of the variation of resistance on the material's surface. In detail, the primary carriers of p-type semiconductor are holes. Massive ionized oxygen species (O^{2-} , O_2^- and O^-) will be generated through trapping electrons from the surface when the air comes in, which creates the insulation core with high resistance and generates hole accumulation phenomenon with high electrical conductivity. Assuming that the reducing gas injected into the chamber, the hole accumulation layer will assimilate electrons and the resistance will increase. After the stable resistance is acquired, pure air can be injected again for making the resistance back to baseline level [192–195]. Above all, one of the most widely recognized models of gas-sensing mechanisms, the adsorption and desorption model is obtained. It should be emphasized that the types of free oxygen species (O^{2-} , O_2^- and O^-) are highly correlated with the working temperature and the specific relationship is shown in Eqs. (4)–(7) [196–198].



For pure MCo_2O_4 gas sensors, numerous strategies were adopted to strengthen the gas sensing properties, including adjusting annealing temperature [109] or heating rate [126] and the pH of the precursor solution [113], changing the stoichiometric ratio [103], altering the amount of structure-directing agent [125] or surfactant agent [199], applying the novel template [94], etc.

For instance, Zhao et al. synthesized NiCo_2O_4 nanoplates by adjusting calcination temperature (300 °C, 500 °C, 700 °C). When the thermal treated temperature increases from 300 °C to 500 °C, the crystallinity of NiCo_2O_4 reaches maximum and the optimal sensitivity can be obtained. As the temperature rises further to 700 °C, a new phase NiO will appear, resulting in the impurity [109,200]. Furthermore, by changing the heating rate (5, 7, and 10 °C min⁻¹) in the synthesis of NiCo_2O_4

nanostructures, different NiCo_2O_4 morphology including the porous spheres, single-shelled and double-shelled hollow spheres can be obtained, respectively [126].

Adjusting the pH of the precursor is an effective way to changing the properties of sensing materials. By adding different amount of citric acid from 2.5 to 7.5 mmol, the grain size of CuCo_2O_4 samples prepared by Zhao et al. first decreased and increased later while the porosity is positively correlated with the amount of citric, which may attribute to the gas evolution and heat release in an acidic environment. In addition, too much citric acid added (7.5 mmol) will tend to form a secondary phase, resulting in impurity of sample [113]. Stoichiometric ratio regulation is determined to be a highly facile route to construct heterojunction and induce more oxygen vacancies. Gawande et al. regulated zinc and cobalt composition varying from 1 : 1 to 1 : 3 to obtain ZnCo_2O_4 nanoparticles, as the results demonstrated that certain sample (Zn : Co = 1 : 2.5) possessed the highest reducibility by temperature programmed reduction H_2 (TPR- H_2) spectra, indicating more available active sites, which benefited gas sensing properties a lot [103]. Adequate structure-directing agent will exert specific influence on the morphology of materials. By adding exactly-measured xylose which owns the agglomeration effect on nanomaterials, the NiCo_2O_4 , MnCo_2O_4 and ZnCo_2O_4 multi-shelled twin nanospheres were obtained while the enhanced sensing mechanism was finely illustrated as shown in Fig. 7a-d [36]. Furthermore, Zhang et al. modified the morphology and composition of NiCo_2O_4 by changing the content of the 2-methylimidazole, which is an effective ligand and structure-directing agent. As shown in Fig. 7e-f, too less 2-methylimidazole (12 mmol) used in synthesis process tends to form irregular hexagonal fragments which are scattered randomly while too much 2-methylimidazole (24, 36 mmol) will impede the growth of crystal and destroy the 3D flower-like structure. Only adding a certain amount of 2-methylimidazole (18 mmol) can lead to the appropriate degree of polymerization reaction driven by M – N bond (M = Ni^{2+} , Co^{2+}) (Fig. 7b) [125].

Through the research of Morán-Lázaro et al., the concentration of dodecylamine (5.4 mmol, 10.8 mmol, 16.2 mmol) exerts a significant effect on the nanostructure and particle diameter of ZnCo_2O_4 [201]. With the concentration of dodecylamine rose, the number of ZnCo_2O_4 nanoparticles gradually increased, while the size became smaller, indicating that dodecylamine inhibited the growth of ZnCo_2O_4 nanoparticles. The main functions of dodecylamine in synthesis are surfactant and agent reduction. As a reducing agent, the electrons of nitrogen can participate as a reducing agent to reduce cationic Co^{2+} to metal nanoparticles Co^0 while the action of surfactants is mainly worked by electrostatic interaction [199,202].

4.2. Elemental doping (noble metal or other elements)

Noble metal doping generally contains the decoration of Ag [132, 135] and Pd [91,138] on MCo_2O_4 semiconductor surfaces, which provides an efficient approach to improving sensitivity and lowering the operation temperature [203]. The superior sensing properties for gas sensors are ascribed to electronic and catalytic impact [204,205]. The functionalization of noble metal doping facilitating the sensing performance is primarily embodied in two ways: (1) modification of Ag and Pd effectively reduces the surface resistance of the material, therefore the lower working temperature detection can be achieved; (2) the load of Ag and Pd nanoparticles enlarges the specific surface area and improves the surface activity, therefore target gas has more contact area and faster reaction speed with the adsorbed oxygen energy onto the MCo_2O_4 semiconductor.

Except for doping precious metals, another effective strategy to improve gas sensitivity is to add heteroatomic doping of another metal. In detail, the grain sizes, porosity and specific surface area of the SMOX will change after metal heteroatoms are doped into the sensing material, the adsorption site and diffusion path of gas molecule can be optimized [206]. Furthermore, the facilitating of gas sensing properties by element

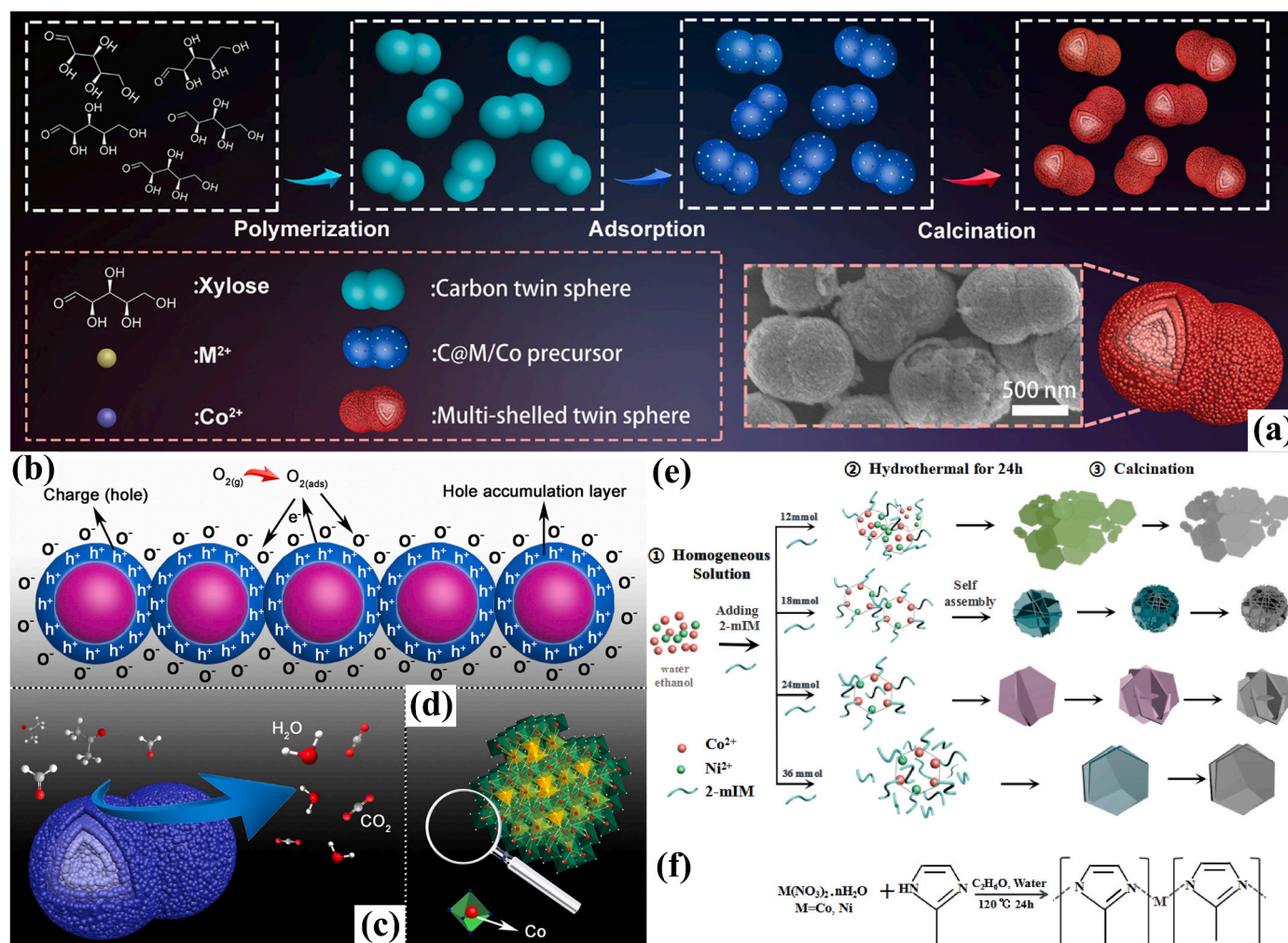


Fig. 7. (a) Synthetic process of $M\text{Co}_2\text{O}_4$ multi-shelled hollow twin spheres. (b–d) Schematic illustration of the sensing mechanism for sensors based on ZnCo_2O_4 materials. (Reproduced with permission from Ref. [36] Copyright© 2019, ACS Publications). (e) Synthetic mechanisms of samples with various amount of 2-methylimidamide and (f) Diagram of the formation process of M - N bond. (Reproduced with permission from Ref. [125], Copyright© 2020, Elsevier).

dopant other than noble metals is mainly reflected in increasing oxygen vacancy concentration and BET surface area, reducing the width of band gap, optimizing the electronic structure and maximizing band bending at the heterojunction interface. After doping 20 mol% of zinc to form $\text{Ni}_{0.8}\text{Zn}_{0.2}\text{Co}_2\text{O}_4$ nanorods, of which the sensing performance to NO_x became better while the higher BET surface area and lower band gap are primary factors [96].

4.3. Developing heterojunctions

Heterojunction is a kind of physical and electronic junction between two or more disparate solid-state materials in nature. Fermi levels of the diverse materials connect to each other while different or same type of charge carriers will vigorously participate during sensing process. Specifically, silicon p-n heterojunctions are widely used in basic elements of diodes, LEDs, transistors and gas sensors [171]. For binary or ternary semiconductor gas sensors, the improved sensing properties and mechanism will change according to the types of constructed heterojunctions.

Based on the type of heterojunction, since $M\text{Co}_2\text{O}_4$ refers to a p-type semiconductor, there are many reports that p-n [37,85–87,95,98,101,102,105,106,110,123,129,137,139,141,207,208] and p-p [76,97,118,119,121,122,131] heterojunctions are constructed while other types of heterojunctions, including graphene [84], reduced graphene oxide [92,112], multi-walled carbon nanotubes [107], carbon cloth [115], g- C_3N_4 [81], polyaniline [127] and polypyrrole [77] are explored as well.

As shown in Fig. 8a, the carriers in the p-n heterojunction include free electrons and holes, which will both flow resulting from the differences in Fermi energy levels of two semiconductors, resulting in a faster carrier transport rate than p-p or n-n heterojunctions. Negative electrons will move from the conduction band of higher Fermi level to another which owns the lower one and the positive holes will transfer in the opposite direction. Finally, an electron depletion layer and a hole accumulation layer will construct at the interface of two materials.

Liu et al. established $\text{ZnO}/\text{ZnCo}_2\text{O}_4$ p-n heterostructure for ethylene glycol sensing enhancement. In detail, the electrons transferred from ZnO to ZnCo_2O_4 and the holes moved in an opposite direction as ZnO possessed the higher Fermi energy level. After the movement of carriers, the Fermi energy levels of two materials tend to coincide, while hole and electron depletion layers are formed. Meanwhile, inner self-established electric field was formed at the contact interface between ZnCo_2O_4 and ZnO , while the boundary barrier was established at the p-n interface and band bending in the semiconductor can be discerned [95]. Gas molecules will generate electrons to the p-n heterostructures when the sensing layer exposed to ethylene glycol, and rapid changes in electron concentration can greatly alter the thickness of the electron depletion layer, further enhancing the sensing performance as shown in Fig. 8c-f [87].

For p-p heterojunction according to Fig. 8b, holes act as charge carriers. As two p-type semiconductors contact with each other, holes will transfer out of the valence band until their Fermi levels are equal

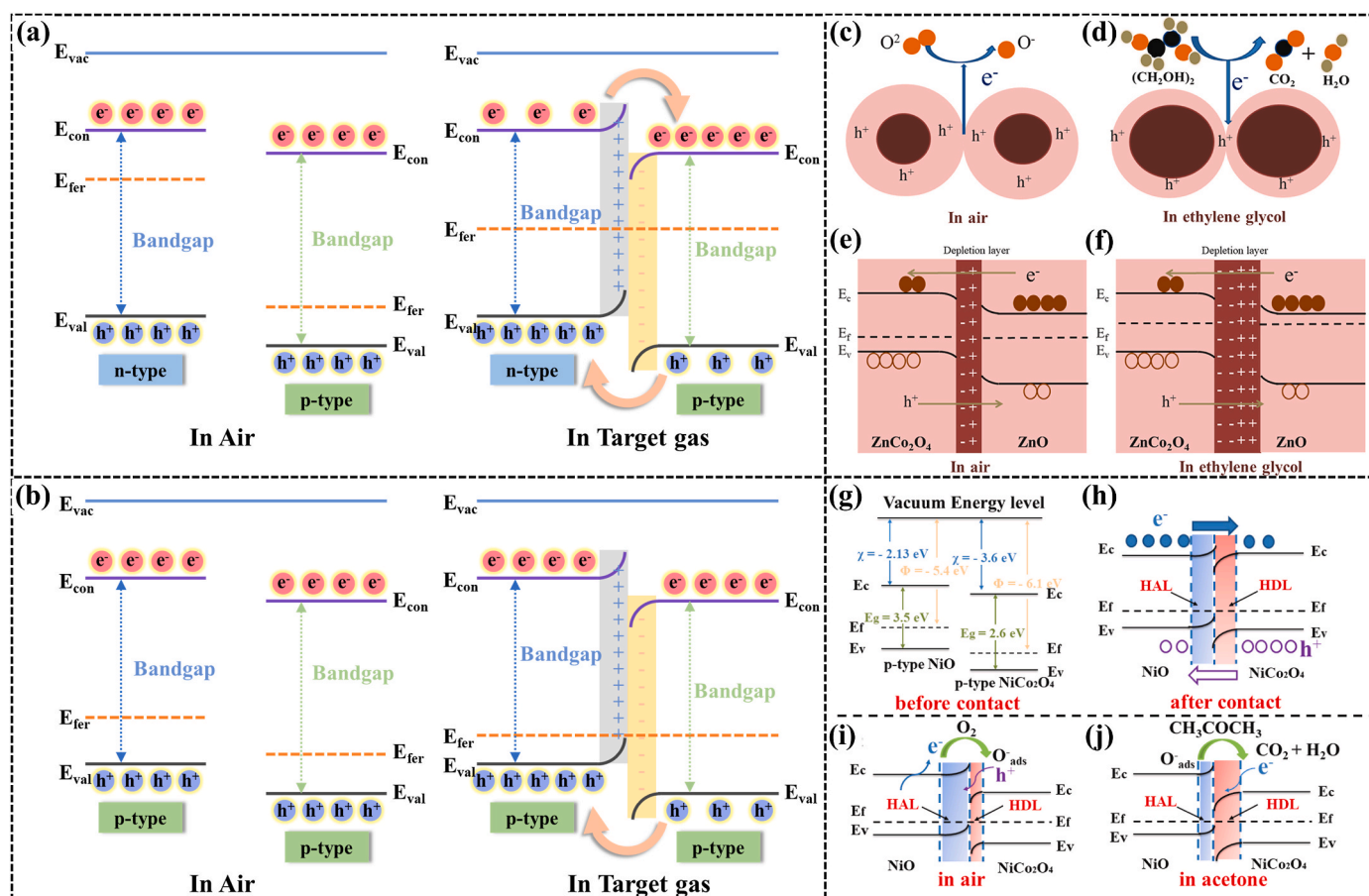


Fig. 8. Schematic diagrams of the energy bands based on (a) p-n heterojunction and (b) p-p heterojunction of $M\text{Co}_2\text{O}_4$ based SMOX. (c–f) The schematic illustration of possible gas-sensing mechanism for ZnCo_2O_4 and $\text{ZnO}/\text{ZnCo}_2\text{O}_4$. (Reproduced with permission from Ref. [87], Copyright© 2022, Elsevier). Schematic illustration of proposed gas sensing mechanism: (g, h) the energy band diagram of the NiO and NiCo₂O₄ before and after contact, respectively; (i, j) the energy band diagram of NiO/NiCo₂O₄ heterojunction expose to air and acetone, respectively. (Reproduced with permission from Ref. [76], Copyright© 2021, Elsevier).

due to their differences of Fermi energy level. Eventually, the absorbed holes will construct a layer of positively charged hole accumulation at the interface of two semiconductors. Meanwhile, a semiconductor that loses its holes will form a negatively charged hole depletion layer. Eventually, the constructed heterojunctions will efficaciously facilitate the accumulation of free electrons at the sensing surface layer, elevate the adsorption of oxygen molecules and increase the height of the energy barrier (Fig. 8g–j).

A novel NiO/NiCo₂O₄ p-p heterojunctions were prepared by Wang et al. for xylene detection. As the band gap width and work function of NiO are 3.5 eV and 5.4 eV while the corresponding value for NiCo₂O₄ is 2.6 eV and 6.1 eV, respectively [121], the free electrons will move from NiO to NiCo₂O₄ until the equal Fermi level is formed. Meanwhile, the hole accumulation layer and depletion layer will generate. Once the NiO/NiCo₂O₄ exposed to xylene, oxygen species will react with xylene molecules and motivate electrons back to the NiO/NiCo₂O₄, therefore tipping the dynamic carrier equilibrium between NiO and NiCo₂O₄, resulting in the decrease of hole accumulation layer [119].

Moreover, classification can be made based on whether $M\text{Co}_2\text{O}_4$ was material main body or appendages. As the $M\text{Co}_2\text{O}_4$ was served as a load recipient, in particular, graphene, as a representative 2D nanomaterial, has been extensively utilized in fabricating SMOX gas sensors for its excellent electronic properties (mobility of $15000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ under normal temperature and pressure), superior heat conductivity, high stability and ease of chemical functionalization [209]. Rathinavel et al. constructed FeCo₂O₄/graphene heterojunction, which owns the large BET surface area of $93.2\text{ m}^2/\text{g}$, boosting the transportation of gas molecules, thus enhancing the efficiency of detection. Furthermore, the

transitory retention of wave generated at the converted envelope-air interface when the target gas molecules interact with the graphene also plays a crucial role in promoting ethanol and acetone sensitivity [84]. Marimuthu et al. constructed a FeCo₂O₄/rGO nanostructures for NH₃ detection [92]. The hybridizing rGO triggering the active electrons migration between the sensing elements and the electrodes is a key to enhance sensing performance [210].

5. Issues and challenges

At present, based on the spinel type and special chemical structure of $M\text{Co}_2\text{O}_4$, it is widely used in gas detection which were listed in Tables 3 and 4. Beyond that, diverse types of $M\text{Co}_2\text{O}_4$ nanomaterials are also used in catalysis, energy storage and other fields as shown in Fig. 1a. Meanwhile, the $M\text{Co}_2\text{O}_4$ characteristics of low synthesis cost, strong structural stability, vibrant electrochemical performance and relative lower resistance, further ensure the effective detection of diverse toxic and explosive gases. Therefore, we intensively reviewed the morphology regulation and gas-sensitive enhancement of $M\text{Co}_2\text{O}_4$ in this paper. Although considerable progress has been achieved in the current development of $M\text{Co}_2\text{O}_4$ based gas sensors, numerous problems and challenges still need to be addressed to achieve high sensitivity, swift respond and recover process, outstanding selectivity and stability, especially in the future. In order to solve the current technical difficulties in this field, the following issues need to be further studied.

- (1) Towards $M\text{Co}_2\text{O}_4$ sensors, they remain a huge challenge to detect target gas effectively in complicated gas environment due to the

intrinsic sensing mechanism of MCo_2O_4 . For reasons of overcoming the disadvantage of poor selectivity in complex atmospheric environments, disparate MCo_2O_4 sensors can be integrated into a single array. Combining with the principal component analysis, linear discriminant analysis and other pattern recognition algorithms, the intelligent MCo_2O_4 sensor arrays can be utilized to deal with the experimental data to differentiate diverse gases [135].

- (2) At present, another difficulty is that the power consumption of MCo_2O_4 gas sensors is too high (Table 4), mainly reflected in long recovery time and exorbitant working temperature. The construction of multiple composite materials is an effective approach to reducing the working temperature of the MCo_2O_4 sensor and improving the gas response value [84,107]. By further surmounting the limitations of elevated operating temperature on the fabrication and application of conventional devices, room temperature MCo_2O_4 sensor will implement satisfying strategies with lower power consumption, higher integrated density and superior precision.
- (3) In general, a major challenge for MCo_2O_4 sensor research is the practical application. When using MCo_2O_4 sensor for agricultural product quality detection (H_2S , NH_3) [90,102], environmental monitoring (HCHO) [98], nondestructive disease detection (acetone) [77], the relative low sensitivity, insufficient stability and high operating temperature restrict the wearability, small size and low power consumption of MCo_2O_4 sensors in actual manufacturing. Therefore, in the future, efforts should be devoted to the development of more tolerant and environmentally recognizable components and nano-transducers of MCo_2O_4 .

6. Conclusion and perspectives

Obviously, spinel type MCo_2O_4 based materials played a crucial role in the development of resistance-type and current-type sensors for inflammable and poisonous gases detection. The dimension and morphology regulation strategies of as-prepared material and coating technology were adopted to strengthen the sensing properties, especially the sensitivity and respond speed. In detail, researchers applied selecting elemental dopants, noble metal functionalization and hetero-junction construction to promote the overall performance of MCo_2O_4 gas sensors. Wet chemical method, co-precipitation and self-templating methods are three common ways to prepare diverse MCo_2O_4 nanostructures. Different MCo_2O_4 nanostructures often represent diverse sensing properties while the sensing functional material which researchers desired should possess relatively large BET specific surface area, sufficient active adsorption sites and excellent physicochemical stability. MCo_2O_4 hollow nanospheres, nanoflowers and hierarchical structures are obtained by some research fellows which can supply high specific surface area and thus enhancing the gas sensors' stability.

With the low fabrication cost, automated manufacturing capabilities and an extensive range of the detected gases, we believe the room-temperature and subzero-temperature MCo_2O_4 sensors will dig up all-time chances for wafer-level fabrication and high-density integration of on-chip MCo_2O_4 sensor arrays. This will enable ubiquitous and credible machine olfaction capabilities beyond the human olfactory powers and further increase the practical application ability of MCo_2O_4 sensor. In addition, for reducing the operating temperature, diverse progressive materials including transition metal dichalcogenides, covalent organic framework and MXenes can be used to prepare MCo_2O_4 sensor. Based on the existing reports, the synthesis of MCo_2O_4 gas sensors by microwave hydrothermal method, combination of traditional hydrothermal and other deposition strategies should be further studied. Furthermore, on the basis of the unique advantage of high scalable and fast response rate, MCo_2O_4 based sensors provide a superb promise for promoting them with the flexible, wearable, self-powered, large-scale and low-cost feature, which will also be the focus of future research.

CRediT authorship contribution statement

Zichen Zheng: Writing – review & editing, Writing – original draft, Methodology, Investigation, Data curation, Conceptualization. **Kewei Liu:** Writing – review & editing, Investigation. **Yiwen Zhou:** Writing – review & editing. **Zicong Zhang:** Writing – review & editing. **Hongyuan Su:** Writing – review & editing. **Xudong Nie:** Writing – review & editing. **Marc Debliquy:** Writing – review & editing, Supervision. **Zexin Yu:** Writing – review & editing. **Chao Zhang:** Writing – review & editing, Visualization, Validation, Supervision, Resources, Funding acquisition.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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